

Effect of long-term mechanical cycling and laser surface treatment on piezoresistive properties of SEBS-CNTs composites

Original

Effect of long-term mechanical cycling and laser surface treatment on piezoresistive properties of SEBS-CNTs composites / Padovano, E., Bonelli, M.E., Veca, A., De Meo, E., Badini, C.. - In: REACTIVE & FUNCTIONAL POLYMERS. - ISSN 1381-5148. - 152:(2020), p. 104601. [[10.1016/j.reactfunctpolym.2020.104601](https://doi.org/10.1016/j.reactfunctpolym.2020.104601)]

Availability:

This version is available at: 11583/2829374 since: 2020-05-26T07:38:56Z

Publisher:

Elsevier B.V.

Published

DOI:[10.1016/j.reactfunctpolym.2020.104601](https://doi.org/10.1016/j.reactfunctpolym.2020.104601)

Terms of use:

This article is made available under terms and conditions as specified in the corresponding bibliographic description in the repository

Publisher copyright

(Article begins on next page)

Effect of long-term mechanical cycling and laser surface treatment on piezoresistive properties of SEBS-CNTs composites.

E. Padovano^{a*}, M.E. Bonelli^a, A. Veca^b, E. De Meo^{b,c} and C. Badini^a

^a Politecnico di Torino, Department of Applied Science and Technology, Corso Duca degli Abruzzi 24, 10129, Torino, Italy

^b C.R.F. S.C.p.A. – Group Materials Labs, C.so Settembrini 40, Torino, 10135, Italy

^c University of Turin, Department of Chemistry and NIS Centre, Via P. Giuria 7, Torino, 10125, Italy

* Corresponding author: elisa.padovano@polito.it; +390110904708; fax: +390110904699

Abstract

The piezoresistive behaviour of SEBS-CNTs nanocomposites was investigated to evaluate their potential applications as strain sensors. Composites containing from 3% wt. to 7% wt. of CNTs were processed by injection moulding in order to evaluate the percolation threshold. The piezoresistive response under flexural strain of nanocomposites with a CNTs content above the percolation threshold was then studied. The nanocomposites showing the most promising performance were tested under cyclic conditions. Conductive tracks were then processed on nanocomposites surfaces (with 3 and 4% of CNTs) by means of a laser treatment. Samples with optimized laser tracks were then submitted to 1000 stretching/releasing cycles, showing improved piezoresistive performance.

Keywords: Polymer-matrix composites (PMCs); Electrical properties; Piezoresistive behaviour; Surface treatment

1. Introduction

Polymer based nanocomposites are recognized to provide the basis for the development of new technologies because of their noticeable properties. They can find applications in several fields as structural components or as functional materials for optical devices, electric flexible components, electromagnetic shields, biomedical devices and strain sensors. In fact, it is well known that the addition of nanofillers such as carbon nanotubes (CNTs) or graphene nanoplatelets (GNPs) to polymeric matrices improves the mechanical behaviour (strength, stiffness, creep and toughness) as well as the electric and thermal conductivity. **These properties not only depend on the characteristics of the nanofiller and the polymeric matrix, but they are also greatly affected by the processing methods, the filler concentration, the homogeneity of filler dispersion and the orientation of fillers with high aspect ratio [1–6].** However, the extremely high conductivity of these nanofillers greatly enhances the conductivity of the insulating polymeric matrices when the filler concentration exceeds the percolation threshold, and a conductive network forms. The peculiar properties of these composites, whose electrical resistance changes with strain, can be exploited for sensors and actuators. When a conductive network is present inside the polymer matrix, an externally applied

40 stress can change the morphology of this network and cause a resistance increase or decrease
41 depending on the kind of strain. A linear and significant change of resistance with strain, as well as
42 the reproducibility of this phenomenon, is required for practical applications in sensors.
43 The effectiveness of CNTs for providing piezoresistive behaviour to polymers has been investigated
44 during the last twenty years and deeply reviewed in some papers [7–10]. From the literature it
45 seems that the influence of the polymer type on the piezoresistivity is small in comparison with the
46 effect exerted by the kind of filler (for instance single-wall or multi-wall CNTs), its concentration
47 and its distribution inside the matrix [7]. Piezoresistive effect is in fact due to the intrinsic
48 piezoresistivity of CNTs, the tunnelling resistance of neighbouring CNTs and the modification of
49 the conductive network formed by CNTs, due to the possible improvement or loss of contact among
50 CNTs [7,8]. Also, the filler aspect ratio is considered to exert a strong influence on both the
51 percolation threshold and the piezoresistivity [10–15].
52 The best piezoresistive response can be found when a proper filler concentration is adopted, while
53 an increase or a decrease of concentration with respect to the ideal one is frequently detrimental
54 [7,16,17]. In fact, when the filler concentration is very low the conductivity can be granted by
55 tunnelling effect only, and the resistance is high. On the contrary, when a densely packed network
56 forms because of rather high filler concentration, the modification of few conductive paths caused
57 by strain has only minor influence on the electrical resistance of the composite. Therefore, the
58 piezoresistive sensitivity of composites with high filler concentration can be poor. However, rather
59 different values of the percolation threshold are reported in the literature for polymer-CNTs
60 composites [7,9,18,19], even though percolation thresholds below 1% wt. were more frequently
61 observed. Generally speaking, CNTs concentration just above the percolation threshold seems more
62 suitable for achieving good strain sensitivity, since in this conditions a conductive network exists
63 and the conductivity is not negligible [9,17,20]. On the other hand, good piezoresistive behaviour
64 has been also observed for composites with rather high content of CNTs [10,16,18]. It was also
65 found that the piezoresistive sensitivity increases with the concentration, while the response
66 repeatability becomes poor and hysteresis appears far above the percolation threshold [21].
67 Most of the papers dealt with the piezoresistivity under tensile strain, but the behaviour under
68 compressive or flexural strain was also investigated [7–9,22]. The monotonic change of resistance
69 with the strain increase was observed for many composites, but only less frequently the resistance
70 modification under cyclic strain variation was investigated [9], and only rarely the materials were
71 submitted to a high number of cycles (namely 100 or 1000 cycles) [16,23]. The sensitivity of
72 piezoresistivity to the extent of the strain or pressure and the kind of applied stress (tensile,
73 compressive, flexural) and the possible phenomenon of saturation were also investigated [7,19,21].

74 Thin composite films [12,17–19,24–27], as well as thermoplastic and elastomeric matrices [10,17–
75 19,21,27–30], seem a good choice for the production of flexible and wearable devices. On the other
76 hand, bulk piezoresistive composites were processed by conventional techniques like melt mixing
77 and additive manufacturing by fused deposition modelling [14,18,31]. Chemical [24], thermal [30]
78 and laser [32] post processing treatments were also investigated with the aim of improving the
79 piezoresistive properties of the polymer-CNTs composites.

80 In the present paper the piezoresistive behaviour of nanocomposites with styrene-b-(ethylene-co-
81 butylene)-b-styrene (SEBS) matrix and multiwall carbon nanotube was investigated. The main
82 purpose is to investigate the influence on the composites piezoresistive response when different
83 strains are applied (by varying the maximum displacement and the rate of displacement); moreover,
84 the reproducibility of the piezoresistive properties over a high number of loading/unloading cycles
85 was assessed. The piezoresistive behaviour of the material was also checked after performing a
86 surface laser treatment.

87 **2. Experimental**

88 **2.1 Materials**

89 The piezoresistive behaviour of nanocomposites with a matrix of styrene-b-(ethylene-co-butylene)-
90 b-styrene (SEBS) thermoplastic elastomer reinforced by carbon nanotubes (CNTs) was
91 investigated. The nanocomposites were produced by diluting a masterbatch containing 7% wt. of
92 CNTs with the unfilled matrix. Pellets of a masterbatch were produced by Kraiburg TPE
93 (Waldkraiburg, Germany) using commercial SEBS (TC7LEZ-920) and NC7000 carbon nanotubes
94 (Nanocyl S.A., Sombreville, Belgium) with an average length of 1.5 μm and a diameter of 9.5 nm,
95 according to the producer datasheet. In order to produce the nanocomposites with different filler
96 concentrations (3, 4, 5, 6 and 7% wt. of CNTs), pellets of the masterbatch were mixed with pellets
97 of unfilled SEBS and then extruded using a twin screen extruder (Haake Eurolab).

98 **2.2 Characterization**

99 Two kinds of samples were processed by injection moulding (Babyplast 6/10P): bars with size
100 $80 \times 10 \times 4 \text{ mm}^3$ were produced for both resistance measurements and electro-mechanical tests, while
101 plates $110 \times 60 \times 2 \text{ mm}^3$ in size were used for optimizing the laser writing process.

102 **2.2.1 Electrical resistance**

103 Two-wires surface resistance measurements were performed by using a multimeter (Keithley
104 2700E, resistance up to 120 M Ω). The surface resistance was measured on both moulded sample
105 bars and conductive tracks obtained by laser scribing (sets of 3 samples). Square electrodes 10×10
106 mm^2 were created 40 mm apart one each other by depositing a silver-based conductive paints on the
107 surface of the as-processed bars; conductive wires were then embedded into the electrodes and

108 connected to the multimeter. In order to measure the resistance of the laser tracks, they were firstly
109 cleaned by using an air jet with the aim of remove the carbonaceous particles which were not well
110 adherent to the tracks; then dot-shaped electrodes were obtained by depositing the silver-based paint
111 at the edges of the tracks.

112 **2.2.2 Electromechanical tests**

113 Electromechanical tests, consisting of three-point bending coupled with resistance measurements,
114 were performed. The experimental set up is shown in Figure 1; **it was specifically designed for**
115 **measuring the strain and the resistance variation at the same time**. The three-point bending was
116 carried out by using a dynamometer (Instron 5544 with a load cell of 2kN) equipped with a
117 potentiometer for the measurement of sample displacement; Bluehill software **was used** for the
118 acquisition of load, stress, displacement and strain. The resistance was measured by a multimeter
119 (Keithley 2700E). The Instron data acquisition system and the multimeter were interfaced with a
120 computer exploiting Labview software, which collected all the experimental data.

121 **Formerly, the variation of resistance during a loading/unloading cycle was recorded under different**
122 **experimental conditions. Eighteen tests were carried out on composites: two samples containing 5%**
123 **wt. and 6% wt. of CNTs respectively were tested by varying the displacement and the holding time**
124 **at the maximum displacement (9 different experimental conditions were tested for each**
125 **composition) in order to investigate how their piezoresistive response can be influenced by different**
126 **ways of applying the strain.**

127 **Long-term electromechanical tests were also performed by repeating the loading/unloading cycles**
128 **up to 1000 times on the same sample. The piezoresistive behaviour during cycling of composite**
129 **samples was then compared with that displayed by the composites showing a conductive track**
130 **(processed by laser functionalization as described in 2.2.3 section). The cycling test was repeated**
131 **two times for each kind of investigated material.**

133 **2.2.3 Laser functionalization**

134 The laser functionalization was performed under nitrogen atmosphere by using a pulsed CO₂ laser
135 with wavelength of 10.6 μm and a maximum power of 100W. The processing parameters such as
136 power delivered, writing speed, defocus, laser frequency and number of writing repetitions on each
137 track were controlled by Flycad software. **For each set of laser treatment conditions four parallel**
138 **tracks with a length of 4 cm and 1 cm apart from each other were written on a composite plate with**
139 **size 110x60x2 mm³; their resistance was then measured. Also, the inter-track resistance and the**
140 **aesthetic characteristic of the samples after functionalization were considered in order to select the**
141 **best processing parameters.** The microstructure of the composites was investigated by examining

142 their fracture surface (obtained after a storage period in liquid nitrogen causing the sample
143 embrittlement) by means of a FE-SEM (Zeiss Merlin). The morphology of the conductive tracks
144 was investigated by using a confocal profilometer (Leica DCM81).

145 **Figure 1**

146 **3. Experimental results and discussion**

147 **3.1 Preliminary assessment of resistance and piezoresistivity behaviour**

148 The microstructure of the nanocomposites with different amount of CNTs can be observed on their
149 cryofracture surfaces (Figure 2). The carbon nanotubes are homogeneously dispersed inside the
150 polymeric matrix and tend to align in the direction of the injection moulding, thus placing
151 perpendicular to the fracture surface. **No evidences of carbon nanotube agglomeration were found
152 on the fracture surface of the masterbatch or on the fracture surface of the samples obtained by
153 diluting it with unfilled SEBS.**

154 **Figure 2**

155 The surface resistance of the composites with different amount of CNTs as reinforcement is
156 compared in Table 1; the nanocomposites are labelled with the acronym of the matrix (SEBS)
157 followed by the percentage of nanotubes. **The resistance decreases as function of the increase of
158 CNTs content, but not in a linear manner: this is due both to the formation of a conductive network
159 and to the so called “skin effect”. In fact, the percolation curves are “S-shaped” and the resistance
160 suddenly decreases when the nominal content of CNTs increases just over the percolation threshold,
161 because a continuous network of conductive filler forms. Moreover, it is well known [33–35] that
162 the injection moulding causes the alignment of the carbon nanotubes along the injection direction.
163 This alignment in the flow direction mainly occurs near the mould surface, where it leads to the
164 reduction of tube-tube contacts and an impairment of network formation. In addition, it has been
165 reported that during the moulding process the carbon fillers tend to migrate from the surface toward
166 the sample core [36]. Therefore, the electrical resistance of the skin is higher than that of the sample
167 core and the resistance measured on the surface is influenced by this skin effect. The percolation
168 threshold for the surface electrical conductivity of the SEBS-CNTs composite under investigation
169 can be placed at around 3-4% wt. of CNTs. **In the composite with a CNTs content corresponding to
170 the percolation threshold (3% of CNTs) the standard deviation (it was calculated for resistance
171 values measured between two points on the sample surface) was very high. Owing to the rather low
172 CNTs content the conductive network is inhomogeneous, and this results in different resistivity
173 values measured in different directions and parts of the sample.****

174 When a load is applied to a polymer/CNTs nanocomposite bar, an elastic strain formerly occurs. At
175 the microscopic level, the polymeric chains are stretched and then they are forced to assume a linear
176 shape. The CNTs that are placed in between the polymer chains are dragged from the change of

177 orientation and elongation of the chains, so they tend to align to the strengthening direction. For this
178 reason, the number of entanglements and of points of contact between the CNTs decreases when
179 elastic strain occurs; then the electrical resistance increases. When the elastic strain is removed, the
180 chains tend to assume the pristine configuration and then the conductive network of CNTs is
181 restored, with a consequent resistance decrease. This change of electrical resistance during loading
182 and unloading is responsible for the piezoresistive behaviour of CNTs-based nanocomposites. This
183 phenomenon can be, in principle, exploited for the fabrication of pressure sensors. However,
184 several other requirements are needed for this application: the resistance variation should occur
185 under small strain levels and contemporary to the deformation; in addition, the resistance variation
186 should happen cyclically every time that a load is applied or relieved.

187 With the aim of assessing the suitability of the elastomeric nanocomposites under investigation for
188 sensor applications, the effect of the mechanical strain on the piezoresistive behaviour was
189 investigated under different loading conditions. Firstly, the piezoresistivity effect was studied
190 during a single cycle of deformation and then the behaviour of nanocomposites during mechanical
191 cycling was evaluated.

192 The first test involved the progressive bending (up to 10 mm of displacement) of the composite bars
193 at the constant displacement rate of 10mm/min and the measurement of resulting resistance
194 variation. The latter, reported as percentage of change with respect to the initial resistance value, is
195 plotted in Figure 3 as a function of the time and the displacement.

196 **Figure 3**

197 For all the specimens the resistance increased **when** the displacement increased, but the curves
198 showed different trends and slopes depending on the filler concentration. Samples with 3% or 4% of
199 CNTs show high resistance values that change in an irregular manner with the displacement, and
200 then they display bad piezoresistive behaviour. Samples with 7% CNTs shows resistance values that
201 increase according to a not linear trend as function of the displacement increase. Actually, the
202 resistance increases less than **that** expected under high displacements. **When the CNTs**
203 **concentration is high, not all the nanotubes change their orientation when the displacement**
204 **increases; the nanotubes are in fact strongly interconnected within a network with several contact**
205 **points that are hardly unleashed. Also according to Villmow et al. [35] the re-arrangement of CNTs**
206 **caused by external stresses becomes more difficult with their concentration increases.** Sample with
207 6% and overall sample containing 5% of nanotubes show an almost linear increase of resistance
208 with the displacement; in particular SEBS-5 sample shows the maximum resistance percent
209 variation (1.2%).

210 Conclusively, contents of CNTs just above the percolation threshold seem to grant the best
211 piezoresistive response. SEBS-5 and SEBS-6 nanocomposites were therefore submitted to further
212 tests which simulate the possible operating conditions experienced by a pressure sensor. **Actually, a**
213 **piezoresistive switch could be strained in different manners in operating conditions. Therefore, the**
214 samples were tested according to a cycle of loading and unloading, keeping constant the
215 displacement rate (10mm/min), but changing the maximum displacement and the period of holding
216 at the maximum displacement. Maximum displacements of 2, 5 and 10 mm and times of staying at
217 the maximum displacement for 2, 5 and 10 seconds were adopted, **thus performing nine tests for**
218 **each sample composition, as shown in Table 2.**

219 Figure 4 shows the displacement change **occurring during one of these cycles** (blue curve), the
220 resistance variation expected for a material showing an ideal piezoresistivity behaviour (red dotted
221 curve), and an example of real variation of electrical resistance (red curve).

222

Figure 4

223 Elastomers should display an elastic behaviour, and then they should recover the original size and
224 shape when the load is removed. The movement of polymeric chains should also cause the recovery
225 of the original configuration of the CNTs network, and the restoration of the pristine electrical
226 resistance as well. However, relaxation phenomena can occur with some delay with respect to the
227 load change. As a result, the curve depicting the real change of resistance with the displacement
228 differs from the ideal one, as shown by the comparison between the two red curves in Figure 4.

229 Although an ideal piezoresistive material should react to a displacement variation by changing
230 immediately its resistance, some delay in the response and a signal instability were observed for the
231 materials under investigation.

232 Moreover, the speed of resistance variation was found to be different from the speed of
233 displacement variation in every part of the cycle; some resistance variation could be also detected
234 when the displacement remained constant (maximum displacement and null displacement at the end
235 of the cycle). The rate of resistance variation can be compared with the rate of displacement
236 variation during loading and unloading in order to assess how much the piezoelectric behaviour
237 deviates from the ideal one. Relaxation effects under a constant load should be also considered to
238 this purpose.

239 The following parameters were then recorded to investigate the material behaviour under a single
240 cycle: the speed of resistance increase during the displacement increase (v_1), the speed of resistance
241 variation during the stay at the maximum displacement (v_2) and the speed of resistance decrease
242 during the displacement decrease (v_3). The best piezoresistive behaviour can be observed when v_1
243 and v_3 are as close as possible to the displacement variation rate and when v_2 is close to zero.

244 The effectiveness of a piezoresistive device is also related to the extent of resistance variation when
245 displacement changes, **while resistance variation should not** occur when the displacement is
246 constant or null. In order to investigate the deviation from the ideal behaviour some parameters
247 related to the resistance changes were also considered (Figure 4): maximum resistance variation
248 during the loading step (ΔR_1), resistance decay during the stay at the maximum displacement (ΔR_2)
249 and resistance deviation with respect to the initial one at the end of each cycle (R_b). Finally, an
250 ideal piezoresistive behaviour should entail that the initial resistance is **fully** recovered as soon as
251 the displacement is completely removed; therefore, the possible delay time (t) and the deviation
252 from the initial resistance value at the end of the cycle (R_b) should be as little as possible. The
253 values assumed by these parameters were calculated from each experimental curve. All the results
254 are reported in Table 2.

255 A further parameter that could be considered is the v_3/v_2 ratio, that should be as high as possible,
256 because it is related to the promptness to react to the start of unloading with a clear signal.

257 As an example, some experimental curves showing the resistance variation during cycles carried out
258 under different conditions are reported in Figure 5.

259 It is evident that the resistance variation profile is affected by the cycling conditions. The results
260 summarized in Table 2 show as the maximum displacement and the period of maintenance at the
261 maximum displacement exert a great effect on the effectiveness of the piezoresistive material, **as**
262 **discussed in the following sections**. Moreover, some differences can be seen for composites with
263 different CNTs content.

264 **3.1.1 Composite with 5%wt. of CNTs**

265 An increase of the maximum displacement or of the holding time at the maximum load causes the
266 increase of the resistance change (ΔR_1) and of the promptness of piezoresistivity response (v_1); the
267 increase of maximum displacement also increases the resistance change speed when the load is
268 progressively removed (v_3).

269 **Figure 5**

270 On the contrary, these parameters seem not to affect in a clear manner the time (t) required to
271 recover the initial resistance value when the load is completely removed. As a matter of fact, not
272 only there is always a delay in the recovery of the initial resistance after unloading, but the material
273 continues to change its resistance in the minutes after unloading, giving rise to final resistance
274 values **that are different from the** initial one (R_b). Actually, not negligible R_b values were found; it
275 means that during a first cycle of loading and unloading some adjustment in the network of CNTs
276 always occurred. During the stay at the maximum displacement the resistance slightly decreases and
277 the speed of this variation (v_2) increases as function of the maximum displacement suffered by the

278 sample and of the holding time. Finally, the v_3/v_2 ratio clearly increases with the maximum
279 displacement.

280 An increase of the holding time at the maximum displacement (and load) also causes the increase of
281 both resistance variation and resistance speed of changing (ΔR_1 and v_1) during loading. However,
282 this parameter does not affect in a clear manner the v_3/v_2 ratio, as it influences both v_3 and v_2 in the
283 same manner. The recovery of the initial value of resistance requires increasing time (t) with the
284 prolongation of the holding time at the maximum load. No clear correlations were found between
285 the holding time and R_b .

286 Conclusively, the best piezoresistivity behaviour during the one-cycle test was observed when high
287 values of maximum displacement and holding time were adopted.

288 **3.1.2 Composite with 6%wt. of CNTs**

289 In the case of composite containing 6%wt. of CNTs an increase of the maximum displacement
290 causes the increase of both the resistance change and the resistance change speed during loading
291 (ΔR_1 and v_1); a similar effect, but much less marked, was observed for v_3 during unloading.

292 Unfortunately, also the resistance variation occurring meanwhile holding the load (ΔR_2 , always
293 very little) and its change speed (v_2) increase with displacement, which results in the decrease of the
294 v_3/v_2 ratio. The time required to recover the initial resistance (t) increases with the displacement and
295 the final resistance always differs from the initial one (even though a correlation between R_b and
296 displacement was not found).

297 An increase of the holding time at the maximum displacement does not result in any improvement
298 of piezoresistive response during the loading step (ΔR_1 and v_1), while a little beneficial effect was
299 observed on the resistance variation rate during the displacement release (v_3). This parameter seems
300 to have only a negligible effect on the resistance variation and resistance variation speed during
301 holding (ΔR_2 and v_2); moreover, no correlation was found between this parameter and R_b . Anyway,
302 ΔR_2 was always found very little. The prolongation of the holding time has also a detrimental effect
303 on the index v_3/v_2 . As already occurred for the composite containing 5% of CNTs, not negligible R_b
304 values were found.

305 Conclusively, the piezoresistive behaviour of the nanocomposite containing 6% of CNTs is only
306 slightly affected by the testing parameters. This is not surprising because the piezoresistive response
307 is usually affected by the CNTs content, **and high CNTs concentrations make the system less**
308 **sensitive to strain.**

309 When the CNTs concentration increases, the resistance decreases, as more percolation paths are
310 present inside the conductive network. When this happens the change of conductivity caused by the
311 application of a load becomes less important as the number of conductive paths is high under both

312 loading and unloading conditions. For this reason, the conditions of load application (defined as
313 maximum strain achieved, period during which the load is hold and then time available for network
314 re-arrangement) affect less the piezoresistive response of the composite with 6% of CNTs. Anyway,
315 in this case good piezoresistive response was also observed when the loading/unloading cycle was
316 carried out with rather small displacement and short holding time.

317 **3.2 Piezoresistivity observed under cycling conditions**

318 A pressure sensor, based on piezoresistive effect, should operate for long periods and also properly
319 react to stress and displacement variations several times. In order to investigate the suitability of the
320 nanocomposites under investigation for exploitation in pressure sensor devices they were submitted
321 to high number of cycles. The samples with 5% wt. and 6% wt. of CNTs were submitted to 100
322 cycles of strain variations in the following conditions: displacement increase/decrease rate of 10
323 mm/min, maximum displacement of 10 mm (for SEBS 5% CNTs) or 2 mm (for SEBS 6% CNTs), no
324 holding period at the maximum displacement. On the base of the results previously obtained, a high
325 displacement value (which enhances the resistance change) was adopted for testing SEBS with 5%
326 wt. of CNTs, while a lower displacement was sufficient to obtain suitable resistance variation when
327 testing SEBS with 6% wt. of CNTs. The electrical signal given by the sample with 5% of CNT
328 showed instability since the first cycling period; this is clearly depicted for the first ten cycles in
329 Figure 6.

330 **Figure 6**

331 In the case of nanocomposite with 5% CNTs it is also clear that stress and strain are out of phase
332 and a noisy electrical signal is obtained; some permanent deformation of the sample was also
333 observed.

334 In the case of sample with 6% of CNTs a certain lack of coherence between electrical signal and
335 displacement was observed, even though the specimens provided a signal during all the testing
336 period (100 cycles). The resistance progressively decreased with the number of cycles increase, but
337 the resistance variation during each cycle was kept almost constant (Figure 6B). The lack of
338 coherence between the imposed displacement and the resistance variation can be explained on the
339 base of the microstructure of the co-polymer SEBS. In these polymeric materials small blocks of
340 styrene are alternated with longer ethylene-butylene blocks. These latter show lower elastic
341 modulus and then greater elastic deformation, but the total deformation is hindered by the stiff
342 styrene blocks. The soft blocks suffer more deformation during each cycle, here the chains stretch
343 and tend to align and the strength of the bonds between the chains increases. On the other hand, the
344 stiff styrene portion of the polymer hinders elastic deformation. The deformation is slowly, and then
345 not completely recovered during the second part of the cycle, which results in some hysteresis.

346 During cycling the sample stiffness progressively increased, as shown by the fact that the load
347 required to achieve the pre-fixed displacement progressively increased too. The CNTs were dragged
348 by the movements of the polymeric chains, changed their relative position and the conductive
349 network characteristics, thus causing a change of the overall resistance of the material.
350 Conclusively the piezoresistive behaviour of these nanocomposites showed some lacks in terms of
351 readiness to react to stress and capability of maintaining the electrical properties during long
352 duration cycling. These drawbacks very likely would hinder the exploitation of these materials in
353 pressure sensors. However, it is well known that the piezoresistivity of nanocomposites can be
354 improved by using laser treatments.

355 **3.3 Functionalization by laser treatment: writing of conductive tracks**

356 The effect of the irradiation by a laser beam of the surface of a polymer/CNTs composite has been
357 investigated in previous papers [32,37]. The laser beam causes the thermal decomposition of the
358 polymeric **matrix, and then the increase of CNTs concentration and material conductivity**. This kind
359 of functionalization can be exploited for the creation “in situ” of conductive tracks and **it constitutes**
360 an attractive method for the realization of metal-free electrical circuits. The effectiveness of this
361 surface laser treatment for improving the piezoresistive behaviour was also proved in previous
362 investigations, for instance for PC-ABS/CNTs composites [32]. For this reason, in the present
363 investigation this kind of treatment was also used to modify the piezoresistive response of the
364 SEBS/CNT composites. The result of the laser treatment can vary depending on the parameters
365 adopted for the laser writing. In fact, the laser power, the writing speed, the number of repetitions,
366 the frequency and the focusing can greatly affect the final electrical resistance of the conductive
367 tracks. Generally speaking, the amount of delivered energy, and then the severity of the thermal
368 decomposition process, increases with the increase of the power and number of repetitions and
369 decreases with the increase of the laser scan speed. Unfortunately, a too much severe laser treatment
370 can result in a damage of the sample and then should be avoided for practical applications. The
371 experimental conditions that allow to obtain the best conductivity improvement without any sample
372 damage depend on the kind of nanocomposite and, in particular, on the kind of polymeric matrix.
373 Therefore, the best experimental conditions for such a kind of functionalization should be assessed
374 for each polymer/CNTs system.

375 Formerly, the laser writing parameters were optimized for the SEBS nanocomposites, and then the
376 piezoelectric behaviour was tested on sample laser treated under these selected conditions. Since the
377 laser treatment causes a conductivity increase, the composites containing 3% or 4% of CNTs were
378 submitted to laser writing. Moreover, as showed in section 3.1, composites with rather high
379 nanotubes content and low resistance do not show the best piezoresistive behaviour. On the

380 contrary, the composites with 3-4% of CNTs show a conductivity just below the percolation
381 threshold, and their conductivity is expected to appreciably increase owing to the laser treatment.
382 In order to assess the most suitable parameters for the laser writing process, several laser treatments
383 were carried out and then the electrical properties of the resulting tracks were compared. The track
384 resistance as well as the inter-track resistance were measured. The inter-track resistance must be
385 very high in order to avoid short circuits. The morphology of the samples was also checked after the
386 treatment; in fact, any change of shape or hole formation should be avoided.
387 On the base of previous experience in laser functionalization, the frequency was fixed at 30 kHz
388 and the defocusing at 0 for all tests. Laser speeds of 100, 200 and 300 mm/s were coupled with
389 powers values equal to 5, 10 and 20% of the total power available. These conditions were combined
390 with 10, 20 and 30 laser runs for the composite with 3% of CNTs and with 5, 10 and 15 repetitions
391 for the composite with 4% of CNTs. Fifty-four laser treated samples were obtained and for each of
392 them the track resistance (measured on the four tracks), the inter-track resistance and the final
393 integrity were investigated. The set of laser writing parameters are summarized in Table 3 with the
394 relevant values of track and inter-track resistances (R_t and R_i respectively). The standard deviation
395 calculated for each set of four tracks obtained under the same conditions shows that very poor
396 reproducibility of resistance values can be achieved when low power and high writing speed are
397 adopted. In these cases, the energy delivered by the laser is low and the treatment affects only the
398 sample surface, where the network of CNTs is not very effective because of the previously
399 mentioned skin effect. In several cases very high resistance, sometimes exceeding the measurement
400 range of the multimeter (reported as OR), was measured between adjacent traces, while in other
401 cases the inter-track conductivity was not negligible and therefore the set of adopted laser
402 parameters inappropriate. Several experimental conditions caused visible damage of the specimens:
403 trials 5-9, 14-18, 23-27 for composites with 3% CNTs and trials 1, 5, 6, 8, 9, 15, 18, 21, 24 for
404 composites with 4% CNTs. Actually, too high energy delivered by the laser, due to high power and
405 number of repetitions combined with low laser speed, has detrimental effect. Samples with poor
406 track conductivity, too low inter-track resistance or showing some damage were discarded.
407 Only few of these sets of laser writing conditions seem suitable for processing properly conductive
408 tracks on SEBS composites. The performances of the tracks showing acceptable characteristics are
409 compared in Figure 7A for the samples with 3% of CNTs and in Figure 7B for the samples with 4%
410 of CNTs. Also, the ratio between inter-track and track resistance is here reported.

411 **Figure 7**

412 Finally, the conditions used for trial T2 ($P=5\%$, $v=200$ mm/s, $N=30$ repetitions) and trial T12 (P
413 $=5\%$, $v=100$ mm/s, $N=10$ repetitions) were selected respectively for the nanocomposite with 3%

414 and 4% of CNTs, and adopted for the further investigation on their piezoresistive behaviour. Using
415 these processing parameters, a single track was obtained on SEBS-3 and SEBS- 4 samples bars,
416 which were then submitted to cyclic electro-mechanical testing.

417 **3.4 Piezoresistive behaviour of nanocomposites after laser treatment**

418 The morphology of the laser tracks can be appreciated by using the confocal microscope (Figure 8).
419 The comparison of Figures 8B and 8D highlights the major importance of the writing speed with
420 respect to the number of repetitions. In fact, the two tracks show similar width but very different
421 depth (d_z parameter in the track profiles). The track depth is around double in the sample with 4%
422 of CNTs treated with a writing speed of 100 mm/s (half of the speed used for the sample with 3% of
423 CNTs), even though only ten laser runs were performed.

424 **Figure 8**

425 These samples were submitted to bend strain cyclically (1000 cycles using a displacement rate of 10
426 mm/min and a maximum displacement of 2 mm), meanwhile their surface electrical resistance was
427 measured. A displacement rate of 10 mm/min and a maximum displacement of 2 mm were adopted
428 for testing, since these conditions proved to be suitable for composites with rather high CNTs
429 concentration (see section 3.2) and the laser treatment causes the local enhancement of filler
430 content. The resistance variation during cyclic tests versus the displacement is shown in Figure 9.

431 **Figure 9**

432 In both cases it can be seen that there is some instability during the initial cycles; in fact, the
433 average resistance increases and the extent of resistance range changes. However, after a first period
434 the system stabilizes. The enlargement of a portion of these curves allows to appreciate how the
435 resistance changes in a coherent and prompt manner with the displacement (Figures 9B and 9D). In
436 fact, the maximum and the minimum resistance values were always observed at the maximum and
437 minimum displacement for both the nanocomposites. The resistance of the material before testing
438 was never recovered at the end of each cycle because of the visco-elastic behaviour of these
439 nanocomposites, which causes a delay in the restoration of the microstructure when the mechanical
440 load is removed. Moreover, some differences in the behaviour of these two materials can be
441 appreciated. The range of resistance variation was much wider, and the curve of resistance change
442 was sharper for the composite containing 3% of CNTs, which gave the best result. Anyway, the
443 laser functionalization was able to greatly improve the piezoresistive response of these
444 nanocomposites. In fact, after the laser treatment the readiness of reacting to a mechanical stress
445 and the reproducibility of the piezoresistive response during long term mechanical cycling
446 appreciably increased. Conclusively, the laser treatment was able to generate tracks with a peculiar
447 microstructure that behave in a different manner from composites with a high content of filler, well
448 dispersed inside the matrix. In fact, the increase of the CNTs concentration in the SEBS matrix gave

449 no advantage for the piezoresistive response, while the **laser action greatly improved** it. Probably
450 the laser treatment not only locally increases the filler concentration, but also gives rise to a very
451 effective network constituted by CNTs and carbonaceous particles coming from the matrix
452 pyrolysis.

453 However, some instability of the average resistance was still observed after the laser treatment,
454 particularly during the initial cycling period. On the other hand, electronics can well offer the way
455 of correcting this lack, so that the peculiar characteristics of this kind of nanocomposites can be
456 exploited for fabrication of pressure sensors.

457 **4. Conclusions**

458 In the present study nanocomposites with SEBS matrix filled with CNTs were **processed starting**
459 **from a masterbatch by compounding with unfilled SEBS and injection moulding, thus** obtaining a
460 fairly good distribution of the nanotubes within the matrix. The threshold for the formation of a
461 conductive network was observed at around 3% wt. of CNTs.

462 The composites with 5% wt. and 6% wt. of CNTs showed the best piezoresistive response. A higher
463 CNTs concentration was detrimental for the piezoresistive behaviour because the conductive
464 network is closely packed and its modification due to strain gives rise to small resistance variations
465 only. The electrical resistance of these nanocomposites increased in an almost linear manner with
466 the displacement increase because of the stretching of the macromolecular chain which caused the
467 modification of the CNTs network. When the strain was released, the resistance decreased because
468 of the restoration of the conductive network, which occurred with some delay with respect to the
469 strain change. During the initial stretching/releasing cycles some adjustment of the conductive
470 network and of the average material resistance happened. The experimental results **showed** that the
471 resistance variation occurring during a single cycle **depended** on the strain variation profile. In fact,
472 the maximum displacement, the displacement rate and the holding time of maximum displacement
473 affected the piezoresistive response. During the maintenance of the maximum displacement some
474 relaxation always occurred, resulting in a slight resistance decrease. The best piezoresistive
475 response was observed for the composite with 5% of CNTs under high displacement and strain
476 holding time. The effect of strain conditions **became** less important when the concentration of CNTs
477 **increased** (e.g. 6% wt. of CNTs).

478 The repetition of the stretching/releasing cycle (100 cycles) put in evidence some lacks in the
479 piezoresistive response of these nanocomposites. Initially the average resistance decreased, but the
480 range of resistance variation only slightly changed during 100 cycles. The main lacks dealt with the
481 readiness of reacting to strain variation and the mismatch between displacement and resistance

482 changes. These drawbacks hinder the exploitation of these materials for practical applications as
483 piezoresistive switches.
484 Therefore, a laser surface treatment, causing the formation of conductive tracks, was exploited to
485 improve the piezoresistive response. Since this treatment causes the local increase of CNTs
486 concentration it was applied to composites with a filler content close to the percolation threshold.
487 However, different combinations of laser processing parameters were required for different CNTs
488 concentrations, in order to obtain tracks with enhanced conductivity and avoiding contemporary any
489 macroscopic damage of the material.
490 Samples carrying conductive tracks processed under optimized conditions showed improved
491 piezoresistive behaviour with respect to not functionalized SEBS/CNTs composites. The laser
492 surface treatment allowed to achieve good reproducibility of the piezoresistive response and
493 coherence between strain cyclically imposed and material resistance during 1000
494 stretching/releasing cycles.

495

496 **Acknowledgement**

497 This research did not receive any specific grant from funding agencies in the public, commercial, or
498 not-for-profit sectors.

499 Experimental Data file “Dataset "Piezoresistive properties of SEBS-CNTs composites"”, is
500 available on Mendeley Data, doi: 10.17632/pc4z4vkpfg.2

501 **References**

- 502 [1] D.G. Papageorgiou, Z. Li, M. Liu, I.A. Kinloch, R.J. Young, Mechanisms of mechanical
503 reinforcement by graphene and carbon nanotubes in polymer nanocomposites, *Nanoscale*. 12
504 (2020) 2228–2267. <https://doi.org/10.1039/c9nr06952f>.
- 505 [2] Y. Li, X. Huang, L. Zeng, R. Li, H. Tian, X. Fu, Y. Wang, W.H. Zhong, A review of the
506 electrical and mechanical properties of carbon nanofiller-reinforced polymer composites, *J.*
507 *Mater. Sci.* 54 (2019) 1036–1076. <https://doi.org/10.1007/s10853-018-3006-9>.
- 508 [3] S. Wu, S. Peng, C.H. Wang, Multifunctional polymer nanocomposites reinforced by aligned
509 carbon nanomaterials, *Polymers (Basel)*. 10 (2018). <https://doi.org/10.3390/polym10050542>.
- 510 [4] K. Hosseinpour, A.R. Ghasemi, Agglomeration and aspect ratio effects on the long-term
511 creep of carbon nanotubes/fiber/polymer composite cylindrical shells, *J. Sandw. Struct.*
512 *Mater.* (2019). <https://doi.org/10.1177/1099636219857200>.
- 513 [5] A.R. Ghasemi, M. Mohandes, R. Dimitri, F. Tornabene, Agglomeration effects on the
514 vibrations of CNTs/fiber/polymer/metal hybrid laminates cylindrical shell, *Compos. Part B*
515 *Eng.* 167 (2019) 700–716. <https://doi.org/10.1016/j.compositesb.2019.03.028>.
- 516 [6] Y. Xiao, X. Zhang, W. Cao, K. Wang, H. Tan, Q. Zhang, R. Du, Q. Fu, Dispersion and
517 Mechanical Properties of Polypropylene/ Multiwall Carbon Nanotubes Composites Obtained
518 via Dynamic Packing Injection Molding, *J. Appl. Polym. Sci.* 104 (2007) 1880–1886.
519 <https://doi.org/DOI 10.1002/app.25852>.
- 520 [7] N.H. Alamus, H. Fukunaga, S. Atobe, Y. Liu, J. Li, Piezoresistive Strain Sensors Made from
521 Carbon Nanotubes, (2011) 10691–10723. <https://doi.org/10.3390/s111110691>.
- 522 [8] W. Obitayo, T. Liu, A Review : Carbon Nanotube-Based Piezoresistive Strain Sensors, 2012
523 (2012). <https://doi.org/10.1155/2012/652438>.
- 524 [9] T. Yan, Z. Wang, Z. Pan, Flexible strain sensors fabricated using carbon-based
525 nanomaterials : A review, *Curr. Opin. Solid State Mater. Sci.* 22 (2018) 213–228.
526 <https://doi.org/10.1016/j.cossms.2018.11.001>.
- 527 [10] F. Aviles, A.I. Oliva-Aviles, M. Cen-puc, Piezoresistivity , Strain , and Damage Self-
528 Sensing of Polymer Composites Filled with Carbon Nanostructures, 1701159 (2018) 1–23.
529 <https://doi.org/10.1002/adem.201701159>.
- 530 [11] U. Szeluga, B. Kumanek, B. Trzebicka, Synergy in hybrid polymer / nanocarbon composites
531 . A review, *Compos. PART A.* 73 (2015) 204–231.
532 <https://doi.org/10.1016/j.compositesa.2015.02.021>.
- 533 [12] F. Avilés, A. May-pat, M.A. López-manchado, R. Verdejo, A. Bachmatiuk, A comparative
534 study on the mechanical , electrical and piezoresistive properties of polymer composites

- 535 using carbon nanostructures of different topology, *Eur. Polym. J.* 99 (2018) 394–402.
536 <https://doi.org/10.1016/j.eurpolymj.2017.12.038>.
- 537 [13] J. Cai, Y. Chen, J. Li, Y. Tan, J. Liu, X. Tang, X. Chen, M. Wang, Asymmetric deformation
538 in poly (ethylene-co-1-octene)/ multi-walled carbon nanotube composites with glass micro-
539 beads for highly piezoresistive sensitivity, *Chem. Eng. J.* 370 (2019) 176–184.
540 <https://doi.org/10.1016/j.cej.2019.03.223>.
- 541 [14] K. Ke, V.S. Bonab, D. Yuan, I. Manas-zloczower, Piezoresistive thermoplastic polyurethane
542 nanocomposites with carbon nanostructures, *Carbon N. Y.* 139 (2018) 52–58.
543 <https://doi.org/10.1016/j.carbon.2018.06.037>.
- 544 [15] V. Eswaraiah, S.S. Jyothirmayee Aravind, K. Balasubramaniam, S. Ramaprabhu, Graphene-
545 functionalized carbon nanotubes for conducting polymer nanocomposites and their improved
546 strain sensing properties, *Macromol. Chem. Phys.* 214 (2013) 2439–2444.
547 <https://doi.org/10.1002/macp.201300242>.
- 548 [16] X. Fu, Stretchable strain sensor facilely fabricated based on multi-wall carbon nanotube
549 composites with excellent performance, *J. Mater. Sci.* 54 (2019) 2170–2180.
550 <https://doi.org/10.1007/s10853-018-2954-4>.
- 551 [17] P. Zhang, S. Lei, W. Fu, J. Niu, G. Liu, J. Qian, J. Sun, The effects of agglomerate on the
552 piezoresistivity of conductive carbon nanotube / polyvinylidene fluoride composites, *Sensors*
553 *Actuators A. Phys.* 281 (2018) 176–184. <https://doi.org/10.1016/j.sna.2018.08.037>.
- 554 [18] Z. Sang, K. Ke, I. Manas-zloczower, Effect of carbon nanotube morphology on properties in
555 thermoplastic elastomer composites for strain sensors, *Compos. Part A.* 121 (2019) 207–212.
556 <https://doi.org/10.1016/j.compositesa.2019.03.007>.
- 557 [19] J. Vicente, P. Costa, J.M. Abete, A. Iturraspe, Electromechanical Properties of PVDF-Based
558 Polymers Reinforced with Nanocarbonaceous Fillers for Pressure Sensing Applications,
559 (n.d.).
- 560 [20] X. Yang, L. Sun, C. Zhang, B. Huang, Y. Chu, Modulating the sensing behaviors of poly (
561 styrene-ethylene-butylene- styrene)/ carbon nanotubes with low-dimensional fi llers for large
562 deformation sensors, *Compos. Part B.* 160 (2019) 605–614.
563 <https://doi.org/10.1016/j.compositesb.2018.12.119>.
- 564 [21] S. Xu, H. Hu, L. Ji, P. Wang, Piezoresistive Properties of Multi-Walled Carbon Nanotube /
565 Silicone Rubber Composites under Cyclic Loads with AC Excitation Piezoresistive
566 Properties of Multi-Walled Carbon Nanotube / Silicone Rubber Composites under Cyclic
567 Loads with AC Excitation, (2019). <https://doi.org/10.1088/1742-6596/1168/2/022075>.
- 568 [22] G. Spinelli, P. Lamberti, V. Tucci, L. Vertuccio, L. Guadagno, Experimental and theoretical

- 569 study on piezoresistive properties of a structural resin reinforced with carbon nanotubes for
570 strain sensing and damage monitoring, *Compos. Part B.* 145 (2018) 90–99.
571 <https://doi.org/10.1016/j.compositesb.2018.03.025>.
- 572 [23] S. Kumar, T.K. Gupta, K.M. Varadarajan, Strong , stretchable and ultrasensitive MWCNT /
573 TPU nanocomposites for piezoresistive strain sensing, *Compos. Part B.* 177 (2019) 107285.
574 <https://doi.org/10.1016/j.compositesb.2019.107285>.
- 575 [24] Y. Wang, S. Wang, M. Li, Y. Gu, Z. Zhang, Piezoresistive response of carbon nanotube
576 composite film under laterally compressive strain, *Sensors Actuators A. Phys.* 273 (2018)
577 140–146. <https://doi.org/10.1016/j.sna.2018.02.032>.
- 578 [25] Q. Li, S. Luo, Q. Wang, Piezoresistive thin film pressure sensor based on carbon nanotube-
579 polyimide nanocomposites, *Sensors Actuators A. Phys.* 295 (2019) 336–342.
580 <https://doi.org/10.1016/j.sna.2019.06.017>.
- 581 [26] M. Li, J. Wang, S. Wang, Effect of microstructure on the piezoresistive behavior of carbon
582 nanotube composite film Effect of microstructure on the piezoresistive behavior of carbon
583 nanotube composite film, (2019).
- 584 [27] S.M. Doshi, E.T. Thostenson, Thin and Flexible Carbon Nanotube-Based Pressure Sensors
585 with Ultrawide Sensing Range, *ACS Sensors.* 3 (2018) 1276–1282.
586 <https://doi.org/10.1021/acssensors.8b00378>.
- 587 [28] Y. Tan, J. Li, Y. Chen, X. Tang, J. Cai, J. Liu, M. Wang, Gentle crosslinking to enhance
588 interfacial interaction in thermoplastic polyurethane / poly (ethylene-co-1-octene)/ multi-
589 walled carbon nanotube composites for conductive improvement and piezoresistive stability,
590 *Polym. Test.* 75 (2019) 142–150. <https://doi.org/10.1016/j.polymertesting.2019.02.004>.
- 591 [29] E. Subramani, S. Raman, K. Werner, S. Wießner, G. Heinrich, A. Das, High-performance
592 elastomeric strain sensors based on nanostructured carbon fillers for potential tire
593 applications, *Mater. Today Commun.* 14 (2018) 240–248.
594 <https://doi.org/10.1016/j.mtcomm.2018.01.013>.
- 595 [30] J. Ramon Dios, C. García-Astrain, P. Costa, S. Viana, Júlio César Lanceros-Méndez,
596 Carbonaceous Filler Type and Content Dependence of, *Materials (Basel).* 12 (2019) 1405–
597 1420.
- 598 [31] M. Kim, J. Jung, S. Jung, Y.H. Moon, Piezoresistive Behaviour of Additively Manufactured
599 Multi-Walled Carbon Nanotube/Thermoplastic, (n.d.).
- 600 [32] A. Caradonna, C. Badini, E. Padovano, A. Veca, E. De Meo, M. Pietroluongo, Laser
601 Treatments for Improving Electrical Conductivity and Piezoresistive Behavior of Polymer –
602 Carbon Nanofiller Composites, (2019). <https://doi.org/10.3390/mi10010063>.

- 603 [33] J. Tiusanen, D. Vlasveld, J. Vuorinen, Review on the effects of injection moulding
604 parameters on the electrical resistivity of carbon nanotube filled polymer parts, *Compos. Sci.*
605 *Technol.* 72 (2012) 1741–1752. <https://doi.org/10.1016/j.compscitech.2012.07.009>.
- 606 [34] T.B. Nguyen Thi, S. Ata, T. Morimoto, T. Okazaki, T. Yamada, K. Hata, Visualizing
607 electrical network in microinjection-molded CNT polycarbonate composite, *Carbon N. Y.*
608 153 (2019) 136–147. <https://doi.org/10.1016/j.carbon.2019.07.019>.
- 609 [35] T. Villmow, S. Pegel, P. Pötschke, U. Wagenknecht, Influence of injection molding
610 parameters on the electrical resistivity of polycarbonate filled with multi-walled carbon
611 nanotubes, *Compos. Sci. Technol.* 68 (2008) 777–789.
612 <https://doi.org/10.1016/j.compscitech.2007.08.031>.
- 613 [36] C.M. Hong, J. Kim, S.C. Jana, Shear-induced migration of conductive fillers in injection
614 molding, *Polym. Eng. Sci.* 44 (2004) 2101–2109. <https://doi.org/10.1002/pen.20215>.
- 615 [37] G. Colucci, C. Beltrame, M. Giorcelli, A. Veca, C. Badini, A novel approach to obtain
616 conductive tracks on PP/MWCNT nanocomposites by laser printing, *RSC Adv.* 6 (2016)
617 28522–28531. <https://doi.org/10.1039/c6ra02726a>.
- 618
619

620 **Figure Captions**

621

622 **Figure 1.** Experimental set up for electromechanical testing.

623 **Figure 2.** Cryofracture surface of SEBS/CNTs nanocomposites: (A) 3% wt. of nanotubes, (B) 7%
624 wt. of nanotubes.

625 **Figure 3.** Resistance change with the displacement increase.

626 **Figure 4.** Cyclic testing of nanocomposites: programmed displacement cycle (blue curve), ideal
627 resistance variation (red dotted curve), and an example of experimental result of resistance change
628 (red continuous curve). Speed of resistance variation (v_1, v_2, v_3) and resistance changes ($\Delta R_1,$
629 ΔR_2) observed in the different parts of the cycle. Delay time for the recovery of the initial
630 resistance (t) and deviation from the pristine resistance value at the end of the cycle (R_b).

631 **Figure 5.** Displacement and resistance variation during a cycle under different conditions for
632 SEBS with 5% CNTs: displacement rate = 10 mm/min, (A) maximum displacement (MD) of 5 mm
633 and stay at the maximum displacement = 2s; (B) MD of 5 mm and stay at the MD = 10s; (C) MD of
634 10 mm and stay at the MD = 2s; (D) MD of 10 mm and stay at the MD = 10s.

635 **Figure 6.** Piezoresistive response during repeated cycles: (A) SEBS-5 during the initial ten cycles,
636 (B) SEBS-6 during 100 cycles.

637 **Figure 7.** Performance of conductive tracks obtained by laser writing on composite with (A) 3% of
638 CNTs and (B) 4% of CNTs

639 **Figure 8.** Nanocomposite with 3% of CNTs after laser treatment ($v= 200$ mm/s, $P= 5\%$, $N= 30$):
640 profile of the track (B) measured along the red line (A); nanocomposite with 4% of CNTs after laser
641 treatment ($v= 100$ mm/s, $P= 5\%$, $N= 10$): profile of the track (D) measured along the red line (C).

642 **Figure 9.** Change of resistance (red curve) and displacement (blue curve) during mechanical
643 cycling. (A) 1000 cycles performed on SEBS-3 (B) coherence of displacement and resistance
644 variations during cycling of SEBS-3, (C) 1000 cycles performed on nanocomposite SEBS-4, (D)
645 coherence of displacement and resistance variations during cycling of nanocomposite SEBS-4.

646

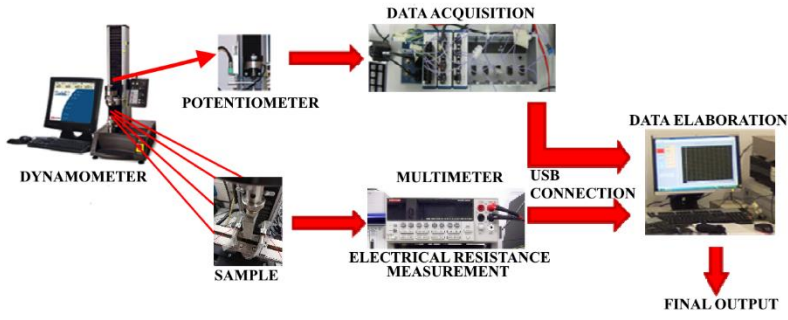
647

648

649

650

651 **Figures**

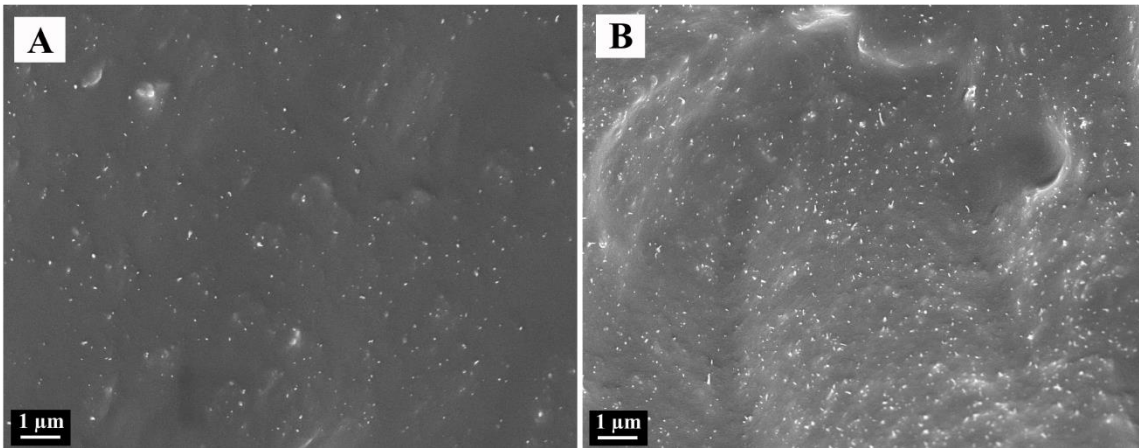


652

653 **Figure 1**

654

655

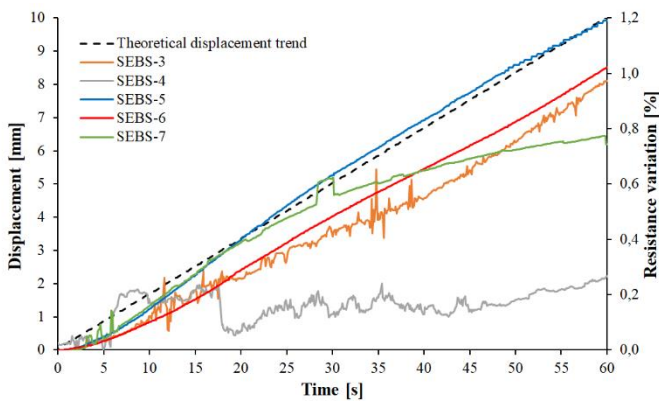


656

657 **Figure 2**

658

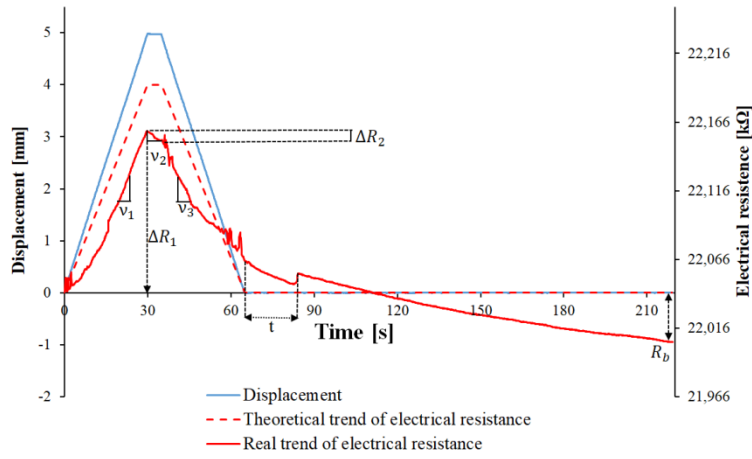
659



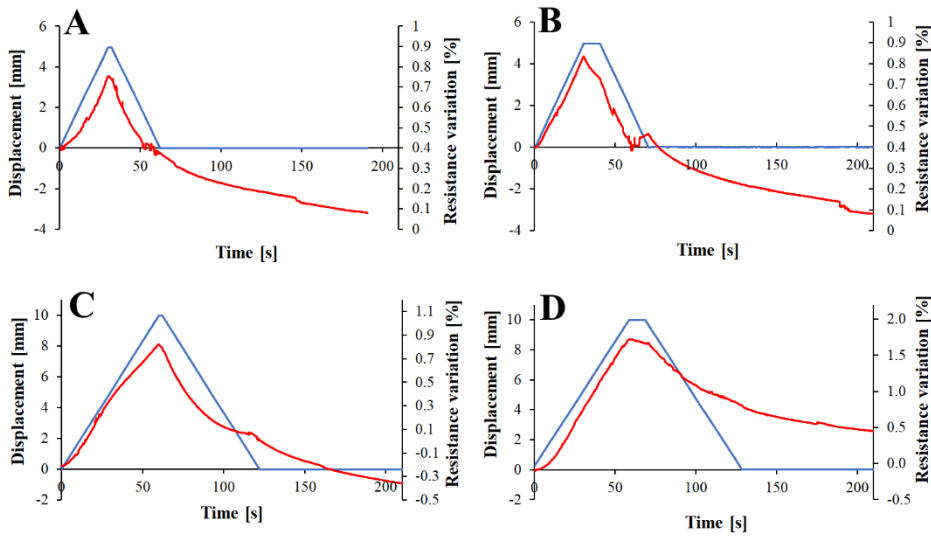
660

661 **Figure 3**

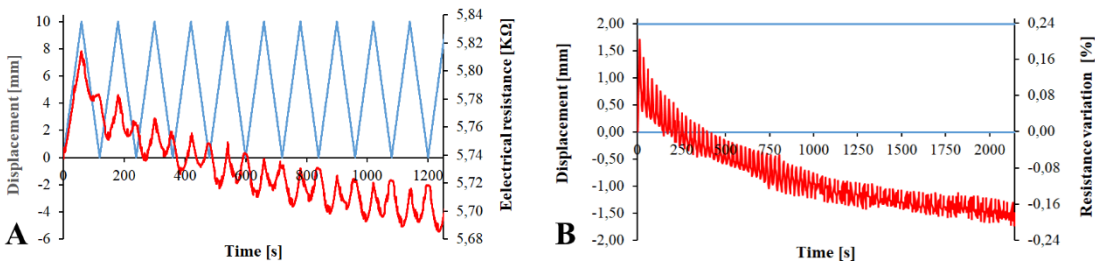
662



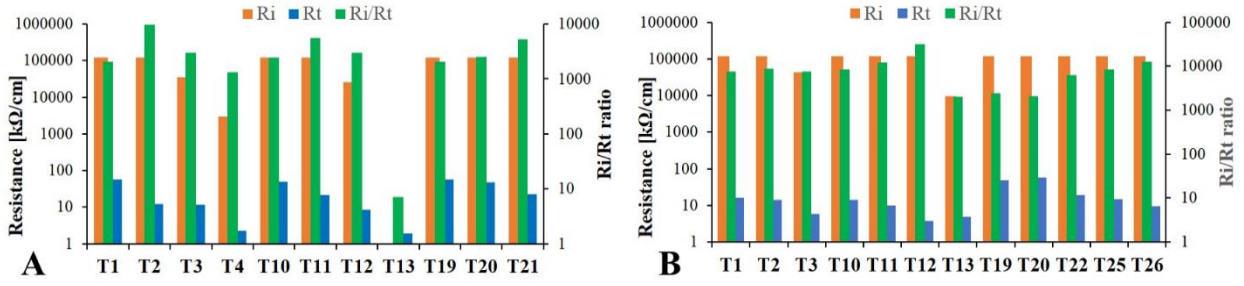
663
664 Figure 4
665
666



667
668 Figure 5
669
670

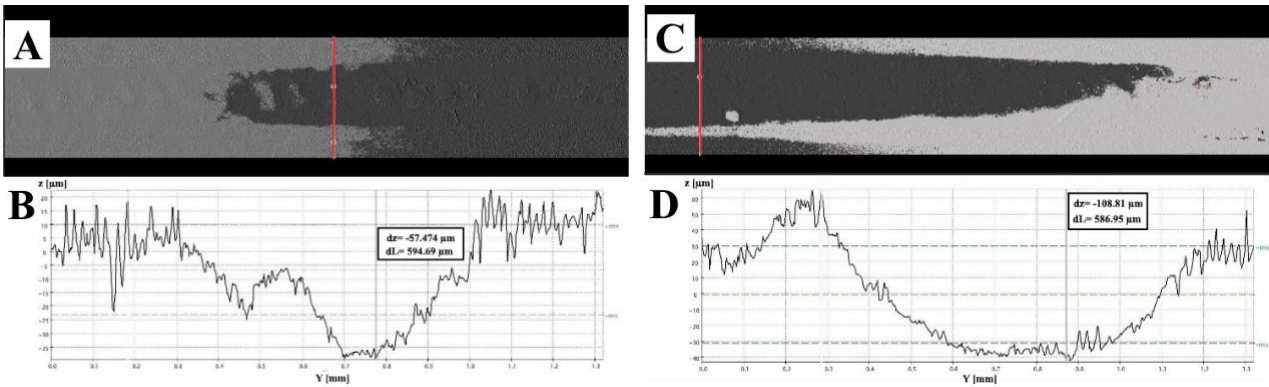


671
672 Figure 6
673



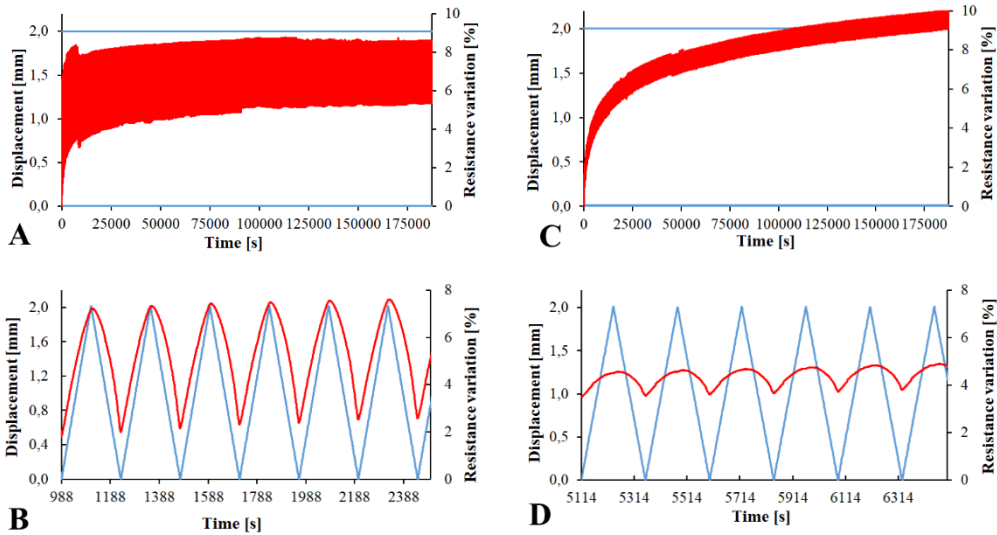
674
675 Figure 7

676
677



678
679 Figure 8

680
681



682
683 Figure 9

684

685 **Tables**

686

687 **Table 1.** Surface resistance of composites with different percent of CNTs

Sample	Resistance (kΩ/cm)
SEBS-3	3400 ± 1358
SEBS-4	86.5 ± 31.7
SEBS-5	3.25 ± 1.50
SEBS-6	0.37 ± 0.09
SEBS-7	0.12 ± 0.09

688

689

Table 2. Loading/unloading tests of nanocomposites: values of **adopted testing parameters**.

Displacement, Holding time	CNTs (%)	ΔR_1 (%)	ΔR_2 (%)	R_b (%)	t (s)	v_1 (Ω/s)	v_2 (Ω/s)	v_3 (Ω/s)	v_3/v_2
2mm,2s	5	-0.32	-0.03	-0.73	(*)	-2.09	-2.60	-0.87	0.34
	6	0.14	-0.01	-0.15	0.28	0.17	-0.01	-0.13	18.83
2mm,5s	5	0.13	-0.04	-0.30	4.91	1.04	-0.81	-1.18	1.47
	6	0.03	-0.03	-0.17	7.80	0.067	-0.08	-0.07	0.90
2mm,10s	5	0.09	-0.09	-0.36	11.19	1.68	-1.92	-2.30	1.25
	6	0.07	-0.04	-0.17	4.30	0.08	-0.07	-0.78	10.68
5mm,2s	5	0.34	-0.02	-0.32	9.20	2.57	-2.10	-2.63	1.25
	6	0.36	-0.02	-0.18	1.50	0.17	-0.08	-0.16	2.07
5mm,5s	5	0.51	-0.03	-0.18	16.60	3.49	-1.31	-2.70	2.63
	6	0.67	-0.05	0.04	(*)	0.29	-0.10	-0.17	1.70
5mm,10s	5	0.42	-0.10	-0.31	10.76	1.30	-0.99	-1.30	1.39
	6	0.22	-0.11	-0.43	23.78	0.16	-0.28	-0.26	0.92
10mm,2s	5	0.98	-0.03	-0.24	36.40	2.30	-3.88	-2.24	0.58
	6	0.58	-0.02	-0.03	99.74	0.32	-0.27	-0.31	1.04
10mm,5s	5	1.45	-0.06	0.22	(*)	2.20	-0.92	-1.22	1.30
	6	0.92	-0.06	-0.01	115.40	0.34	-0.25	-0.22	0.89
10mm,10s	5	1.81	-0.08	0.46	(*)	6.80	-1.31	-5.22	3.98
	6	0.42	-0.12	-0.54	44.27	0.25	-0.35	-0.35	1.00

(*) The sample never recovered the pristine resistance value during the test.

705

706

707

708

709

710

711

712

713

714

715

716

717

718

719

720

721

722

723

Table 3. Set of parameters for the laser writing trials and resulting electrical resistance.

Trial	SEBS-3					SEBS-4		
	V (mm/s)	P (%)	N repetition	R _t (kΩ/cm)	R _i (kΩ/cm)	N repetition	R _t (kΩ/cm)	R _i (kΩ/cm)
T1	300	5	30	57.84 ± 35.22	OR	15	16.15 ± 0.18	OR
T2	200	5	30	12.32 ± 3.55	OR	15	13.85 ± 0.25	OR
T3	100	5	30	11.63 ± 3.11	34994 ± 1	15	5.85 ± 0.05	43700 ± 1
T4	300	10	30	2.23 ± 0.13	2980 ± 2	15	4.83 ± 0.54	440 ± 5
T5	200	10	30	1.55 ± 0.18	79 ± 1	15	0.47 ± 0.14	2.1 ± 0.1
T6	100	10	30	0.89 ± 0.1	11 ± 1	15	1.00 ± 0.07	6.7 ± 0.1
T7	300	20	30	0.57 ± 0.07	2.8 ± 0.2	15	1.11 ± 0.18	4.3 ± 0.3
T8	200	20	30	0.21 ± 0.04	0.8 ± 0.1	15	0.49 ± 0.08	1.9 ± 0.1
T9	100	20	30	/	/	15	0.24 ± 0.03	0.9 ± 0.1
T10	300	5	20	49.16 ± 35.84	OR	10	14.31 ± 1.01	OR
T11	200	5	20	21.44 ± 2.11	OR	10	10.01 ± 0.14	OR
T12	100	5	20	8.47 ± 4.20	25400 ± 1	10	3.77 ± 0.65	OR
T13	300	10	20	1.94 ± 0.31	14 ± 1	10	4.92 ± 0.10	9700 ± 99
T14	200	10	20	2.71 ± 0.06	3900 ± 11	10	2.03 ± 0.97	9.5 ± 0.6
T15	100	10	20	1.32 ± 0.11	54.5 ± 0.1	10	2.15 ± 0.30	11.2 ± 0.1
T16	300	20	20	2.00 ± 0.32	13.4 ± 0.1	10	6.11 ± 0.48	84 ± 1
T17	200	20	20	0.80 ± 0.19	3.4 ± 0.3	10	0.49 ± 0.08	1.8 ± 0.1
T18	100	20	20	/	/	10	0.18 ± 0.01	0.7 ± 0.1
T19	300	5	10	57.66 ± 9.11	OR	5	49.2 ± 16.30	OR
T20	200	5	10	47.73 ± 22.84	OR	5	57.56 ± 22.89	OR
T21	100	5	10	22.74 ± 0.40	OR	5	15.48 ± 1.28	OR
T22	300	10	10	10.64 ± 0.46	6.1 ± 0.1	5	19.5 ± 3.63	OR
T23	200	10	10	5.65 ± 1.01	120 ± 1	5	12.79 ± 2.69	3050 ± 3
T24	100	10	10	4.89 ± 0.26	250 ± 3	5	3.64 ± 1.96	17.4 ± 0.1
T25	300	20	10	8.86 ± 1.54	250 ± 2	5	14.51 ± 1.04	OR
T26	200	20	10	2.55 ± 0.17	11 ± 1	5	9.61 ± 0.51	OR
T27	100	20	10	0.66 ± 0.17	2.5 ± 1.0	5	0.97 ± 0.14	3.2 ± 0.1

725

726

727

728

729

730