

Microstructure and thermal conductivity of Si-Al-C-O fiber bonded ceramics joined to refractory metals

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different approaches [3]. Several authors have successfully used brazing as a method for joining fiber bonded SiC ceramics to themselves [4].

In this work, we explore the joining of Si-Al-C-O fiber bonded ceramic materials to Cu-Clad-Mo, using Ag-Ti-Cu based active fillers, as previously used to join refractory metals to carbon, carbides and borides [5,6]. Since most of the proposed applications are related to thermal management for these systems [7–9], we measure the thermal conductivity of both the bare ceramic as well as the joints. The resulting conductivity can be well explained by considering two material layers in series, confirming that the interfacial resistance of the brazing seam is very low.

2 Materials and methods

The fiber-bonded ceramic composite was obtained from Ube Industries (Japan) and is composed of pressed Si-Al-C-O fibers that adopt a polygonal cross section after hot-pressing as shown in Figure 1-a. These are highly crystalline fibers composed mostly of β -SiC grains with a small amount of unreacted C at their interfaces (Figure 1-b). The refractory metal was Copper-clad Molybdenum (Climax Specialty Metals, USA) in 1 mm thick sheets, with a 600 μm layer of Mo (>99.95%) sandwiched between two 200 μm layers of oxygen-free Cu. Brazing was performed using a single 50 μm thick sheet of Ag-Cu-Ti alloy with a liquidus temperature of 815 $^{\circ}\text{C}$ (Cusil ABA, Morgan Advanced Ceramics, USA). Table 1 shows relevant properties of the materials used in this work.

Table 1. Main room-temperature properties of Cu-clad Mo [10], Cusil-ABA [11] and fiber-bonded SiC ceramic (data from the manufacturer).

Material	Composition	Density ($\text{g}\cdot\text{cm}^{-3}$)	Young modulus (GPa)	Thermal expansion coefficient ($10^{-6}/\text{K}$)	Thermal conductivity ($\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-1}$)
Cu-Clad Mo	60 vol. % Mo Cu	9.55	283,5	11.6	200-250
Cusil-ABA	63 % Ag 35,25 % Cu 1,75 % Ti (wt. %)		130	18.5	180
Fiber-bonded ceramic	88 vol. % SiC C	3.10	300	3.20	71,5 (in-plane) 52,2 (through thickness)

Joining was performed by heating over the liquidus temperature of the brazing alloy under argon at a nominal pressure of 2 bar in a tube furnace. Heating rate was 15 °C/min up to 720 °C, held for 90 minutes to homogenize the temperature across the interface, and then at 5 °C/min up to 900 °C, maintained for a time t_B (10 or 20 minutes). Since fiber bonded SiC is transversely isotropic, the effect of fiber orientation was explored by producing joints in planes parallel or perpendicular to the ceramic fiber mats.

Thermal diffusivity was measured using the Laser-Flash technique (LFA1600 from Linseis, Germany) in vacuum. Samples were 10 x 10 mm². In the case of the ceramic, diffusivity was measured across two directions, one parallel and the other perpendicular to the pressing direction. In the case of the joint, the orientation was such that the ceramic's pressing direction is parallel to the braze. This orientation shows highest conductivity and is the most difficult to join, as the fiber ends are perpendicular to the brazing seam. Thermal conductivity $k(T)$ was calculated from thermal diffusivity $\alpha(T)$ using:

$$k(T) = \rho\alpha(T)C_p(T)$$

Where ρ is the density and $C_p(T)$ is the specific heat capacity. Heat capacity for SiC was interpolated following Snead *et al.* [12], while that of Cu-Clad-Mo was taken as 0.313 J g⁻¹K⁻¹ at 150 °C [10] and considered constant for the studied temperature range.

3 Results and discussion

Figure 1-c shows an example of one of the joints obtained in this work, for the case of $t_B = 20$ min. In the central region the brazing alloy can be observed (marked with a rectangle), with lighter contrast corresponding to Ag-rich areas. No pores or defects can be seen in the metal-alloy interface, as Ag enrichment of the Cu cladding helps improve its ductility thus minimizing stresses due to thermal mismatch, as shown in joints of Cu-clad-Mo to C-C fiber composites [13]. A closer inspection of the ceramic-metal interface (Figure 2, a-d) shows an eutectic microstructure with two phases, one Ag-rich (light contrast) and another Cu-rich (dark contrast). In all cases a continuous reaction layer free of pores and other defects was obtained, due to the good wettability of these alloys onto SiC ceramics [14].

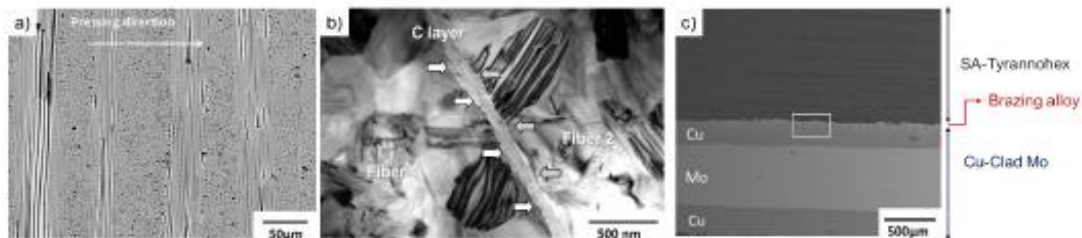


Figure 1. a) SEM micrograph of the fiber-bonded ceramic, b) transmission electron microscope image of the interface between two fibers, c) low magnification SEM micrograph of the ceramic joined to Cu-clad-Mo (the white rectangle depicts the schematic location of the micrographs in Figure 2).

To better determine the reaction products, EDS maps were obtained (Figure 2-e) at the ceramic-alloy interface. As is readily observed, Ti is concentrated near the ceramic surface, which can be attributed to the formation of TiC by reaction with the ceramic's outer carbon layer. These results are comparable to joints of fiber bonded ceramics to themselves using Cusil-ABA [4].

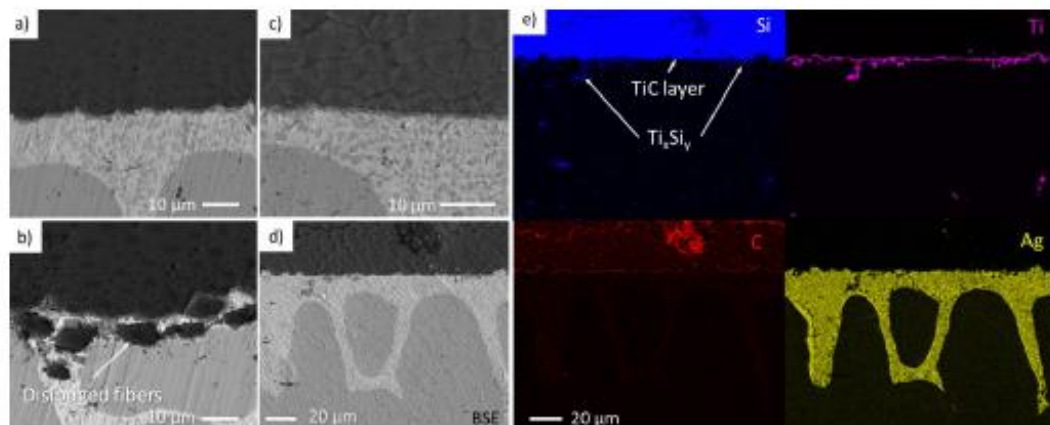


Figure 2. SEM micrographs of the ceramic-alloy interface from SiC/Cusil-ABA/Cu-clad-Mo obtained for: a) and b) $t_B=10$ min; c) and d) $t_B=20$ min. e) Si, Ti, C and Ag EDS compositional maps from the region in panel d).

The preferential segregation of Ti at the ceramic/braze interface is consistent with the significant chemical affinity of Ti towards C; the carbide formation at the interface is helpful in promoting wetting and bonding. The TiC layer is continuous and homogeneous, as shown in Figure 2-e; this

aspect is beneficial for the joint soundness, since the spreading and the wettability of the brazing alloy are uniform on the mating surfaces. Additionally, the preferential segregation of Ti near the composite surface and the subsequent TiC formation occurs regardless of the fiber orientation. Some titanium silicides also form underneath the TiC reaction layer, as can be seen from EDS maps.

Main results of the thermal conductivity of the fiber-bonded SiC material studied in this work are presented in Figure 3-a, including data from polycrystalline β -SiC [12]. SiC thermal conductivity decreases with temperature, as is the case in ceramics when conductivity is dominated by phonon scattering. Since grain boundaries also act as phonon scatterers, small grained SiC materials will exhibit lower thermal conductivities. In our case, grain size is in the range $d \sim 200$ -500 nm and thermal conductivity is in the range of CVD grown β -SiC polycrystals with grain sizes *ca.* 3 μm . The high thermal conductivity observed for its grain size can be attributed to the presence of highly aligned turbostratic carbon at the fibers' interfaces (Figure 1-b).

Conductivities of Cu-Clad-Mo, fiber-bonded ceramic and the ceramic/Cusil-ABA/Cu-Clad-Mo joints are depicted in Figure 3-b. When considering conductivity, the joint can be described as three layers in series (metal, bonding alloy, and ceramic,) with different thicknesses x_m , x_a and x_c respectively. Joint conductivity can be described then as a function of each material's conductivity as [15]:

$$k_{joint} = \frac{x_m + x_a + x_c}{\frac{x_m}{k_m} + \frac{x_a}{k_a} + \frac{x_c}{k_c}}$$

Using the previous equation, we have estimated the expected thermal conductivity of the joints and plotted the results alongside the measurements. The good agreement between the rule of mixtures' prediction and the experimental values when accounting for experimental uncertainty indicates that no significant interfacial thermal resistance is present in the joint.

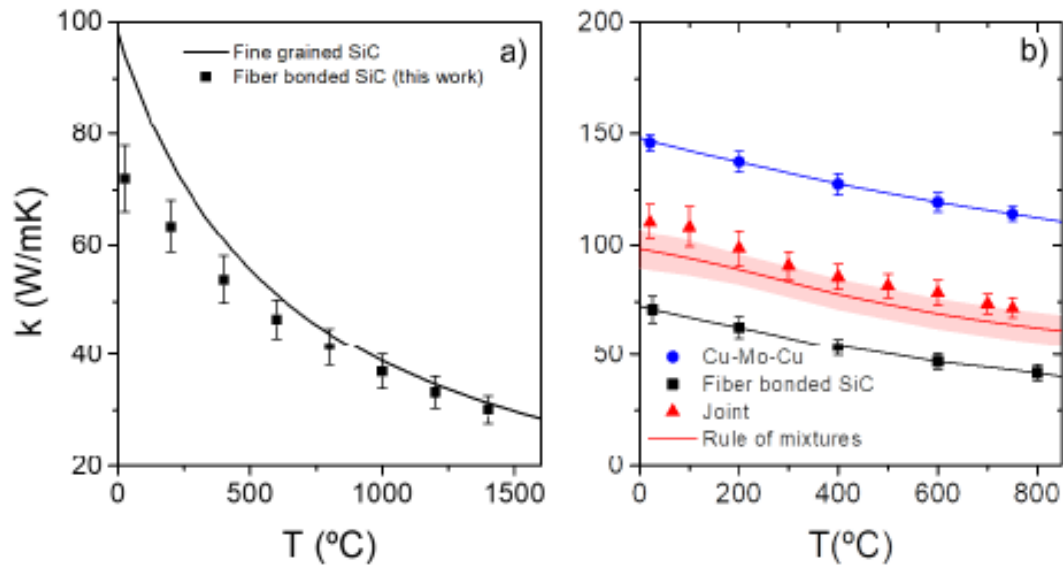


Figure 3. a) Measured in-plane thermal conductivity for fiber-bonded SiC, including data for β -SiC [12]. b) Thermal conductivity as a function of temperature for Cu-clad-Mo, fiber bonded SiC and the joint. The solid line represents the rule of mixtures, while the shaded band shows the related uncertainty.

4 Conclusions

We have successfully joined fiber-bonded SiC ceramics to Cu-clad-Mo using an Ag-Ti-Cu based brazing alloy. A brazing temperature of 900 °C for 10-20 min results in a sound joint regardless of orientation. The reaction layer is 1-2 μm thick and free of pores and defects. Ti migrates towards the ceramic side of the joint to form a TiC layer, while a eutectic microstructure of Ag and Cu forms the bulk of the joint.

The thermal conductivity across the joint was found to be consistent with the metal and ceramic being in series with no significant interfacial resistance. The joint is highly conductive and forms an almost perfect thermal interface between the two materials, confirming the quality of the obtained brazing layer.

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