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Antibacterial, highly hydrophobic and semi transparent Ag/plasma polymer nanocomposite coating on cotton fabric obtained by plasma based co-deposition

#### Original

Antibacterial, highly hydrophobic and semi transparent Ag/plasma polymer nanocomposite coating on cotton fabric obtained by plasma based co-deposition / Irfan, M.; Polonskyi, O.; Hinz, A.; Mollea, C.; Bosco, F.; Strunskus, T.; Balagna, C.; Perero, S.; Faupel, F.; Ferraris, M.. - In: CELLULOSE. - ISSN 0969-0239. - ELETTRONICO. - (2019). [10.1007/s10570-019-02685-6]

Availability:

This version is available at: 11583/2749413 since: 2019-09-03T10:11:46Z

Publisher: Springer

**Published** 

DOI:10.1007/s10570-019-02685-6

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- 5 plasma based co-deposition
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## **Abstract**

This study aims at deposition and characterization of antibacterial, hydrophobic and semitransparent metal/plasma polymer nanocomposite coating, containing Ag nanoparticles, onto cotton fabrics intended to be used in medical applications. The nano composite coatings were obtained via a simple, one step and ecofriendly plasma based co-deposition approach where silver was magnetron sputtered simultaneously with plasma polymerization of hexamethyldisiloxane (HMDSO) monomer. The nanocomposite thin films containing different concentration of silver were deposited either by varying silver sputter rate or thickness of the plasma polymer matrix to obtain a good balance between optical properties of the coated fabric and its long term antibacterial performance. The obtained coatings were investigated in detail with respect to their composition, morphology, optical properties, nanoparticle size distribution, silver ion release efficiency, antibacterial performance, water contact angle and washing stability of the coating. The thickness of the plasma matrix was found to be more important in controlling the release of silver ions as well as affecting the optical properties of the coating. The water contact angle on the coated fabric was up to 145°, close to super hydrophobicity. The coating showed effective antibacterial efficacy against Staphylococcus epidermidis (a Gram positive bacterium) which was present even when fabric was subjected to 10 repeated washing cycles indicating good washing stability of the coating.

**Key words:** plasma polymerization, sputtering, silver nanoparticles, plasma

53 polymer, optical properties, silver ion release properties

## 69 1. Introduction

70 Medical textiles used in the health care infrastructures are an important potential 71 source of nosocomial infections. In general, bacterial infections pose a great threat 72 worldwide not only to the public health but also to the economy as they elongate 73 the average hospitalization period along with other associated costs. Since textiles 74 are in close contact with human body, they should not be a source of transmitting 75 infectious diseases to patients or health care workers (Perelshtein et al. 2015). On 76 the other hand, because of their greater surface area as well as ability to retain 77 moisture, textiles provide a conducive environment for microorganisms to grow. 78 Being closer to the skin of the wearer, they can contribute to the cross-79 contamination of pathogenic bacteria (Brunon et al. 2011) in different 80 environments like home, hospitals and food industry where textiles are used 81 immensely (Chadeau et al. 2010). Particularly, fabrics made of cellulosic fibers, 82 for example cotton fabric, are more susceptible to microbial growth than synthetic 83 fibers due to their ability to retain moisture. However, cotton fabrics provide 84 superior comfort properties and thus are used widely not only in traditional 85 textiles but also in medical textiles (Fei et al. 2018). This necessitates antibacterial 86 functionalization of the textile surfaces and provides an impetus for research in 87 antibacterial textiles. Consequently, research output regarding antibacterial 88 functionalization of textile surfaces has increased tremendously. 89 Various antibacterial agents have been studied for imparting antibacterial 90 functionality to textile surfaces which include organic antibacterial agents e.g. 91 quaternary ammonium compounds (Lin et al. 2018), chitosan (Zemljič et al. 2017) 92 and its derivatives (Stawski et al. 2016), N halamines (Liu et al. 2015), triclosan 93 (Foksowicz-Flaczyk et al. 2016), polybiguanides (Gao and Cranston 2010), 94 inorganic antibacterial agents e.g Ag (Xu et al. 2017) and Pt and Zn (Ponomarev 95 et al. 2018) nanoparticles, CuO (El-Nahhal et al. 2018), ZnO (Ghayempour et al. 96 2017) and natural antibacterial agents e.g natural dyes (Mariselvam et al. 2017), 97 curcumin (Pisitsak et al. 2015), aloe vera (Ali et al. 2014) and other plant extracts 98 (Savoia 2012). Among these, silver nano particles are attractive because of their 99 antimicrobial performance against a wide variety of bacteria and fungi (Wu et al. 100 2018). Although cytotoxicity of metal nanoparticles is a matter of concern, 101 however, silver nanoparticles are considered less toxic to humans at lower 102 concentrations (Jamuna-Thevi et al. 2011). Most of the wet chemistry routes for

103 the synthesis and application of silver nano particles to textiles have 104 disadvantages of agglomeration, non-uniform dispersion, use of expansive (and 105 sometimes toxic) chemicals and complex chemical processes (Wu et al. 2018) as 106 well as have environmental concerns. Additionally, wet treatments may require 107 higher quantities of antibacterial agents for achieving maximum antibacterial 108 activity resulting in high weight add-on on the fabric. Lin J et. al. (Lin et al. 2018) 109 reported 8% weight add-on on pristine cotton fabric after coating polymeric antibacterial agent to achieve 100% bacterial reduction. The treatment was also 110 111 reported to cause 6% reduction in the air permeability of the cotton fabric due to 112 blockage of interfiber spaces. 113 Alternatively, plasma based processes including plasma polymerization and 114 sputtering are increasingly being studied for textile surface functionalization due 115 to their environment friendly nature as well as their ability to modify only the 116 surface of the textiles while preserving their bulk properties (Irfan et al. 2017). 117 Various plasma techniques (sputtering, plasma polymerization etc) can be used 118 separately or in combination to fabricate metal nanoparticles embedded in a 119 matrix (Kratochvil et al. 2018). Among these, nano composite films composed of 120 silver nano particles embedded in a matrix, obtained by co-sputtering or sputtering 121 and plasma polymerization, have been studied extensively because of their 122 interesting functional properties (Hlídek et al. 2009). Nano composite films 123 comprising metal nano particles embedded in an inorganic matrix are relatively 124 simpler to obtain via co-sputtering (Irfan et al. 2017) than those where metal 125 particles are embedded in a plasma polymerized matrix. Hexamethyldisiloxane (HMDSO) is the most commonly used plasma polymer precursor to fabricate 126 127 nano composite coatings containing silver particles. 128 A variety of configurations of plasma techniques to obtain silver nanoparticles 129 embedded in a plasma polymer matrix have been reported. These include single 130 electrode deposition of both metal nanoparticles and polymer matrix (Hlídek et al. 131 2009; Despax and Raynaud 2007; Körner et al. 2010; Peter et al. 2011), 132 fabrication of metal nanoparticles separately via gas phase condensation (GPC) 133 (Schmittgens, Wolf and Schultheiss 2009) or gas phase aggregation (Kuzminova 134 et al. 2016, Kylian et al. 2017) of silver nano particles and various others 135 configurations (Brunon et al. 2011; Beyene et al. 2010; Deng et al. 2014). Each 136 technique has its own merits and demerits. In single electrode deposition, the

137 deposition rate of silver and plasma polymer needs to be balanced to obtain 138 homogeneous film due to silver target poisoning with plasma polymer (Körner et 139 al. 2010; Peter et al. 2011). While the synthesis of matrix and nanoparticles in 140 separate plasma regions necessitates the instalment of additional gas phase 141 aggregation or condensation source on the sputtering chamber (Schmittgens, Wolf 142 and Schultheiss 2009; Kuzminova et al. 2016). 143 Given the importance of aesthetic properties of textiles, transparent or 144 semitransparent antibacterial coatings with effective broad spectrum antibacterial 145 activity along with high washing stability has always been an area of interest for 146 researchers. Nonetheless, for most of the coatings, a trade off is achieved between 147 being antibacterial, transparent as well as having high washing stability. Silver 148 nano materials are well-known antimicrobial agents effective against various 149 types of microorganisms. However, whether incorporated within a polymer matrix 150 (Ramirez et al. 2018) or coated on the surface of a substrate (Brunon et al. 2011), 151 they impart color to the matrix or substrate due to surface plasmon resonance of 152 the silver nanoparticles or clusters. There are studies reporting antibacterial 153 textiles coated with silver nano particles/clusters with an acceptable level of 154 transparency. Most of these approaches include lowering the silver concentration 155 in the coating (Brunon et al. 2011; Chadeau et al. 2010) to reduce the intensity of 156 surface plasmon resonance of the silver particles. But this can limit antibacterial 157 performance or compromise long term antimicrobial activity which is associated 158 with sustained release of silver ions (Körner et al. 2010) from the coating. 159 In this study nano composite coating consisting of silver nano particles embedded 160 in plasma polymer matrix was deposited via a simple plasma based co-deposition 161 scheme on a green colored cotton fabric meant to be used in medical applications. 162 Separate power supplies to the two sputtering electrodes, along with other 163 operational parameters, ensured independent control and manipulation of the 164 matrix, silver nanoparticles amount, size and their distribution within the matrix 165 with no observation of silver target poisoning. In addition, high deposition rates 166 are possible to achieve. The matrix precursor monomer (HMDSO) was 167 introduced, polymerized and deposited from the surface of an RF electrode 168 whereas silver was simultaneously co-sputtered from DC electrode. Using this 169 scheme, five different compositions of the composite coating were obtained by varying the silver concentration to have a balance between aesthetic look and 170

antibacterial performance of the fabric. The silver concentration was varied by two approaches to reduce the coloration of the fabric. In first approach, silver concentration was lowered to minimum in a 150 nm thick composite thin film. While in the second approach, silver dispersion was increased in a relatively thicker matrix while keeping the silver concentration per unit fabric area higher to get the benefit of reduced coloration combined with silver concentration enough for sustained silver ion release ability of the coated fabric. The obtained coatings were then investigated in detail for their morphology, composition, silver nano particle size and distribution, optical properties, potential of silver ion release, antibacterial effect against a Gram + bacterium and washing stability of the coating on the fabric.

# 2. Experimental

#### 2.1 Method

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The schematics of the deposition system used in this study is shown in Fig. 1. The system was composed of a main cylindrical ultra-high vacuum (UHV) chamber equipped with turbo molecular pump and rotary pump to create high vacuum. Two balanced magnetrons (Thin Film Consulting, ION'X-2UHV, diameter of 2 inch), one connected to the RF generator (13.56 MHz) and the other one connected to the DC power supply, were installed on the main chamber. RF magnetron was used to deposit the plasma polymer and was equipped with a carbon target in order to prevent metal sputtering from the magnetron surface. DC magnetron was equipped with a silver target (2" dia, 99.99% Testbourne Ltd.) for sputter deposition of silver. The simultaneous deposition of silver and plasma polymer resulted in a nanocomposite thin film composed of plasma polymer matrix and silver nanoparticle inclusions. Due to independent power supplies, both the matrix thickness and silver concentration can be controlled separately. The matrix of the composite coating was obtained via plasma polymerization of the precursor hexamethyldisiloxane (HMDSO) which was introduced into the vacuum chamber through an inlet in the magnetron. Ar was used as working gas and was introduced in the chamber through an inlet in the DC magnetron. During simultaneous deposition, silver was deposited within the growing plasma polymer matrix on the surface of the substrate where nanoparticles are formed via self-

203 organization. With the adopted scheme, no evidence of silver target poisoning due 204 to plasma polymer sputtering was observed as has been reported in literature for 205 such co-depositions (Drábik et al. 2015). 206 The silver concentration within the composite coating was varied by two different 207 approaches: (1) by varying the deposition power of silver while keeping the 208 coating thickness (or more accurately the matrix thickness) same or (2) by 209 increasing the matrix thickness and keeping the silver deposition power constant. 210 The objective of the two approaches was to obtain a balance between an 211 acceptable level of transparency by reducing the silver concentration or particle 212 size while maintaining long-term antibacterial effect. The second approach was 213 adopted to get higher silver concentration on a unit fabric area with its expected 214 properties of controlled release of silver ions from relatively thicker coating 215 matrix that can prove beneficial for extended life of the coated fabric in repeated 216 use or when subjected to multiple washing cycles. 217 In order to obtain the coatings with above two approaches, a composite coating 218 with 150 nm thickness with maximum silver concentration obtained at silver 219 sputtering power of 50 W was selected as starting point for this study. This starting point was selected after initial investigations keeping in view that even 220 221 this maximum silver concentration did not completely overshadow the green color 222 of the cotton fabric in visual inspection thus preserving the original aesthetic look 223 of the fabric to reasonable extent. The coating was obtained by applying a DC 224 power of 50W to silver and RF power of 25W for HMDSO at 0.2 sccm monomer 225 flow and 10 sccm Argon flow. Next, silver concentration in the coating was 226 reduced either (1) by reducing silver sputtering power to 30W and 15W and 227 keeping HMDSO deposition parameters the same or (2) by keeping the silver 228 sputtering power at 50W and increasing HMDSO flow (up to 0.5 sccm) to obtain 229 greater dispersion of silver in a thicker matrix. The system was evacuated to a base pressure of about 3.0 x 10<sup>-4</sup> Pa and deposition pressure was about 0.4 Pa. 230 231 Table 1 summarizes the obtained coatings along with respective process which 232 were deposited after preliminary investigations to find stable RF plasma 233 discharge. The deposition time was kept constant for each deposition (10 min) 234 with constant Ar flow (10 sccm). The samples are labelled by following the 235 scheme (CF-coating thickness-silver sputtering power) where CF represents 236 cotton fabric.

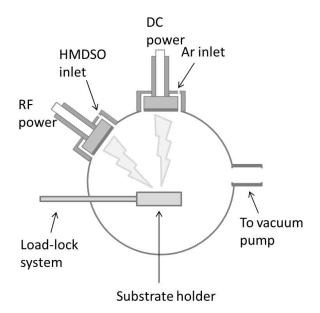


Fig. 1 Schematic of the deposition system

240 Table 1 List of samples together with corresponding deposition conditions

Sample	RF Power (W)	DC Power (W)	Ar flow (sccm)	HMDSO flow (sccm)	Coating Thickness (nm)	Silver filling factor (%)
CF-150nm- 50W	25 W	50 W	10	0.2	$150 \pm 7$	13
CF-150nm- 30W	25 W	30 W	10	0.2	150 ± 5	8
CF-150nm- 15W	25 W	15 W	10	0.2	150 ± 8	5
CF-300nm- 50W	25 W	50 W	10	0.4	300 ± 10	8
CF-400nm- 50W	25 W	50 W	10	0.5	400 ± 12	6

#### 242 **2.2 Characterization**

243 Chemical composition of the deposited polymer matrix was analyzed using 244 Fourier transform infrared reflection absorption spectroscopy (FTIR-RAS, Bruker 245 Equinox 55). FTIR spectra were obtained for plasma polymer coatings deposited 246 at 10, 25 and 50 W RF. Results were analyzed (discussed in the result section) and 247 25 W was selected to deposit plasma polymer throughout the study. In addition, 248 FTIR spectra were also obtained for thicker polymer matrix obtained by 249 increasing HMDSO flow while keeping RF power constant at 25W. The spectra 250 were obtained in the wave length range 500-4000 cm<sup>-1</sup> and resolution of 4 cm<sup>-1</sup>. For FTIR analysis, polymer coatings were deposited on gold coated silicon 251 252 wafers. 253 Analysis of the surface chemistry of the cotton fabrics coated with nanocomposite 254 films was performed using X-ray photoelectron spectroscopy (XPS, Omicron 255 Nanotechnology GmbH) operating with Al anode at a power of 240 W. In order to 256 detect the elements present on the surface the survey spectra were recorded (with 257 pass energies of 100 eV). High-resolution spectra were obtained for the elements 258 detected during survey scans using pass energy of 30 eV. The morphology of the 259 composite coating was analyzed using scanning electron microscope (SEM, Zeiss 260 Ultra Plus). Silver nano particle's shape and size distribution were determined 261 using transmission electron microscopy (TEM, JEM-2100, JEOL, 200 kV, LaB6). 262 TEM samples were prepared by depositing 25-30 nm of composite coating on 263 carbon coated copper grids. The obtained images were processed by an image 264 processing software ImageJ (ImageJ) and size histograms for silver nanoparticles 265 were obtained. 266 Optical properties of the composite coating were assessed using UV-Vis 267 spectroscopy (Ellipsometer Woollam M2000 UI). Coatings were deposited on quartz glass to obtain UV-Vis transmittance spectra in the wavelength range 250-268 269 1000 nm. The total concentration of the silver deposited under three silver 270 sputtering powers (15W, 30W and 50W) was determined through Inductively 271 Coupled Plasma Mass Spectroscopy (ICP-MS, Thermo Scientific iCAPTM Q ICP-MS). For this purpose, coated cotton samples (1x1 cm<sup>2</sup>) were dissolved in a 272 273 mixture of nitric acid (65%) and H<sub>2</sub>O<sub>2</sub> (30%). The liquor was then filtered. The 274 instrument (ICP-MS) was calibrated using solution containing 125, 250, 500 and 275 1000 ppb of silver. The potential of the composite coatings to release silver ions 276 in aqueous media was determined through silver ion release test. In silver ion 277 release test, fabric samples (1x1 cm<sup>2</sup>) were immersed in 10 ml water (milliQ) in 278 small plastic bottles at room temperature. 1 ml water sample was drawn from the 279 bottles after 3, 24 and 72 hours and quantity of ionic silver leached from the fabric 280 was determined through ICP-MS (Thermo Scientific iCAPTM Q ICP-MS). 281 Three samples for each coating were tested and average concentration was 282 reported. 283 Antibacterial performance of the cotton fabric coated by nanocomposite thin film 284 was assessed via "Inhibition halo test" performed against Staphylococcus 285 epidermidis LMG 10474, a Gram positive bacterium. For the test, a bacterial 286 inoculum was prepared by means of a water suspension of the colonies grown 287 over night on Nutrient agar plate at 37°C, the suspension was diluted to obtain a 288 value of optical density at 620 nm (O.D.<sub>620</sub>) between 0.8 and 1. The suspension 289 was spread on the surface of Mueller Hinton agar plate by means of an inoculating 290 loop. Coated and uncoated cotton fabric samples (1x1 cm2) were placed in 291 contact with the inoculated agar plates and incubated at 37 °C for 24 hours. At the 292 end of the incubation period the microbial growth was observed to identify the 293 presence of the inhibition halo around the fabric samples and/or the lack of the 294 growth under them. 295 In order to analyze the washing stability of the coating, the "Inhibition halo test" 296 was also performed on coated fabric samples subjected to 10 washing cycles. 297 Coated fabric samples of size 2x2 cm<sup>2</sup> were washed in 100 ml water in a beaker. 298 Washing was performed in a thermostat bath at 60 °C for 30 minutes with 299 oscillations set at 150 per minute. Soap solution was composed of AATCC 300 standard detergent without optical brightener (WOB) at a concentration of 2 g/l. 301 Energy Dispersive X ray spectroscopy (EDX) was also performed on washed 302 samples to assess the presence of the coating on fabric surface by detecting the 303 elements of coating. 304 Finally, the water wettability of the coated fabric surface was determined with 305 water contact angle method. A water droplet of 5 µl was dropped on the surface of 306 the coated and uncoated fabric samples and image was captured after 10 seconds 307 to measure contact angle. Five measurements were taken for each coating and 308 average value of contact angle was reported.

## 3. Results and Discussion

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### 3.1 Compositional and Morphological Characterization

311 Infrared spectroscopy was employed to investigate the structure of the plasma polymer at different applied RF power for a given flow rate of HMDSO and at 312 313 different flow rates of HMDSO at fixed applied power. Fig. 2 shows FTIR spectra 314 to identify different kinds of bonding in plasma polymerized HMDSO films 315 according to literature (Despax and Raynaud 2010). The absorption band with the highest intensity is present at 1081 cm<sup>-1</sup> which represents Si-O-Si asymmetric 316 317 stretching vibration and may also be overlapped with CH2 wagging band in the range 1040-1060 cm<sup>-1</sup>. This strong wagging band due to CH<sub>2</sub> is usually 318 319 accompanied with CH2 scissor vibrations which are represented by a small peak at 320 1357 cm<sup>-1</sup> associated with Si-CH<sub>2</sub>-Si. This small peak originating due to Si-CH<sub>2</sub>-321 Si suggests that Si-CH<sub>2</sub>-Si bridge building can be one of the mechanisms of plasma polymerization in the adopted deposition scheme (Rau and Kulisch 1994). 322 The peak at 1460 cm<sup>-1</sup> and at 1406 cm<sup>-1</sup> are representative of CH<sub>3</sub> asymmetrical 323 and symmetrical bending respectively and also points to the polymeric structure 324 of the thin film. The absorption band present at 1700 cm<sup>-1</sup> indicates presence of 325 oxygen in C=O whereas a small peak at 1620 cm<sup>-1</sup> represents contribution from 326 C=C (Hanus et al. 2008). The absorption band at 2145 cm<sup>-1</sup> represents Si-H bond 327 (Rau and Kulisch 1994). The peak at 1260 cm<sup>-1</sup> is due to Si-CH<sub>3</sub> derived from Si-328 329 (CH<sub>3</sub>)<sub>2</sub> groups (Radeva et al. 2014). The peaks at 2966 cm<sup>-1</sup> and 2902 cm<sup>-1</sup> are 330 representative of asymmetric and symmetric stretching vibrations of CH2 in the Si-CH<sub>2</sub>-Si fragments. The absorption peak at 850 cm<sup>-1</sup> can be associated with Si-331 332 CH<sub>3</sub> stretching vibration originating from Si-(CH<sub>3</sub>)<sub>3</sub> end groups whereas a small 333 shoulder peak in its vicinity appearing at 807 cm<sup>-1</sup> can be assigned to stretching 334 vibrations of the Si-CH<sub>3</sub> bond derived from Si-(CH<sub>3</sub>)<sub>2</sub> and Si-CH<sub>3</sub> groups 335 (Radeva et al. 2014). When power is increased from 10 to 25 and 50 W for a 336 fixed monomer (HMDSO) flow of 0.2 sccm and constant Ar flow of 10 sccm, the intensity of the peak at 850 cm<sup>-1</sup> reduced suggesting the removal of CH<sub>3</sub> groups, 337 338 during plasma polymerization (Radeva et al. 2014). Since RF plasma at 25 W was found to be more stable than at 50 W, therefore, 25 W was selected for deposition 339 340 of plasma polymerized HMDSO films along with co-sputtering of silver for the 341 rest of the experiments. These results indicate successful plasma polymerization

and deposition of SiC<sub>x</sub>O<sub>y</sub>H films on the substrates. Moreover, the structure of the plasma polymer can be varied by the applied RF power.

The structure of the plasma polymer films can be varied either by changing the power (as discussed above) or by changing the monomer (HMDSO) flow rate. This can be seen from FTIR spectra of plasma polymer films obtained at increased HMDSO flow rate (0.4 and 0.5 sccm) at 25 W. Films obtained at

increased HMDSO flow again exhibit an increased intensity at 850 cm<sup>-1</sup>

suggesting higher number of CH<sub>3</sub> groups due to increased monomer flow rate.

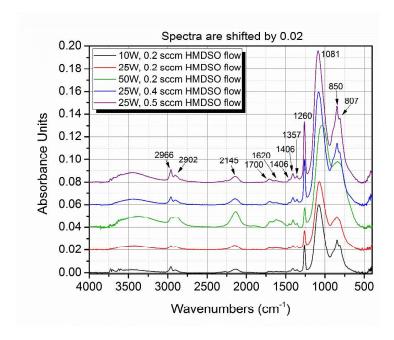
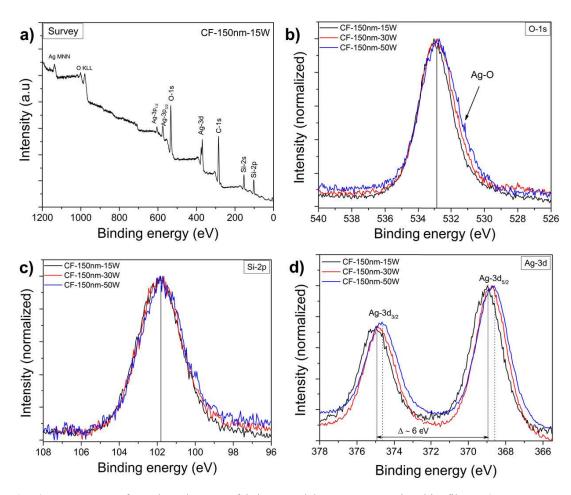


Fig. 2 FTIR spectra of plasma polymer thin films deposited onto gold-coated silicon wafers at different RF power and HMDSO flow rates. Ar flow was fixed at 10 sccm

Further chemical characterization of the composite thin film coatings comprising silver nanoparticles embedded in the plasma polymer matrix and deposited onto cotton fabric was performed using XPS. Survey spectrum (not shown here) of the uncoated fabric surface indicated the presence of carbon and oxygen as these are the main constituents elements of cellulosic fibers. Additionally, silver and silicon peaks appeared in the survey spectrum of the cotton fabric coated by nanocomposite thin film (Fig. 3a). For the detailed interpretation of XPS results only three samples with constant film thickness and different Ag content were selected. High resolution XPS spectra of O-1s, Si-2p and Ag-3d are shown in Fig. 3 b, c and d, respectively. Their detailed analysis with respect to chemical shifts yields an important information about the bonding between different elements present on the surface and, thus give the information about the chemical structure

of nanocomposite thin films. All XPS spectra were charge referenced for aliphatic 366 367 carbon at 285.0 eV. The O-1s peak located at around 532.8 eV represents O-Si bonds (Brunon et al. 2011) originating from organosilicon like structure 368 369 (Alexander et al. 1996). Generally, high resolution O-1s XPS peaks for the three 370 selected samples are almost identical, except only a small shift of the peak 371 shoulder located at lower binding energy (~531.0 eV). More specific, this 372 component of the peak is increasing with the increasing silver concentration in the 373 nanocomposite films. Such peak can be attributed to Ag-oxide groups which are 374 formed either due to formation of Ag-O bonds in the polymeric films or due to 375 surface oxidation after the exposer to the atmosphere. 376 High-resolution Si-2p peak, located at 101.8 eV can be assigned to Si atoms 377 bonded to oxygen (correlation with O-1s interpretation, see above) as well as 378 different hydrocarbons (FTIR spectra confirm similar observations). According to 379 the literature, components of Si-2p peak located at 101.5, 102.1 and 102.8 eV 380 have been reported to be originating from R<sub>3</sub>SiO, R<sub>2</sub>SiO<sub>2</sub> and RSiO<sub>3</sub> respectively 381 (R being hydrocarbon) (Saulou et al. 2012). As one can notice from Figure 3c all 382 three Si-2p spectra for different samples are identical, suggesting that there is no 383 influence of silver concentration on the chemical structure of the plasma 384 polymerized HMDSO matrix. 385 Fig. 3d shows a comparison of high resolution of Ag-3d for nanocomposite thin 386 films containing different amount of Ag inclusions. For the sample with the 387 lowest Ag concentration (CF-150nm-15W) Ag-3d<sub>5/2</sub> peak positioned at 368.8 eV 388 and Ag-3d<sub>3/2</sub> peak at 374.8 eV with spin orbit separation of 6 eV suggest silver in 389 a metallic form (Ag<sup>0</sup>) (Deng et al. 2014). However, the comparison of Ag-3d 390 peaks for samples with higher Ag amount shows a shift of the Ag peaks to lower 391 binding energy by roughly 0.25-0.35 eV that might be an indication of a slight 392 silver oxidation caused, most likely, by the exposure to the ambient atmosphere 393 (Moulder and Chastain 1992). 394 Elemental atomic concentrations obtained from XPS measurements are also 395 shown in the Table 2. The key information from the table is to confirm the 396 presence and show variation of silver concentration on the surface of samples 397 depending upon the deposition conditions. The surface silver concentration increases with increase in silver sputtering power for a given coating thickness 398 399 and decreases with increase in coating thickness at a given silver sputtering

power. The surface atomic percent of silver is 2.7, 4.9 and 8.7 % when sputtered at 15 W, 30 W and 50 W respectively. When coating thickness was increased to 300 and 400 nm by increasing the HMDSO flow while maintaining the silver sputtering power at 50 W, the silver atomic percent on the surface decreases and comes closer to what was observed at 30 W (for 150 nm) and 15 W (for 150 nm). This is also in agreement of decreasing silver particle size (TEM micrographs) due to greater dispersion of the silver within the bulk of the thicker matrix. Thus it can be concluded that plasma polymerization of HMDSO monomer resulted in an organosilicon like coating containing silver nanoparticles with surface silver concentration that increased or decreased depending upon the deposition conditions hence affecting the distribution of silver within the polymer matrix.



**Fig. 3** XPS spectra for selected cotton fabric coated by nanocomposite thin films: a) Survey spectrum for CF-150nm-15W; b), c) and d) high resolution XPS spectra of O-1s, Si-2p and Ag-3d peaks, respectively, for three selected samples with constant film thickness and various silver amount. All the high resolution spectra are normalized

Table 2 Elemental atomic percent from XPS for uncoated and coated cotton fabric

	Sample	C	0	Si	Ag
	CF-uncoated	76.5	23.4	-	-
Fixed coating	CF-150nm-	53.3	24.0	20.0	2.7
thickness and	15W				
Increasing silver	CF-150nm-	56.7	21.8	16.6	4.9
sputtering	30W	2017			
power	CF-150nm-	55.8	20.6	14.9	8.7
	50W				
Increasing	CF-150nm-		20.5	14.9	8.7
coating	50W				
thickness and	CF-300nm-	50.4	23.9	21.1	4.6
fixed silver	50W				
sputtering	CF-400nm-		24.0	21.6	3.2
power	50W				

The morphology of the composite coatings deposited onto cotton fabric can be seen in SEM images shown in Fig. 4. The surface of the uncoated cotton fiber was relatively smooth showing original features of the fiber surface. After deposition of nanocomposite thin films, dense granular topography, typical for sputter deposited coatings (Irfan et al. 2017), can be seen on the surface of the fibers indicating presence of the coating. However, silver nanoparticles are not visible on the fiber surface. This is expected as silver nanoparticles are supposed to be embedded within the plasma polymer matrix according to the adopted deposition scheme. The SEM images show uniform and conformal coverage even on uneven fiber surfaces. Further, EDX analysis (not shown here) revealed the presence of the elements Ag, Si along with C and O. This confirms the presence of silver nanoparticles and organosilicon like polymer on coated cotton fibers.

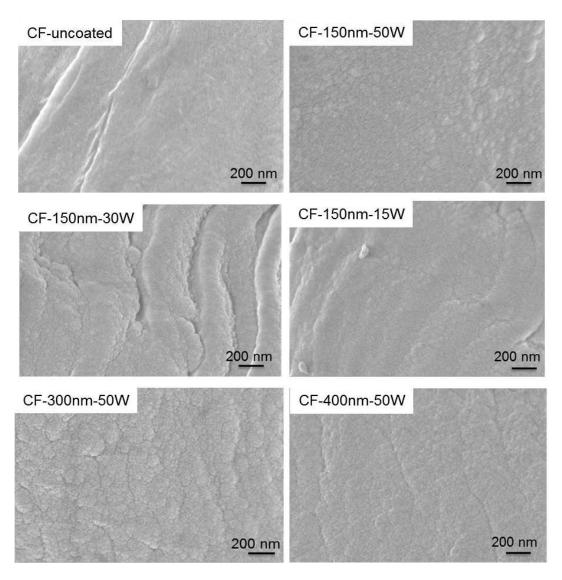


Fig. 4 High magnification SEM images of cotton fibers coated with nanocomposite thin films comprising silver nanoparticles embedded in plasma polymer matrix

TEM was performed to analyze the size of the silver nanoparticles and their distribution in the polymer matrix obtained under different deposition conditions and results are shown in Fig. 5. The size distribution of the silver nanoparticles was determined using image processing software ImageJ. The figure shows that spherical silver nanoparticles were homogeneously distributed within the amorphous plasma polymer matrix. However, some non spherical particles can also be seen at increased deposition power of silver (150nm-50W and 150nm-30W) which were formed due to coalescence of smaller particles within the growing polymer matrix as result of higher deposition rate of silver. In addition, under higher deposition powers of silver, few particles as large in diameter as 34 or 44 nm can also be observed in TEM micrograph shown in Fig. 5. At lower deposition power (150nm-15W), a bimodal size distribution was observed with

majority of the silver particles of 4 nm followed by 8 nm in size. With increase in 448 449 silver sputtering power (150nm-30W and 150nm-50W) while keeping the 450 deposition of the plasma polymer the same, an increase in the diameter of the 451 nanoparticles can be observed from the size distribution histograms in Fig. 5. This 452 results in normal distribution with maximum in size distribution at around ~6 nm 453 with increase in silver sputtering power (150nm-50W). 454 When thickness of the plasma polymer matrix was increased by increasing the 455 flow rate of the HMDSO monomer while keeping the silver sputtering power 456 fixed at 50 W (300nm-50W), bimodal size distribution for silver nanoparticles 457 was observed again. Since the nucleation of the metal nanoparticles and their 458 growth is controlled by their surface diffusion coefficients in the plasma matrix 459 (Drábik et al 2015), the thicker plasma matrix may limit the diffusion of silver 460 atoms reducing the size of the silver nanoparticles leading again to bimodal size 461 distribution even at higher silver sputtering power. Thus, at thickness of 300 nm, 462 the average diameter of the majority of the silver nanoparticles was around 4 nm 463 followed by 8 nm in diameter. With further increase in matrix thickness to 400 nm 464 (400nm-50W), a further increase in the number of silver particles of ~4 nm diameter occurred followed by an increase in number of even smaller 465 466 nanoparticles of ~2 nm. An equivalent decrease in the number of nanoparticles 467 with higher diameter of about 8 nm is also evident. Because of this reason, the 468 average diameter of silver nano particles, reported in Fig. 5, decreased with 469 decrease in silver sputtering power while it also decreased even to greater extent 470 with increase in coating matrix thickness. In all the cases, the size of the majority 471 silver nanoparticles varied between 2 to 14 nm which is similar to what has been 472 reported earlier for the nano composite coating obtained under a different plasma 473 configuration (Peter et al 2011). The inset images in Fig. 5 show the selected area 474 electron diffraction (SAED) pattern of the composite films. Well ordered 475 diffraction rings visible as bright dots indicate the presence of crystalline silver 476 nanoparticles (Drábik et al. 2015). These diffraction rings are more visible for the 477 composite coating containing higher silver concentration and larger nanoparticles 478 (150nm-50W, 150nm-30W) than those with lower silver concentration and 479 smaller particles (150nm-15W, 300nm-50W, 400nm-50W). The variation in the 480 size of the silver nanoparticles and their distribution was also reflected in the corresponding UV-Vis spectra of the composite coating obtained under different conditions as discussed below.

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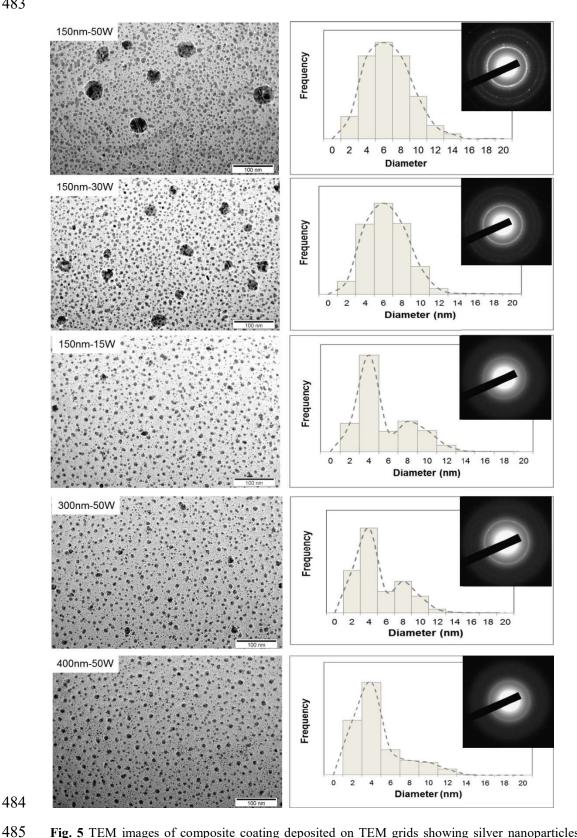


Fig. 5 TEM images of composite coating deposited on TEM grids showing silver nanoparticles embedded within the plasma polymer matrix under different deposition conditions. Images were processed using ImageJ to produce size histograms

#### 3.2 Optical Properties

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Optical properties of the composite coatings were evaluated with UV-Vis spectroscopy and results are shown in Fig. 6 a,b. The transmittance spectra were acquired for different nanocomposite thin films deposited onto quartz glass substrates. UV-Vis spectra are combined into two groups: 1) fixed total thickness of nanocomposite thin films with different Ag filling factor (Fig. 6a) and 2) constant Ag amount and varied thin film thickness (Fig. 6b). In both cases absorbance peaks with different intensities can be seen depending upon the total silver concentration as well as silver nanoparticle size distribution in accordance with TEM results. The spectrum of the quartz glass substrate coated with 150 nm thick plasma polymer only is shown for comparison purposes. The coating with minimum silver concentration resulted in maximum transmission. At lower deposition power (150nm-15W), silver nanoparticles have bimodal size distribution as discussed in the TEM analysis and hence exhibit two absorption peaks at around 345 and 420 nm. Traditionally, the optical absorption of metal nanoparticles is ascribed to be due to Localized Surface Plasmon Resonance (LSPR) (Wiley et al. 2006). However, due to intraband excitation of the conduction electrons by incident photons, it can also be described quantum mechanically. The maximum absorbance wavelength, according to quantum theory of nanoparticles, is linked with the conduction band energy (Gharibshahi et al. 2017). Contrary to the bulk metals, conduction electrons are not completely free in metal nanoparticles rather some of them are linked with individual atoms while others are free to move. When photon of light hit the metal nanoparticles, these conduction electron get intra band excitations. Smaller sized particles are composed of fewer metal atoms which reduces the potential attraction between metal ions and conduction electrons in the nanoparticle. This leads to increased conduction band energy for smaller sized nanoparticles. On the other hand, larger particles are composed of greater number of metal atoms leading to increased attraction between conduction electrons and metal ions. Thus conduction band energy of larger sized particles is reduced (Gharibshahi et al. 2017). It has been reported that smaller sized silver particles (up to 2nm) can exhibit more than one absorption peaks due to molecule-like optical transitions. As their size grows (>2nm), the absorption band is influenced by surface plasmon resonance of free electrons in the particles (Bakr et al. 2009).

With the increase in silver deposition power (150nm-30W and 150nm-50W), silver nanoparticle size and concentration increased, therefore, the absorption intensity increased significantly and shifted from 420 nm to 440 nm whereas the absorption peak at 345 nm vanished giving rise to a broad single peak (Fig. 6a). Increased absorption intensity and red shift suggests that at higher silver sputtering power we have larger silver nanoparticles with decreased nanoparticle to nanoparticle distance due higher density of silver nanoparticles in agreement with TEM observations. However, when matrix thickness of the coating was increased while keeping silver sputtering power constant at 50 W, the silver nanoparticles again showed bimodal size distribution due to decrease in silver nanoparticles size. The increase in matrix thickness from 150 nm to 300 and 400 nm for silver sputtering power of 50 W (300nm-50W and 400nm-50W) brought the overall silver filling factor approximately to the same level as that obtained for 30 W (for 300nm-50W) and 15 W (for 400nm-50W) although the total silver concentration within the thin films was higher at higher power (at 50W). Therefore, the absorption peaks both at 345 nm and 420 nm appeared again (300nm-50W and 400nm-50W in Fig. 6b). However, the transmittance level of the coatings with similar silver filling factors but different total silver amount in the nanocomposite (for example 150nm-15W and 400nm-50 W) was different. The coating with higher silver concentration can absorb more than that with lower silver concentration (Brunon et al. 2011). This implies that the transparency of the nanocomposite film can be increased to certain extent by increasing the matrix thickness while maintaining higher silver concentration within the thin film. However, it cannot be brought to the level of the thin film having lower silver concentration although both thin films may have same silver filling factor with respect to total volume of the thin film. The images of cotton fabric coated with nanocomposite thin films are shown in Fig. 7 which demonstrate variation in the color of the fabric under different deposition conditions. Maximum preservation of the original look of the fabric can be seen for fabric sample CF-150nm-15W which is deposited with minimum silver concentration. Whereas maximum color variation occurred for fabric sample CF-150nm-50W that contains maximum silver concentration deposited at 50W. However, this variation in color with maximum silver concentration was restored to reasonable extent for samples CF-

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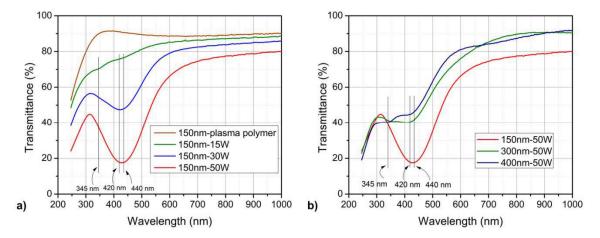
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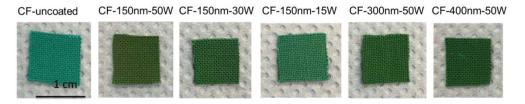
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300nm-50W and CF-400nm-50W which contain same silver concentration dispersed in a thicker matrix.



**Fig. 6** Transmittance UV-Vis spectra of the Ag/ppHMDSO nanocomposite coatings deposited onto quartz glass: a) Constant thickness of nanocomposite thin film with different silver amount; b) Varied thickness of the films (150 nm, 300 nm and 400nm) with constant deposition rate of silver



**Fig. 7** Photographs of fabric samples coated with nanocomposite thin films prepared under various deposition conditions

### 3.3 Silver Concentration and Silver Ions Release

The total amount of silver deposited onto cotton fabric under three silver sputtering powers, namely 15 W, 30 W and 50 W (CF-150nm-15W, CF-150nm-15W and CF-150nm-15W), was determined via ICP-MS to be 55, 129 and 286 ppm (mg of silver per kg of fabric) respectively. For other two samples (CF-300nm-50W and CF-400nm-50W), the silver sputtering power was maintained at 50 W similar to CF-150nm-50W, therefore, these samples are also expected to contain same amount of silver (286 ppm) as that of sample CF-150nm-50W. However, in samples CF-300nm-50W and CF-400nm-50W, this silver amount is dispersed in a thicker matrix. It is of interest to evaluate the potential of these nanocomposite coatings to release silver ions in wet environment.

In aqueous environments, silver nanoparticles are oxidized and release silver cations (Ag<sup>+</sup>). Antibacterial activity of silver nanoparticles is considered to be associated with release of silver ions from the coating (Körner et al. 2010). Thus, a controlled and sustained release of silver ions from the coating may contribute to sustained and prolonged antibacterial activity. Therefore, silver ion release behaviour of the deposited antibacterial coatings was evaluated by immersing the coated fabric samples in water and results are shown in Fig. 8. The Figure shows silver ion release profiles from the coated fabric dependent on immersion time, silver concentration as well as coating thickness. It is worth mentioning that no abrupt release was observed in any case after immersing the sample in water. This is the benefit of embedding silver nanoparticles in the matrix which controls the release of silver ions into the aqueous environment. Direct exposure of the silver nanoparticles to the water may cause an abrupt release of silver ions in water (Kuzminova et al. 2016). The graph shows that higher the silver concentration in a given coating thickness (Fig. 8a) (obtained at higher silver sputtering power), higher is the release of silver ions. On the other hand, increasing the coating thickness while keeping the silver concentration per unit fabric area the same (Fig. 8b, samples CF-150nm-50W, CF-300nm-50W and CF-400nm-50W) results in decrease in the release of silver ions. This implies that increased thickness of the matrix reduced the kinetics of silver ions release. It should be noted that all the samples in Fig. 8b were deposited at same silver sputtering power (50W) thus containing approximately same concentration of silver on unit fabric surface area but dispersed in the coating matrix of varying thickness. This concentration was measured to be 286 ppm i.e mg of silver per kg of cotton fabric via ICP-MS. The cumulative release of Ag<sup>+</sup> both from CF-150nm-50W and CF-150nm-30W was more than 90% of the total deposited silver within three days of immersion in water. Whereas, it was 30, 23 and 15% from CF-150nm-15W, CF-300nm-50W and CF-400nm-50W respectively. The release of silver ions from (CF-150nm-30W) is higher than that from CF-300nm-50W and CF-400nm-50W despite the fact that concentration of silver deposited at 30 W is less than that deposited at 50 W. This indicates that higher thickness of the coating (matrix) suppressed the release of silver ions even if the concentration of silver per unit fabric area was higher than that in relatively thinner coating. This justifies the hypothesis that

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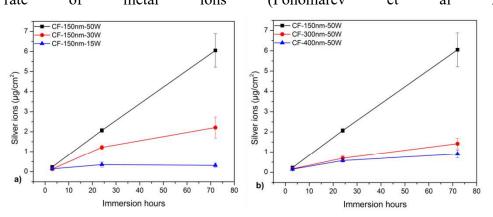
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dispersing relatively higher silver concentration in thicker coating will provide more sustained supply of silver ions extending the life of the coated fabric in repeated use along with benefit of reduced coloration (due to smaller particle size) as mentioned earlier. This is of particular importance keeping in view that the controlled release of metal ions is considered very difficult to achieve as it is not only dependent on the concentration of metal particles. Hydrophobic nature of the matrix (discussed later in the article) is also responsible in supressing the release of silver ions from the thicker matrices as has also been reported by Kylian et al. (Kylian et al. 2017). Other factors like surface roughness, surface oxidation kinetics and particle agglomeration state are also key factors to control the release rate of metal ions (Ponomarev al 2018). et



**Fig. 8** Silver ion release profiles for cotton fabric coated with nanocomposite thin films. The measurements were carried via ICP-MS on water samples drawn after 3, 24 and 72 hours of immersion in water

#### 3.4 Antibacterial performance

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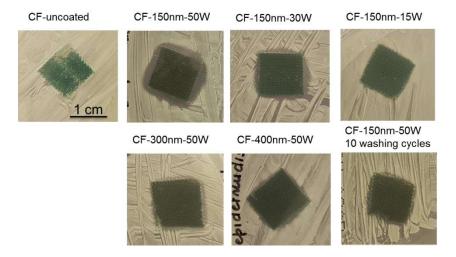
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Silver nano particles are well known antimicrobial agents which demonstrate biocidal action against a wide variety of bacteria, both Gram positive and Gram negative, as well as fungi (Balagna et al. 2012) along with low cytotoxicity and absence of drug resistance (Wu et al. 2018). The exact mechanism for biocidal action of silver nano particles, as well as other metal nano particles, is not yet completely understood. However, experimental evidence suggests that antimicrobial activity of silver nano particles is associated with silver ions and hence on the kinetic of silver ions release from the silver nano particles (Ponomarev et al. 2018). In this study, antibacterial activity of the prepared coatings was evaluated against *Staphylococcus epidermidis* and results are shown in Fig. 9. The Figure shows that, compared with the untreated control sample (CF-uncoated), where bacterial growth is well visible under fabric surface, all the

coated samples showed antibacterial activity against S. epidermidis. However, the 640 641 size of the inhibition halo is different for different samples. 642 In case of higher silver ion release from the coating (CF-150nm-50W and CF-643 150nm-30W), a well-defined inhibition halo can be seen around the samples. This 644 indicates higher concentration of the active agent leached from the sample into the 645 surrounding medium. As shown in Fig. 8, the minimum concentration of silver 646 ions was released from CF-150nm-15W which also contains minimum silver 647 concentration as compared with other samples. It can be observed that there 648 wasn't bacterial growth under the fabric sample despite the absence of a well-649 defined inhibition halo. For the samples CF-300nm-50W and CF-400-50W, which 650 contain the same concentration of silver as that of CF-150nm-50W, but dispersed 651 in thicker matrices, a very small inhibition halo can be observed. This is due to a 652 more controlled release of silver ions from the thicker coating, CF-300nm-50W 653 and CF-400-50W, than from the thinner CF-150nm-50W. The formation of the 654 inhibition halo depends on the leaching of the antimicrobial agent in the 655 surrounding medium (Tomšič et al. 2008). This fact decreases the antibacterial 656 agent concentration on the fabric surface and, consequently, diminishes the 657 subsequent antimicrobial performances. Therefore, controlled release of silver 658 ions will be more sustainable to maintain antibacterial activity for longer times. In 659 addition, compared with other metal ions, silver gives bactericidal action at very 660 low concentrations (Ponomarev et al. 2018). This property of silver nanoparticles 661 combined further with controlled release can be useful for many other 662 applications where cytotoxicity can be a concern while obtaining antibacterial 663 activity for example in wound dressings or where the silver nanoparticle 664 containing coating is in direct contact with human skin. 665 In order to assess the washing stability of the coating, antibacterial performance 666 was also evaluated on samples subjected to multiple washing cycles. For this 667 purpose, fabric samples CF-150nm-50W were washed 10 times and subjected to 668 the "Inhibition halo test". Fig. 9 shows that, although the size of the inhibition halo decreased after the washing treatment, the antibacterial activity was still 669 670 present, in fact no microbial growth, under the sample surface, was observed. The 671 washing stability of the thin films sputtered on textile surfaces has been found to 672 be poor as reported by Wang et el. (Wang et al. 2008). The results of this study show that reasonable washing stability for textiles deposited with sputtered thin films can be obtained.

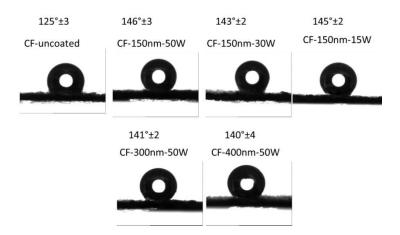




**Fig. 9** Inhibition halo test on cotton samples deposited with nanocomposite thin films with varying silver concentration and coating thickness. Antibacterial activity was present even after 10 washing cycles

# 3.5 Water wettability

Cotton fibers are hydrophilic due to abundance of hydroxyl groups in its structure that make cotton fabrics stain easily when in contact with liquids. Polysiloxanes are low surface energy polymers and thus make the surfaces hydrophobic on which they are applied. Fabric surfaces treated with polysiloxanes may result in high hydrophobicity to super hydrophobicity (Hao et al. 2016). Hydrophobicity of the cotton fabric coated by silver nanoparticles-plasma polymer thin film was assessed through water contact angle test and results are shown in Fig. 10. Uncoated cotton fabric showed a water contact angle of 125°. High water contact angle on uncoated cotton fabric was due to dense weave structure of the fabric which may lead to higher water contact angle due to surface patterning and roughness. After deposition, the water contact angle increased from 125° up to 146° making the fabric surface highly hydrophobic, closer to super hydrophobicity.



**Fig. 10** Water contact angle on cotton fabric coated with nanocomposite thin films showing highly hydrophobic surface after coating

#### Conclusion

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In summary, an antibacterial, highly hydrophobic and semi transparent nanocomposite thin film was deposited on green colored cotton fabric intended to be used in medical wear. The nanocomposite thin film, composed of silver nano particles embedded in the plasma polymer matrix, was obtained via ecofriendly plasma based co-deposition scheme. Polymer matrix was obtained by polymerizing HMDSO monomer and its subsequent deposition on the substrate. The coating was deposited under five different deposition conditions mainly to vary the silver concentration and its size distribution within the polymer matrix to influence optical properties as well controlling the leaching of silver ions. Silver concentration was varied either by decreasing silver deposition power without changing plasma polymer matrix thickness or by increasing plasma polymer matrix thickness without changing silver deposition power. Thickness of the plasma polymer was found to be more important in controlling the release of ionic silver in aqueous medium along with reduction in optical absorbance. While decreasing silver deposition power without changing polymer matrix resulted in more transparent coating due to lower silver concentration and smaller sized particles. Thus variation in the silver concentration or matrix thickness led to the variation in the silver nano particle size and their distribution within the coating matrix which consequently influenced the optical properties as well as silver ions release from the coating. The deposition resulted in bimodal distribution of silver nano particles where silver concentration was lower or matrix thickness was higher. The thin films deposited under all the five conditions demonstrated effective antibacterial activity against S. epidermidis LMG 10474 in the

- "Inhibition halo test". with size of the halo depending upon silver ion release
- 722 profiles from the respective coatings. The coating showed a certain degree of
- washing stability as it was able to retain antibacterial activity when subjected to
- 724 10 washing cycles. The coating rendered the surface of the cotton fabric high
- hydrophobicity reaching water contact angle as high as 146°. This work shows
- that independent manipulation of the process parameters for silver and plasma
- 727 polymer in the co-deposition scheme can yield nano composite coatings
- 728 comprising silver nano particles with acceptable transparency and long term
- antibacterial properties for practical applications.

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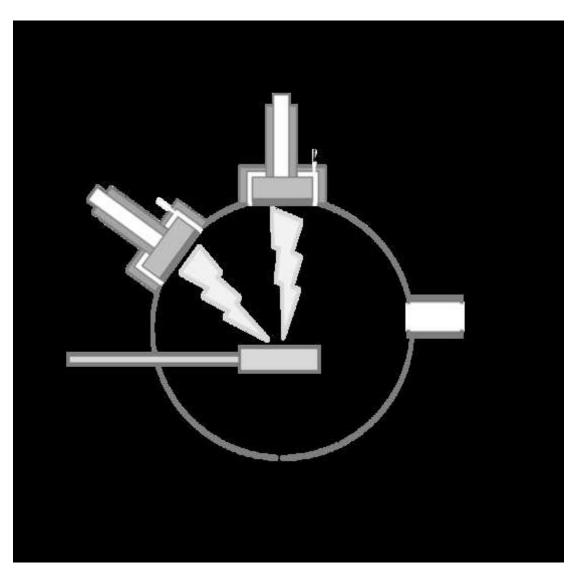
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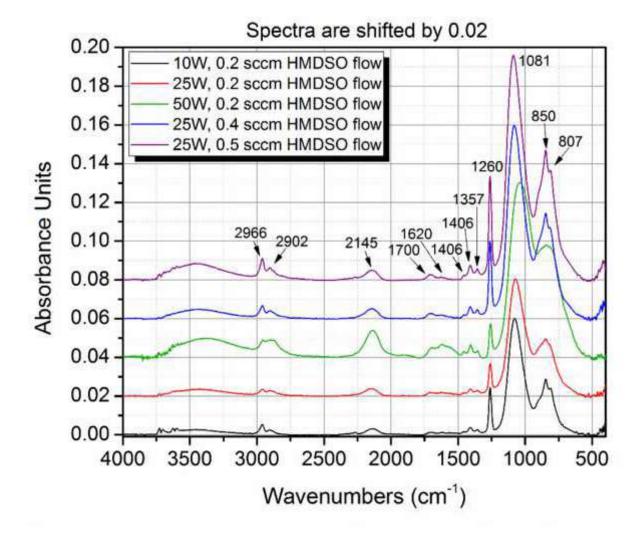
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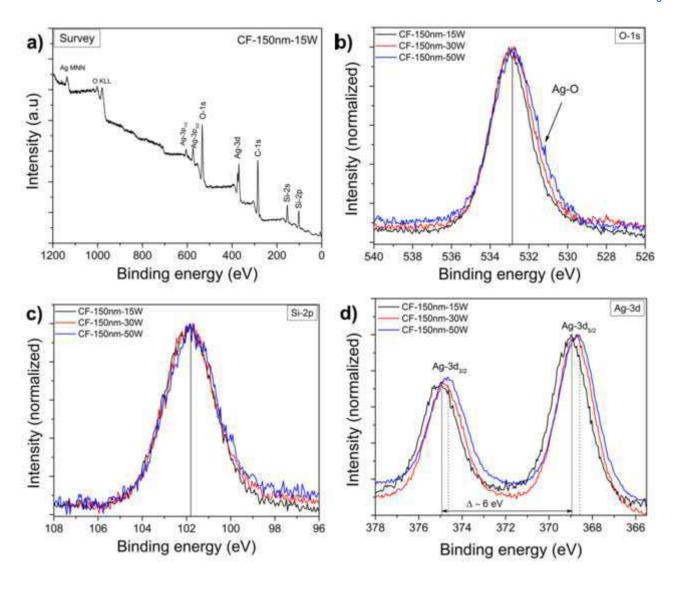
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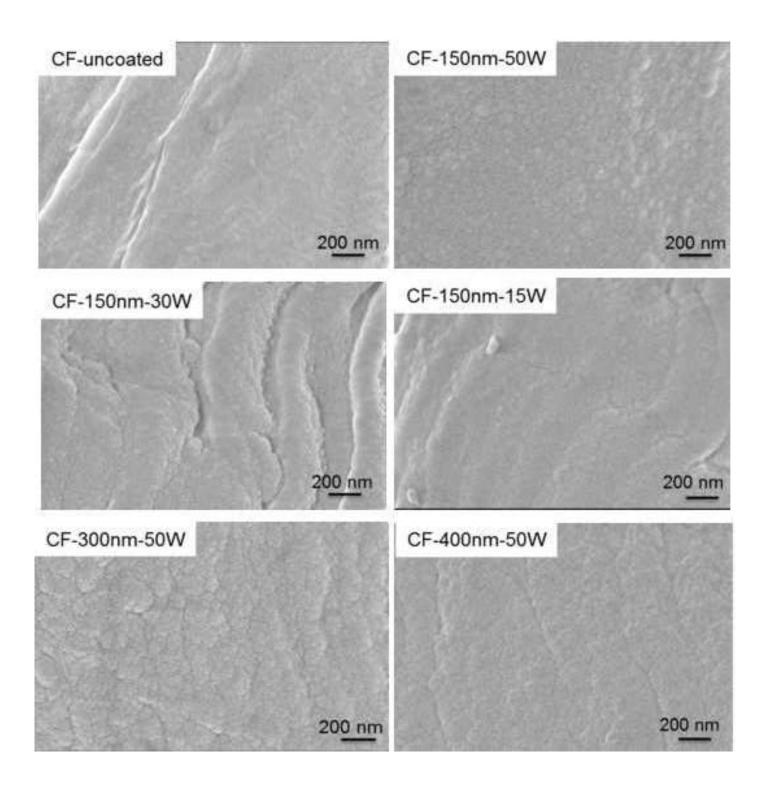
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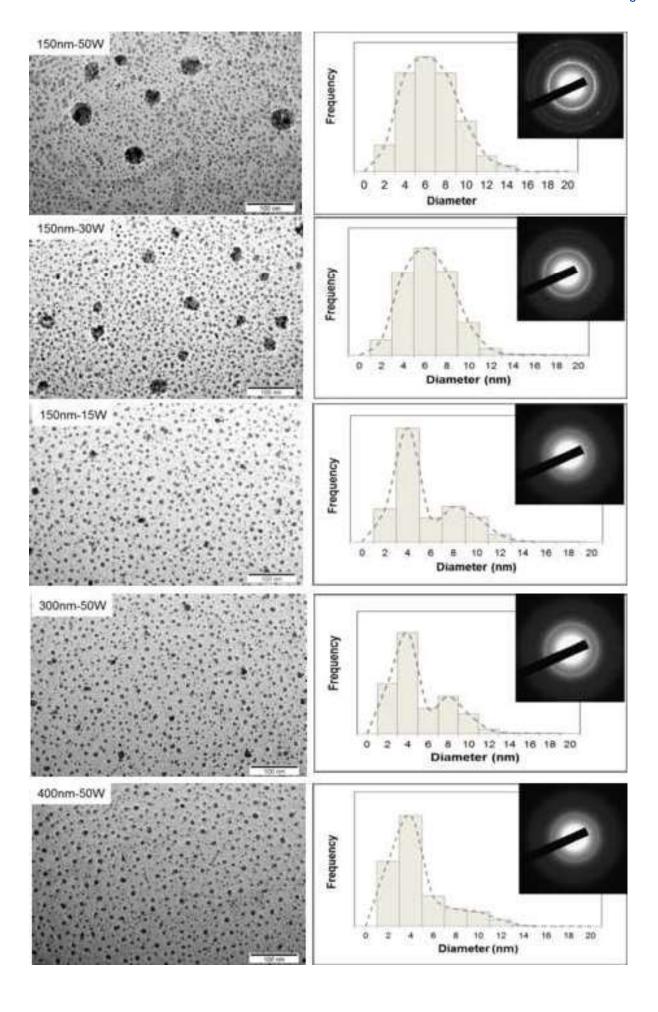
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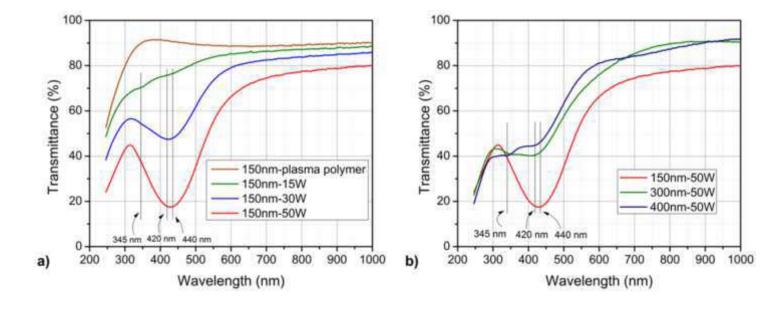












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