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3 Efficient recovery of lithium from spent lithium-ion battery

- 4 raffinate by Mn and Al-based adsorbents: pretreatment,
- 5 adsorption mechanism, and performance comparison
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19 Abstract:

As a strong wave of retired lithium-ion battery approaches, lithium extraction from 20 spent lithium-ion battery raffinate (SLR) becomes increasingly critical for 21 environmental protection and for sustainable lithium supply. To understand the factors 22 23 that affect maximum recovery of lithium from SLR, the organic and inorganic components of SLR were initially determined. The organic matter content (up to 760.5 24 mg/L) seriously impacted the recovery rate of lithium. Therefore, SLR was managed 25 with a series of pretreatment techniques, including coagulation, biochar aerogel 26 27 adsorption, and ultrafiltration, achieving more than 84.3% removal of organic substances. H_{1,33}Mn_{1,67}O₄ and Li/Al layered double hydroxides adsorbents were then 28 synthesized by solid state reaction method and hydrothermal method, respectively, 29 30 granulated into spheres with a PVC skeleton, and applied to recycle lithium from pretreated SLR in a fixed bed adsorption column. The results indicated that both Mn 31 and Al-based adsorbents exhibited rapid adsorption kinetics, reaching saturation within 32 33 2 h. The Mn-based adsorbent exhibited superior adsorption selectivity for Li⁺ and higher Li⁺/Na⁺ separation factor (α_{Na}^{Li}) compared to Al-based adsorbent, with partition 34 coefficients and α_{Na}^{Li} values equal to 6.62 mL/g, 8.79 for the former material, and 4.92 35 mL/g, 8.17 for the latter. On the other hand, the Al-based adsorbent displayed better 36 stability with negligible Al loss, while Mn loss from the related adsorbent was less than 37 0.2% in every adsorption-desorption cycle. Notably, both adsorbents demonstrated 38 excellent reusability with their adsorption capacity maintained after twenty adsorption-39 desorption cycles. 40

41 Key words:

- 42 Spent lithium-ion battery raffinate; Pretreatment; Lithium recycling; Mn/Al-based
- 43 adsorbent; Comparative analysis.

1. Introduction

As a critical rare metal [1], lithium has extensive application in various industrial 46 applications, chiefly, in lithium batteries [2] due to its light mass density (0.534 g/cm^3) 47 [3], high electrode potential (-3.05 V) [4], low equivalent weight (6.94 g/Faraday) [5], 48 and long service life. In the context of "dual carbon" objective, the market for new 49 energy vehicles powered by lithium-ion batteries (LIB) has experienced exponential 50 growth as a response to the scarcity of fossil fuels and climate change [6], resulting in 51 a substantial increase in the production of lithium-ion batteries. By 2030, China's 52 53 electric vehicle production is projected to reach 15 million per year [7], accounting for 45% of the global total electric vehicles based on reports from the International Energy 54 Agency [5]. However, after a lifespan of 3 to 10 years [8], a great deal of electric 55 56 vehicles will be retired, leading to a significant accumulation of spent LIBs [9,10]. The China Society of Automotive Engineers estimates that, by 2030, the amount of 57 decommissioned power batteries will reach 3.5 million tons. However, the global 58 59 lithium recycling rate from LIBs is below 1% [11]. Survey of Development Research of the State Council indicates that the standardized recovery rate of China's new energy 60 61 vehicle power battery is less than 25% in 2023.

If handled improperly and if disposed as domestic waste, the heavy metals and toxic electrolytes in spent LIBs will cause environmental problems and threats to human health [12,13]. However, a large amount of valuable key metal ions, such as Li, Co, Ni, and Mn in spent LIBs have high grade and are potentially easier to recover than extraction from natural ores, which makes spent LIBs a secondary, economical source of valuable minerals [10,14,15]. In particular, the content of Li in spent LIBs reaches
up to 5 to 7 wt% [16], which is substantially higher than any other lithium resource.
Therefore, recovering rare metals from spent LIBs is of great significance for mitigating
environmental pollution and ensuring a sustainable supply of lithium as raw material
for the manufacturing of new batteries.

72 Hydrometallurgy [17], pyrometallurgy [18], biometallurgy [19]. and electrochemical extraction [20] have been employed in research and industrial settings 73 for metals recovery from spent LIBs. Among these methods, hydrometallurgy 74 75 combined with leaching and extraction is the most common method adopted for metal ion recycling [21]. The hydrometallurgy process requires a large volume of strong acid 76 and water for leaching, followed by pH adjustment, typically performed by addition of 77 78 NaOH, to extract Mn, Co, and Ni in sequence [22]. Spent lithium-ion battery raffinate (SLR) is the leachate of spent LIBs obtained after the extraction of Mn, Co, Ni, and Li. 79 It contains large concentrations of Na and residual Li, and it is characterized by high 80 81 values of total dissolved solids (TDS) and total organic carbon (TOC). The lithium concentration in the raffinate can reach up to 1000 mg/L, higher than that found in 82 traditional salt lake brine, seawater, or shale gas wastewater. It is thus potentially 83 feasible to further extract Li from the raffinate to further increase lithium supply and to 84 85 alleviate environmental pollution.

Currently, most literature studies on this topic focus on lithium recycling directly from the spent LIBs or from salt lake brines, with limited attention paid to lithium recycling from RSL. Presently, methods for lithium recovery from aqueous lithium resources include evaporation and precipitation [23], solvent extraction [24], adsorbents adsorption [25], membrane treatment [4,26], electrochemical methods [27], or a combination of the above. Among these approaches, membrane and electrochemical methods are not readily applicable, due to the high TDS and TOC content of RSL. Instead, adsorbents have gained widespread attention due to their simplicity in preparation, low synthesis contamination, high adsorption capacity and selectivity, as well as reusability and economic potential [25].

There are three main types of inorganic metal-based lithium ion adsorbents 96 97 extensively applied for lithium extraction, including layered Al-based adsorption materials, Mn-based ion sieves, and Ti-based ion sieves [25,28]. The lithium adsorption 98 process of these metal-based ion sieves is mainly governed by structural memory effect 99 100 [28]. Ti-based ion-exchange sieves have strong Ti-O bonds, which endow them with a stable framework, excellent acid endurance, and exceptional Li adsorption ability [29]. 101 However, high cost thwarts the practical application of these Ti-based materials. Al-102 103 based adsorbents have exhibited better potential in industrial applications, owing to their technological maturity, inexpensive raw materials, simple synthesis procedure, 104 and mild preparation conditions [30]. For example, Zhong et al [31]. synthesized two-105 dimensional hexagonal Li/Al layered double hydroxides by coprecipitation to recover 106 107 Li from Qarhan salt lake brine at ambient temperature, with an adsorption capacity of 7.27 mg/g. Sun et al. prepared Li/Al layered double hydroxides and extracted Li from 108 Qaidam salt lake brine by hybrid method of reaction-coupled separation, reaching Li 109 loss lower than 3.93% [32]. In addition, Mn-based adsorbents present excellent 110

adsorption performance, selectivity, regeneration properties, and low cost [33]. Compared with $H_{1.6}Mn_{1.6}O_4$ and $HMnO_2$ [34], the synthesis of $H_{1.33}Mn_{1.67}O_4$ is simpler due to lower energy requirement [35]. For instance, Tian et al. synthesized $H_{1.33}Mn_{1.67}O_4$ and recovered lithium from shale gas wastewater, reaching a Li partition coefficient higher than 1073 mL/g [36]. Both Al-based and $H_{1.33}Mn_{1.67}O_4$ adsorbents have reliable application potential for lithium recovery from SLR.

This study discusses and evaluates the technical feasibility of lithium recycling 117 from SLR, as a novel potential source of lithium. We first investigate the characteristics 118 119 of SLR and the effect of pretreatment, including coagulation, biochar adsorption, and ultrafiltration, on organic pollutant removal. Then, we discuss the synthesis of Li/Al 120 layered double hydroxides and of Li1.33Mn1.67O4 adsorbent precursors by one step 121 122 method and solid-phase reaction method, respectively, and the granulation of the adsorbents powder into spheres aimed at lithium recycling from SLR in a fixed bed 123 adsorption column. The adsorption kinetics, adsorption selectivity, multi-stage 124 adsorption performance, and regeneration capability of lithium recovery from SLR by 125 both Mn and Al-based adsorbent granules are evaluated. Adsorbents are also assessed 126 in terms of adsorption/desorption performance during 20 cycles of utilization. The main 127 objective is to verify the technical feasibility of lithium recovery from real SLR by 128 pretreatment combined with adsorption, which has important repercussions on further 129 development of lithium recycling strategies. 130

131 **2. Materials and methods**

132 2.1. Materials

133	AR-grade MnCO ₃ , Li ₂ CO ₃ , AlCl ₃ ·6H ₂ O, NaOH, and HCl and were purchased
134	from Kelong Chemical Co., Ltd. (Chengdu, China) Polyaluminum chloride (PACl) was
135	purchased from Liming chemical plant (Zigong, China). Polyvinyl chloride (PVC) and
136	dimethylacetamide (DMAC) were supplied by Macklin Biochemical Co., Ltd.
137	(Shanghai, China). The raw SLR solution was provided by Botree Recycling
138	Technologies Co., Ltd. (Suzhou, China). All chemicals were used as received.
139	2.2. Synthesis of Mn and Al-based adsorbent powders precursor
140	The Mn-based adsorbent precursor for Li1.33Mn1.67O4 was prepared by solid state
141	reaction method, with synthesis processes detailed in our previous work [36]. Briefly,
142	a proper amount of MnCO3 and Li2CO3 powders with Li/Mn molar ratio of 1.33:1.67
143	were well-mixed in a ceramic mortar, then calcinated in a tube furnace at 500 °C for 4
144	h under ambient air with heating rate of 3 °C/min, followed by cooling to ambient
145	temperature. The Al-based lithium adsorbent precursor was synthesized by a one-step
146	method. AlCl ₃ ·6H ₂ O and LiCl were dissolved in ultrapure water at an Al/Li mole ratio
147	of 1.3:1. NaOH solution with a concentration of 5 mol/L was added dropwise to the
148	above solution with assistance of a peristaltic pump at a rate of 3.5 mL/min while
149	stirring at a speed of 150 r/min. The reaction temperature was 75 °C and the terminal
150	pH was 6-7. The obtained solid was separated from the aqueous solution by
151	centrifugation and dried to obtain the Al-based adsorbent precursor.
152	2.3. Granular Mn and Al-based adsorbent preparation

Adsorbent precursor powders were granulated into spheres by anti-solvent method,
as follows: 3.5 g PVC was added in 40 mL DMAC solution under magnetic stirring at

155 60 °C until dissolved completely. Then, 4.0 g adsorbent precursor powder was added to 156 this solution to form a uniform slurry. After that, the slurry was dripped by a syringe 157 pump into ultrapure water to form granules. The obtained spheres were rinsed in 158 ultrapure water to remove DMAC. Eventually, the Mn-based spheres and Al-based 159 spheres were separately eluted in 0.5 mol/L HCl solution and deionized water for 5 h 160 to eliminate the template Li⁺ ion and to obtain the granular adsorbent.

161 2.4. Experimental set-up

The lithium recycling system consisted of a pretreatment system and a fixed bed 162 163 adsorption stage, as depicted in Fig. 1. All the tests were conducted under ambient temperature of roughly 25 °C. The raw raffinate of spent lithium-ion battery was 164 pretreated by coagulation, biochar aerogel adsorption, and ultrafiltration (UF), in 165 166 sequence. A commercial PACl coagulant with concentrations of 500, 1000, 1200, 1500, 2000 mg/L was tested to determine the optimum dosage. All the coagulation tests were 167 operated in a programmable jar test apparatus (ZR4-6, Zhongrun Water Industry 168 169 Technology Development Co., Ltd, China). The tests involved fast stirring at 200 rpm for 2 min, followed by slow stirring at 40 rpm for 20 min and static precipitation for 30 170 min. The obtained supernatant liquid was collected for the subsequent tests. The biochar 171 aerogel was synthesized according to the protocol described in our previous works 172 [37,38]. The biochar aerogel adsorption process was conducted in a shaker at 150 rpm 173 for 2 h with the goal to remove additional organic matter on the clarified supernatant 174 from the previous coagulation step. Subsequently, the solution was pumped into the 175 ultrafiltration device to eliminate turbidity and further organic pollutant. The UF 176

treatment was performed in a cross-flow equipment at a constant pressure of 1.0 bar
with an effective membrane area of 17.6 cm². The UF process consisted of eight cycles,
each cycle including 1 h filtration and 5 min back washing with ultrapure water.
The lithium recovery procedure was conducted in a fixed bed column. Considering
the concentration of Li⁺ ion in the SLR and the adsorption capacity of the adopted

adsorbent based on previous studies [39], the volume ratio between adsorbent granules 182 and SLR was set to 1:1.5. To prepare the fixed bed for adsorption, 10 mL of Mn or Al-183 based granules were placed in an adsorption column, while 15 mL of pretreated SLR 184 185 was put into a beaker as a feed solution to the column. Adsorption was conducted in a closed-loop system, with the pretreated SLR circulating between the adsorption column 186 and the beaker for 5 h. Adsorption kinetic behaviors were explored by pseudo-first and 187 188 second-order models. The amount of adsorbed lithium at various times was computed with Eq. (1): 189

$$q_t = \frac{(C_0 - C_t)V}{m} \tag{1}$$

190 where C_0 , *m*, and *V* represent the initial lithium concentration, the mass of the adsorbent 191 granules, and the solution volume, respectively. q_t and C_t stand for the adsorbed lithium 192 amount and the Li⁺ concentration in the solution at time *t*.

Adsorption selectivity of lithium with respect to other interfering metal ions was assessed by determining the partition coefficient (K_d), the concentration factor (CF), and the separation factor (α_{Me}^{Li}), calculated with the following Eqs. (2-4):

$$K_{\rm d} = \frac{C_0 - C_e}{C_e} \times \frac{V}{m} \tag{2}$$

$$CF = \frac{q_e(Me)}{C_0(Me)} \tag{3}$$

$$\alpha_{Me}^{Li} = \frac{K_d(Li)}{K_d(Me)} \tag{4}$$

The reusability of adsorbent granules was assessed by adsorption-desorption cycles, including adsorption, desorption, and ultrapure water washing steps. Specifically, the desorption process was carried out in the same way of the adsorption step described above, but with a solution consisting of 0.5 mol/L HCl for Mn-based adsorbent and ultrapure water for Al-based adsorbent, which recirculated between a beaker and the column for 5 h.



Fig. 1: Schematic flow diagram of the lithium recovery system comprising
pretreatment and adsorption steps.

205

206 2.5. Analytical methods

Dissolved organic carbon (DOC), UV₂₅₄, and zeta potential of raw solution and coagulation supernatant under various coagulant dosage were measured with a TOC analyzer (TOC-L, Shimadzu, Japan), UV spectrophotometer (Orion AquaMate 8000, Thermo Fisher, USA), and Zeta-size nano instrument (Nano-ZS90, Malvern, UK), respectively. The three-dimensional excitation-emission matrix fluorescence spectra of the raw SLR and the solutions obtained after each treatment step were analyzed with a
fluorescence spectrophotometer (F7100, Hitachi, Japan). The ion concentration in
solution was measured with an atomic absorption spectrometer (AAS, PinAAcle 900T,
PerkinElmer, USA) and with ion chromatography (Dionex Integrion HPLC, Thermo
Fisher, USA).

The crystalline structure of two types of adsorbent powders, their morphologies, 217 and the elemental distribution of granules were characterized with X-ray diffraction 218 (XRD, DX2700, China) and field emission scanning electron microscopy (FE-SEM, 219 220 Hitachi, Japan). Nitrogen adsorption-desorption isotherms were obtained with an ASAP 2460 analyzer (Micromeritics, USA) under liquid N₂ atmosphere (77 K). The specific 221 surface area and pore size distribution of the granules were determined applying the 222 223 Brunaue-Emmett-Teller equation and the Barrett-Joyner-Halenda (BJH) method, respectively. 224

225 **3. Results and discussion**

226 3.1. Performance of SLR pretreatment and solution characteristics

The characteristics of raw SLR solution are summarized in Table 1. Effects of PACl concentration on the TOC, UV₂₅₄, and zeta potential of the SLR solution are instead presented in Fig. 2. When PACl dosage increased from 0 to 1500 ppm, TOC and UV₂₅₄ values drastically decreased from 760.5 mg/L to 204.3 mg/L, and from 0.023 cm^{-1} to 0.009 cm⁻¹, respectively. When PACl content further increased to 2000 ppm, TOC and UV₂₅₄ values diminished only slightly to 194.8 mg/L and 0.007 cm⁻¹. Therefore, the optimum PACl dosage was determined to be 1500 ppm. The addition of

PACl decreased the absolute value of the zeta potential, which increased from -10.9
mV (raw SLR) to -3.4 mV (PACl content of 1500 ppm) at the unadjusted pH equal to .
A higher PACl concentration likely resulted in more hydroxide precipitation from the
suspension [40]. Overall, PACl had remarkable performance on organic matter removal,
removing up to 73% of organic matter and representing a suitable choice for SLR
coagulation due to the availability and low cost of the coagulant.

241

Table 1: Characteristics of raw spent lithium-ion battery raffinate.

Parameter	Li (mg/L)	Na (mg/L)	K (mg/L)	Ca (mg/L)	Mg (mg/L)	TOC (mg/L)	UV ₂₅₄ (cm ⁻¹)	TDS (mg/L)	pН
Values	11,470	31,210	126.4	54.56	57.50	769	0.023	112,000	5.54







Fig. 2: Variation of TOC, UV₂₅₄, and zeta potential of the SLR solution with the

increased dosage of PACl coagulant. Lines connecting the data points are only a guide

246

247

248 Upon clarification, the supernatant was further treated with biochar aerogels and

for the eye.

UF membranes. The influence of coagulation and biochar aerogel adsorption processes 249 on UF membrane performance is presented in Fig. 3a. The use of raw SLR as feed 250 251 solution caused the severest UF membrane fouling, with the membrane flux declining severely to 45.6% of the initial flux, and then recovering to 61.7% of the initial flux at 252 253 the end of the first cycle of filtration and backwashing. After 8 cycles, the observed flux was only 33.9% of that measured with a pristine membrane, due to the high TOC value 254 of the raw SLR solution and the suspended materials blocking the membrane pores, 255 which was verified with SEM imaging, as shown in Fig. S2. The coagulation treatment 256 257 somewhat alleviated the UF membrane fouling. In particular, the membrane flux decreased by roughly half in the first filtration cycle, and recovered to 62.5% of the 258 initial flux after backwashing. In this case, the flux decline was mainly attributed to 259 260 scaling and inorganic components depositing on the UF membrane surface (see Fig. S2). The additional aerogel adsorption treatment further reduced membrane fouling. At 261 the end of first cycle, the flux declined to 63.3% of the initial flux and, after 262 backwashing, the flux recovered to 76.0% of the initial value. At the end of the eighth 263 cycle of filtration, the flux was approximately 55% of the flux measured with a pristine 264 membrane. 265



Fig. 3: (a) UF membrane performance for raw and pretreated raffinate of spent lithium-ion battery, and (b) related TDS, TOC variations.

As shown in Fig. 3b, the TOC decreased to 121.8 mg/L and 119.3 mg/L after 269 270 aerogel adsorption and UF filtration, respectively. Throughout the various pretreatment processes, the TDS concentration remained constant; see Fig. 3b. The evolution of 271 organic matter in the SLR solution was further investigated with three-dimensional 272 273 fluorescence. Excitation-emission (EEM) spectra of raw SLR and the product stream from every processing stage are reported in Fig. 4(a-d), with the spectra divided into 274 five regions based on previous research analyses [41]. One main peak of emission 275 276 intensity was observed in all solution samples, centered between region IV and region V, which are related to soluble microbial byproducts and humic-like substance, 277 respectively. The fluorescence regional integration (FRI) analysis indicated that DOM 278 was mainly composed by soluble microbial byproducts (33.7%) and humic-like 279 substances (63.2%); see Fig. 4e. Note that the organic matter composition may be 280 altered in previous processing steps, including the organic extraction agent used for the 281 recovery of rare earth metals from the spent lithium-ion battery. 282

The raw SLR solution had the highest fluorescence intensity. Upon treatment through PACl coagulant and biochar aerogel adsorption, the peak intensity declined sharply, and then it decreased slightly upon UF filtration. This result is consistent with the TOC removal efficiency reported in Fig. 3b and with the trends in FRI values, which suggest especially high removal of humic-like substances. Indeed, the biochar aerogel typically exhibits a high affinity with humic-like matters, promoting their removal through hydrogen bonding and hydrophobic interaction.





Fig. 4: Three-dimensional excitation-emission (EEM) fluorescence spectra of all SLR
samples: (a) raw water, (b) supernatant from the coagulation step, (c) solution upon
biochar aerogel adsorption, and (d) UF filtrate. (e) F_{max} of EEM spectra obtained with
fluorescence regional integration analysis.

295 3.2. Characteristics of the Mn and Al-based adsorbents

Fig. 5a and 5e present representative XRD patterns and SEM micrographs of the 296 two kinds of adsorbent precursor powders. The Mn-based powder showed nearly the 297 same diffraction peaks of the standard XRD pattern (PDF No. 46-0810), with main 298 peaks at $2\theta = 18.8^{\circ}$, 36.5° , 44.4° , and 64.5° . The spinel structure of the adsorbent 299 observed in the SEM analysis was similar with that described by previous research 300 reports [36]. Both XRD and SEM results suggest the successful synthesis of the Mn-301 based adsorbent. The XRD pattern of Al-based adsorbent powder was also consistent 302 303 with the diffraction peaks of the standard pattern (PDF No. 31-0700), with main peaks at $2\theta = 11.4^{\circ}$, 23.2° , 40.5° , and 63.2° . The SEM analysis of the Al-based adsorbent 304 suggested a lamellar structure, a layered crystal formed by superposition of octahedral 305 layers of oxygen coordination, indicating the one step coprecipitation method 306

307 successfully synthesized the Al-based adsorbent.

Analyzing the granules produced by combining the precursor powders with PVC, 308 309 SEM cross-sectional micrographs presented in Fig. 5b and 5f suggest that both adsorbent granules had a partly hollow and highly porous interior structure, with 310 311 micropores on the surface. The EDS mapping results indicate that Mn and Al were 312 finely distributed, suggesting that the adsorbent powders were uniformly incorporated within the PVC skeleton. Nitrogen adsorption-desorption isotherms of both granule 313 adsorbents, presented in Fig. 5c and 5g, indicate that the adsorbents had mesopores, 314 315 because of the IV isotherm with an H3 hysteresis loop. The amount of adsorbed N2 was 68.5 cm³/g and 112.6 cm³/g for the Mn and for the Al-based adsorbent, respectively. 316 The larger N₂ adsorption was attributed to the relative loose structure of layered Al-317 318 based crystal compared to the spinel structure of the Mn-based adsorbent, as observed with SEM analysis. The inset graphs in Fig. 5c and 5g display the pore size distribution 319 according to the BJH method. Mn-based adsorbent had bimodal mesoporous structure 320 321 distribution with a minor peak corresponding to a small portion of mesopores and a major peak related to the main mesopores. Its mean pore size ranged from 21 to 24 nm. 322 In contrast, the Al-based adsorbent displayed a unimodal mesoporous structure 323 distribution with an average pore size in the range 15-18 nm. Based on BET surface 324 area plot (Fig. 5d and 5h), the Mn-based adsorbent possessed a BET surface area of 325 $20.3 \text{ m}^2/\text{g}$ and a total volume of $0.106 \text{ cm}^3/\text{g}$, while the corresponding values for the Al-326 based adsorbent were 36.6 m²/g and 0.174 cm³/g, respectively. Overall, the structure, 327 surface area, and pore characteristics of the materials were highly suitable to achieve 328

- 329 potentially high Li adsorption capacity and kinetics, which are the topic discussed in
- the following chapters.





Fig. 5: Mn and Al-based adsorbent properties: (a, e) XRD patterns of the precursor

- powders; (b, f) representative SEM micrographs and EDS mapping of the granules
 obtained by combining the adsorbent powders with PVC; (c, g) N₂ adsorption desorption curves and resulting pore size distribution of the granules; (d, h) BET
 surface area plot.
- 337

338 3.3. Adsorption kinetics

339 The lithium adsorption behavior of both Mn and Al-based adsorbent granules was investigated through adsorption kinetics. The data were fitted to the pseudo-first and 340 second-order kinetic models and the results are displayed in Fig. 6. Both adsorbent 341 granules displayed a rapid lithium adsorption from the SLR solution within the first 1 342 h, followed by a substantial decrease in the adsorption rate until steady state was 343 reached after roughly 2 h. A relatively rapid adsorption rate observed in this work 344 compared with other previous research [42] may be partly attributed to the higher TDS 345 concentration of SLR (112.3 g/L). The osmosis pressure gradient generated by the 346 347 salinity difference between SLR solution and the adsorbent spheres may act as a driving force that promotes the transfer of lithium ion from the aqueous phase to solid phase. 348 Fitted kinetic parameters summarized in Table 2 suggest that the differences in R^2 349 values between pseudo-first-order kinetic model and pseudo-second-order kinetic 350 model were negligible. Therefore, the adsorption process was likely governed by a 351 combination of physical sorption and chemical sorption. According to previous studies, 352 the adsorption kinetics of Mn and Al-based adsorbents are mainly controlled by 353 chemical adsorption [31,36]. Some physical adsorption may possibly be attributed to 354



356 the inner pore structure of the adsorbent granules.

Fig. 6: Results of lithium adsorption kinetics test onto (a) Mn-based adsorbent and (b)
Al-based adsorbent. Data points were fitted using pseudo-first and pseudo-second
order kinetics, with fitting results represented by a red and a blue line, respectively.

361Table 2: Results of fitting the lithium adsorption kinetics data with pseudo-first-order

362		and pseudo	o-second-	order models.		
	Pseudo-first-model q _{e,cal} (mg/g)	k₁ (min⁻¹)	R ²	Pseudo-second-model q _{e,cal} (mg/g)	K₂ (g·min⁻¹h⁻¹)	R ²
Mn-based	4.25	3.66	0.9947	4.40	2.36	0.9867
Al-based	4.10	3.43	0.9955	4.31	1.81	0.9984

364 3.4. Performance of the integrated multiple stage adsorption process

An integrated three-stage adsorption process was designed and evaluated to maximize the recovery of lithium from SLR. Results presented in Fig. 7 imply that the adsorption on both adsorbent granules decreased in subsequent adsorption stages, likely due to the reduced concentration gradient. At the end of three-stage adsorption, the

concentration of Li⁺ ion decreased from 1147 mg/L to approximately 373 mg/L and 420 369 mg/L for Mn and Al-based adsorbent, respectively, and the recovery rate reached values 370 371 of 67.5% and 63.4%. The Mn-based adsorbent displayed a relatively superior recovery performance, attributed to specific recognition of Li⁺ ions by ion exchange of Li⁺/H⁺, 372 373 which is possibly a more suitable mechanism compared to the intercalation phenomena characterizing the main adsorption mechanism of the Al-based adsorbent. However, 374 overall, both adsorbents had a desirable performance for lithium recycling from 375 raffinate of spent lithium-ion battery. 376



377

Fig. 7. Multi-stage adsorption performance of Mn and Al-based adsorbent granules.

379

380 3.5. Selectivity performance

It is important to assess the selectivity of both Mn and Al-based adsorbents for lithium ion in the presence of interfering ions (Na⁺, K⁺, Ca²⁺, Mg²⁺) in the SLR, especially the Na⁺ ion characterized by high concentrations up to roughly 31200 mg/L. Fig. 8 presents the partition coefficients (K_d), while other adsorption and selectivity

parameters are summarized in Table 3 and Table 4. The lithium partition coefficients 385 associated with both adsorbents were larger than those of other ions, particularly Na⁺. 386 387 Specifically, the K_d value characterizing the Mn-based adsorbent (6.62 mL/g) was greater than that determined for the Al-based adsorbent (4.05 mL/g), consistent with 388 adsorption capacity results presented above. Compared with the layered crystal 389 structure of Al-based adsorbent, the relatively higher selective separation of Mn-based 390 adsorbent to lithium may be attributed to its spinel structure providing an ionic sieve 391 effect that promotes preferential adsorption of ions with radius similar to that of the 392 393 adsorption sites. It should be noted that the adsorption capacity for both adsorbents to Na⁺ was also large, which may be explained by the following reasons. On the one hand, 394 the concentration of Na⁺ ion in SLR was 27.2 times higher than the Li⁺ concentration, 395 396 which may promote much more substantial physical adsorption onto the surface and inner cavity of the PVC skeleton. This hypothesis was corroborated by the EDS 397 mapping determined on granule adsorbent after they were used for adsorption; see Fig. 398 S3. On the other hand, although the selectivity for Li⁺ with respect to Na⁺ was high for 399 both adsorbent materials, a proportion of adsorption sites were indeed consumed by 400 Na⁺, which is an unavoidable phenomenon because of the concentration ratios in the 401 SLR solution. That being said, the combination of capacity and selectivity results 402 suggests the suitability of both adsorbent for lithium uptake from SLR, with a slight 403 preference for Mn-based materials, possibly due to their more appropriate structural 404 405 characteristics.





407 Fig. 8. Partition coefficients (K_d) determined for both adsorbent granules for

coexisting cations.

410 Table 3: Adsorption selectivity of Mn-based adsorbent from SLR.

	C ₀ (mg/L)	C _e (mg/L)	q _e (mg/g)	K₁ (mL/g)	CF (mL/g)	\mathbf{Q}_{Me}^{Li}
Li⁺	1,147	750	4.96	6.62	4.33	1
Na⁺	31,200	29,400	22.2	0.75	0.71	8.79
K⁺	126	107	0.24	2.21	1.88	3.00
Ca ²⁺	54.6	49.9	0.06	1.16	1.06	4.09
Mg ²⁺	57.5	42.4	0.19	4.46	3.29	1.49

	C ₀ (mg/L)	C _e (mg/L)	q _e (mg/g)	<i>K</i> d (mL/g)	CF (mL/g)	$\pmb{\alpha}_{Me}^{Li}$
Li⁺	1,147	823	4.05	4.92	3.53	1
Na⁺	31,200	29,800	17.9	0.60	0.57	8.17
K+	126	114	0.15	1.32	1.20	3.71
Ca ²⁺	54.6	43.6	0.14	3.14	2.51	1.41
Mg ²⁺	57.5	42.6	0.18	4.38	3.24	1.12

416

Table 4: Adsorption selectivity of Al-based adsorbent from SLR.

418 3.6. Reusability and stability of Mn and Al-based adsorbents

The reusability of both adsorbent granules was evaluated in twenty cycles of 419 adsorption and desorption. The lithium adsorption/desorption capacity assessed in 420 every cycle is presented in Fig. 9. The adsorbents exhibited relatively stable 421 performance, with adsorption capacities fluctuating around 4 mg/g for Mn-based 422 423 adsorbent and 3.5 mg/g for Al-based adsorbent across the cycles. Concurrently, both adsorbents showed stable desorption behavior, with desorption capacities fluctuating 424 around 3.5 mg/g and 2.8 mg/g for Mn-based granules and Al-based granules, 425 426 respectively. For Mn-based granules, the seemingly inconsistent value of lithium desorption compared to that of previous adsorption in the first cycle may be explained 427 with the incomplete washing out of adsorbent precursor from the synthesized material, 428 429 resulting in some template Li⁺ still occupying the binding sites during the first cycle of adsorption, then washed out during the subsequent desorption step. 430

Fig. 9a presents results obtained in the investigation of the adsorbent stability. Mn
loss from Mn-based materials was 0.45% after the first cycle, and it decreased to around

0.2% for the following cycles. A higher Mn loss in the first cycle is likely ascribed to 433 the incompletely fixed Mn element in the adsorbent crystal structure during the 434 sintering procedure, with a portion of the element that was only attached to the surface 435 of the powdered adsorbent, thus easily eluted once in contact with the desorption 436 solution. The results presented in Fig. 9b suggest that during the 20 cycles, the Al-based 437 granule suffered instead negligible Al loss, namely, less than 0.06‰. A relatively larger 438 Mn loss during all cycles may be attributed to the effect of acid treatment, which would 439 likely induce electron transfer from trivalent manganese in the crystal structure to 440 tetravalent manganese in the surface. After obtaining sufficient electrons, tetravalent 441 manganese would convert to bivalent manganese, in turn subject to dissolution [43]. 442 Overall, the remarkable stability of both adsorbents imply the potential for long-term 443 444 application of both adsorbents for lithium selective adsorption recovery.



Fig. 9: Adsorption-desorption capacity of Li⁺ and related Mn, Al loss in different,
subsequent cycles of adsorption and desorption: (a) Mn-based adsorbent granules, (b)

449

Al-based adsorbent granules.

450

451 **4. Conclusion**

This paper discussed materials and their application in an integrated approach for 452 lithium recovery from spent lithium-ion battery raffinate (SLR), combining 453 pretreatment of the solution via PACl coagulation, biochar aerogel adsorption, and 454 455 ultrafiltration, with lithium adsorption onto Mn and Al-based adsorbent granules. The pretreatment steps effectively removed organic constituents of the SLR, which 456 decreased from roughly 761 mg/L to 119 mg/L without observable lithium-ion loss. 457 458 Both Mn and Al-based adsorbent granules exhibited rapid adsorption of lithium from the pretreated SLR, reaching saturation within 2 h, with final capacity in the range 4-5 459 mg of lithium per g of adsorbent granular material. Notably, the partition coefficients 460 of Li⁺ for Mn-based (6.62 mL/g) and Al-based (4.92 mL/g) adsorbents were 461 significantly higher than those of other ions, particularly Na⁺ (0.75 mL/g for Mn-based 462 adsorbent and 0.60 mL/g for Al-based adsorbent), suggesting effective separation of 463 Li⁺ and Na⁺. The Mn-based adsorbent showed superior adsorption capacity compared 464 to Al-based adsorbent, attributed to its favorable structure. Both adsorbents also 465 presented remarkable reusability. After twenty adsorption/desorption cycles, the 466 adsorption capacities of Mn and Al-based adsorbents reached a stable behavior, 467 maintaining values of 4.5 mg/g and 3.5 mg/g, respectively, which was only about 10-468

469	15% lower than the values measured for the as-synthesized materials in their first cycle
470	of application. While the Mn-based adsorbent provided a better adsorption capacity, it
471	experienced a relatively higher element loss, equal to approximately 0.2% Mn loss per
472	cycle. On the contrary, Al loss was negligible from Al-based materials. This study
473	indicates the effectiveness of pretreatment followed by Mn and Al-based adsorbent
474	adsorption as a promising and technically feasible strategy for lithium recovery from
475	SLR.
476	
477	Notes
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479	
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491 considered responsible for them.

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