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Lithium Extraction from Shale Gas Flowback and

Produced Water Using H_{1.33}Mn_{1.67}O₄ Adsorbent

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Abstract: The rapid growth of demand for lithium, especially in lithium batteries calls for an increase in the supply of lithium resources. We synthesized H_{1,33}Mn_{1,67}O₄ adsorbent by solid-phase reaction method to recover lithium for the first time from shale gas flowback and produced water (SGFPW) in the Sichuan Basin, China. In order to improve the adsorption capacity and selectivity of lithium, Na₂CO₃ precipitation pre-treatment was employed to remove some divalent ions (Ca²⁺, Ba²⁺, Sr²⁺, Mg²⁺) and to increase the pH prior to lithium adsorption. The adsorption capacity of lithium in the pre-treated water (P-SGFPW) was 16.2 mg/g higher than that in SGFPW (13.3 mg/g), and the partition coefficient of Li⁺ increased from 731.6 to 1074mL/g after precipitation. After four cycles of adsorption and desorption, the adsorption capacity of lithium was stable and showed only a slight decrease equal to 3.53% and 5.35% in P-SGFPW and SGFPW, respectively. H_{1.33}Mn_{1.67}O₄ is a promising adsorbent for lithium extraction from SGFPW, and precipitation before adsorption can improve its performance significantly.

Keywords: Lithium recovery; Hydrogen manganese oxide (H_{1.33}Mn_{1.67}O₄); Adsorption; Shale gas flowback and produced water (SGFPW); Precipitation

1. Introduction

Lithium is regarded as "the energy metal of the 21st century" and "the white oil of the future" and may be as precious as gold (Tarascon, 2010). Lithium is one of the most electrochemically active metals, and it is a solid element with the highest redox potential, the largest specific heat capacity, the smallest density, the lightest weight, and the highest energy density at room temperature (Miatto et al., 2021; Swain, 2017; Zhang et al., 2021a). Therefore, lithium is widely used in various high-value products, such as batteries, ceramics, glass, in the nuclear industry, medicine, aerospace (Draaisma, 2019; Sun et al., 2021; U. S. Geological Survey, 2022; Zhang et al., 2021b). For example, electric vehicles and portable electronic devices deploy lithium batteries due to their high energy density, long service life, and relative environmental friendliness (Scrosati et al., 2011; Xie and Lu, 2020). The rapid development of the electric vehicle industry under the global "dual carbon" goal has led to a sharp increase in the demand for lithium batteries, with the proportion of lithium batteries in the global end-use markets increasing from 27% in 2011 to 74% in 2021 (U. S. Geological Survey, 2012, 2022). Global lithium consumption is estimated to be 93,000 tons in 2021, an increase of 33% from 70,000 tons in 2020, but lithium production (excluding U.S. production) only increased 21% (U.S. Geological Survey, 2022).

At present, lithium resources are mainly distributed in ores, seawater, and brines, e.g., oilfield brines, continental brines, geothermal brines (Swain, 2017). Extraction of

lithium from ores require cumbersome steps and has significant environmental impacts, including water pollution, soil pollution, and air pollution due to the use of copious quantities of chemicals, as well as the production of substantial amounts of acidic wastewater and solid waste (Hou et al., 2021; Yang et al., 2018). The total mass of lithium in seawater is abundant, but its concentration is very low (0.17 mg/L), making its selective extraction challenging from this source. Extraction of lithium from continental brines in China, Chile, and Argentina has been studied and widely applied, but oil and gas produced water is still an untapped lithium resource (Hou et al., 2021). Lithium concentrations in Marcellus shale gas flowback and produced water (SGFPW) ranges from 4 to 202 mg/L, and the median lithium concentration in SGFPW in the Sichuan Basin is 33 mg/L (Haluszczak et al., 2013; Xie et al., 2022). Horizontal drilling and hydraulic fracturing technologies have promoted the large-scale exploitation of shale gas, but they produce a large amount of FPW (Tang et al., 2022). It is estimated that the volume of SGFPW will reach 499-3585 million m³ globally and 50-73 million m³ in China by 2030 (Kondash et al., 2018; Liu et al., 2022). SGFPW is a potential resource for lithium recovery, and recovery of lithium from this wastewater as a secondary raw material would not only ease the imbalance between supply and demand, but also offset part of the cost of shale gas wastewater treatment by creating new value for this resource in the framework of circular economy (Kumar et al., 2019; Robbins et al., 2022). However, suitable and commercially feasible technologies for lithium recovery from SGFPW needs to be

improved.

Currently, there are several technologies for lithium recovery from solution, including precipitation (Zhang et al., 2019), extraction (Xiong et al., 2022), adsorption based on aluminum-based adsorbents (LiCl-2Al(OH)₃·nH₂O) (Zhong et al., 2021), manganese-based adsorbents (HMn₂O₄ (Han et al., 2020), H_{1,33}Mn_{1,67}O₄ (Seip et al., 2021), H_{1.6}Mn_{1.6}O₄ (Su et al., 2022)) and titanium-based adsorbents (H₂TiO₃ (Zhu et al., 2021), H₄Ti₅O₁₂ (Zhao et al., 2021)), as well as membrane-based methods, such as nanofiltration (Wang et al., 2021), electrodialysis (Chan et al., 2022), bipolar membranes electrodialysis (Bunani et al., 2017), membrane capacitive deionization (Shi et al., 2019), forward osmosis (Sutijan et al., 2022), electrochemical methods (Jang et al., 2021) and other hybrid techniques like reverse osmosis-electrodialysis (Qiu et al., 2019) or nanofiltration-membrane distillation (Pramanik et al., 2019)). Among these technologies, adsorption is one of the most promising methods for recovering lithium from oil and gas produced water due to minimal required pretreatment, low environmental impact, and high recovery efficiency (Safari et al., 2020).

Aluminum-based adsorbents are currently the only adsorbents used in industry, but their adsorption capacities are low (Sun et al., 2021). According to the research results discussed by Chen *et al.*, when the ratio of the adsorbent to brine is 1 g/30 mL, the equilibrium adsorption capacity of the aluminum-based adsorbent is only roughly 7.1 mg/L (Chen et al., 2020). Alternative options are H_{1.6}Mn_{1.6}O₄ and H₄Ti₅O₁₂, which

however cannot be directly synthesized by solid phase reaction method, and intermediate products LiMnO₂ and LiTi₂O₃ need to be synthesized first, respectively. The temperature required for the synthesis of HMn₂O₄ and HTi₂O₃ by the solid phase reaction method is high, as these materials are typically calcined at 800 °C for 5 h (Ji et al., 2017) and 700 °C for 4 h (Lawagon et al., 2019), respectively, while H_{1.33}Mn_{1.67}O₄ only needs 4 h at 500 °C (Tang et al., 2020). In fact, H_{1.33}Mn_{1.67}O₄ is by far the most convenient adsorbent to synthesize among those discussed above. Moreover, due to its high chemical stability, high adsorption capacity, high selectivity, low toxicity, and low cost (Xu et al., 2019), H_{1.33}Mn_{1.67}O₄ is surely one of the most promising adsorbents for lithium recovery from SGFPW.

This is the first study of lithium recovery from SGFPW in the Sichuan Basin, China. In this study, we synthesize Li_{1.33}Mn_{1.67}O₄ precursor by solid phase reaction method and obtain the H_{1.33}Mn_{1.67}O₄ adsorbent by pickling with HCl. Then, we evaluate the feasibility of H_{1.33}Mn_{1.67}O₄ applied to lithium recovery from SGFPW by adsorption kinetics, adsorption isotherm, adsorption selectivity, and reusability. We also investigate the effect of pH values and pre-treatment by Na₂CO₃ precipitation on lithium recovery from the SGFPW. The results are supported by in-depth determination of physical and chemical properties of the H_{1.33}Mn_{1.67}O₄ adsorbent.

2. Materials and methods

2.1. Materials

The SGFPW used in the experiment was collected from a shale gas well, which

is located in Gongxian, Sichuan Basin, China, in the Lower Silurian Longmaxi Formation, as described in our previous work (Tian et al., 2020). The collected SGFPW samples were stored in plastic buckets under dark conditions before experiments. Lithium carbonate (Li_2CO_3), manganese carbonate (MnCO₃), sodium carbonate (Na₂CO₃), hydrochloric acid (HCl), sodium hydroxide (NaOH) were purchased from Kelong Chemical (Chengdu, China). All chemicals used in the experiments were of analytical grade. Ultrapure water with a resistivity of 18.25 M Ω was obtained from an ULURURE ultrapure water machine (Chengdu, China).

2.2. Preparation of the $H_{1,33}Mn_{1,67}O_4$ adsorbents

Li₂CO₃ and MnCO₃ were mixed and ground uniformly at a Li/Mn mole ratio of 1.33:1.67. The mixture was placed in a ceramic boat and calcined at 500 °C for 4 h in a muffle furnace in ambient air at a ramping rate of 3 °C/min, and then cooled to room temperature. The product obtained was Li_{1.33}Mn_{1.67}O₄ (LMO). In order to obtain the H_{1.33}Mn_{1.67}O₄ (HMO) adsorbent, 1 g LMO was added to 1 L of 0.5 mol/L HCl and stirred at 200 rpm for 24 hours for delithiation. Finally, the adsorbents were washed with ultrapure water, filtered, and then dried at 50 °C for 24 h.

2.3. Characterization of H_{1,33}Mn_{1,67}O₄ adsorbents

The surface morphology and element electron binding energy were measured by field-emission scanning electron microscopy (FE-SEM) (Regulus-8230, Hitachi, Japan) and X-ray photoelectron spectroscopy (XPS) (AXIS Supra, Kratos, U.K.), respectively. Powder X-ray diffraction (XRD) patterns were recorded on a DX-2700

(Dandong Haoyuan, China) instrument equipped with a Cu K α radiation (k = 0.15418 nm) in the 2 θ range of 10°-80°; the step size of 0.05° and accumulation time of 0.5 s were adopted during scanning. N₂ adsorption-desorption isotherms were measured at the temperature of liquid N₂ (77 K) using an ASAP 2460 analyzer (Micromeritics, USA). The specific surface area was calculated with the Brunaue-Emmett-Teller (BET) equation. The pore size and distribution were determined with the Barrett-Joyner-Halenda (BJH) method.

2.4. Adsorption experiment of different ions

An appropriate amount of Na₂CO₃ with a molar ratio of 1.2 times the sum of divalent ions (Ca²⁺, Mg²⁺, Ba²⁺, Sr²⁺) was added to SGFPW to precipitate these components and eliminate their influence on Li adsorption. The mixture was stirred for 20 minutes at 200 rpm, and let stand for 20 minutes (Jang and Chung, 2018). Finally, the supernatant was drawn for subsequent adsorption experiments. The water quality parameters of SGFPW with or without precipitation pre-treatment are shown in Table S1 (Supporting Information, SI). Adsorption experiments were conducted in both the SGFPW and pre-treated SGFPW (P-SGFPW), and the ion concentration at each step was measured by Dionex Integrion HPIC (Thermo Fisher, USA) after filtration through a membrane with a 0.45 μm pore size.

2.4.1. The effect of pH on extraction of Li⁺

The pH values of the SGFPW were controlled in the range of 4 to 11 by 0.1 mol/L of HCl and NaOH. A mass of 0.05 g of the HMO adsorbent was added to 50

mL of SGFPW under different pH values, stirred at 200 rpm for 48 h, and the Li⁺ concentration in the supernatant was thus measured. The equilibrium adsorption capacity of Li⁺ was calculated according to Eq. (1)

$$q_e = \frac{\left(C_0 - C_e\right)V}{m} \tag{1}$$

where q_e (mg/g) is equilibrium adsorption capacity, V (L) is the volume of the solution, m (g) is the weight of HMO adsorbent, C_0 and C_e are the initial and equilibrium Li⁺ concentrations in solution (mg/L), respectively.

2.4.2. Adsorption isotherm of Li⁺

Different quantities of HMO adsorbents (0.015, 0.025, 0.05, 0.065, 0.075, 0.1 g) were added to 50 mL of the SGFPW or P-SGFPW. The mixture was stirred at 200 rpm for 48 h, and then the Li⁺ concentration of the supernatant was determined. The equilibrium adsorption capacity of Li⁺ was calculated based on Eq. (1). The Langmuir model and Freundlich model were applied to explore the adsorption mechanism, and details are described in Text S1 (SI).

2.4.3. Adsorption kinetics of Li⁺

A mass of 0.05 g of HMO adsorbent was dispersed in 50 mL of the SGFPW or P-SGFPW. The mixed liquid was stirred at 200 rpm for 72 h and the supernatant were taken at regular intervals to measure the concentration of Li⁺. The adsorbed amount of lithium at various times was calculated by Eq. (2)

$$q_t = \frac{\left(C_0 - C_t\right)V}{m} \tag{2}$$

where q_t (mg/g) and C_t (mg/L) represent the adsorbed amount and the Li⁺ concentration in solution at time t, respectively. V (L) is the volume of the solution, m (g) is the weight of HMO adsorbent and C_0 is the initial lithium concentration in solution. The pseudo-first-order kinetic model and pseudo-second-order kinetic model were used to investigate the adsorption kinetic behaviors, and the details of the two models are in Text S2 (SI).

2.4.4. Adsorption selectivity of Li⁺

In order to compare the adsorption behavior of lithium ions and other metal ions, $0.05~{\rm g}$ of adsorbent was added to 50 mL of SGFPW or P-SGFPW and stirred at 200 rpm for 48 h. Metal ion concentrations in the supernatant were then determined, and the partition coefficient (K_d , Eq. (3), concentration factor (CF, Eq. (4), and separation factors (α_{Me}^{Li} , Eq. (5) were calculated to investigate the adsorption selectivity of HMO

$$K_d = \frac{C_0 - C_e}{C_e} \times \frac{V}{m} \tag{3}$$

$$CF = \frac{q_e(Me)}{C_0(Me)} \tag{4}$$

$$\alpha_{Me}^{Li} = \frac{K_d \left(Li \right)}{K_d \left(Me \right)} \tag{5}$$

where Me represents Li⁺, Na⁺, K⁺, Mg²⁺, Ca²⁺, Ba²⁺, Sr²⁺. C_0 and C_e are the initial and equilibrium Me concentrations in solution (mg/L), respectively. V (L) is the volume of the solution and m (g) is the weight of HMO adsorbent. $q_e(Me)$ (mg/g) and $C_0(Me)$ (mg/L) represent the equilibrium adsorption capacity and initial concentration of Me in solution, respectively.

2.4.5. Reusability study of adsorbents

After each adsorption, the adsorbent was rinsed with ultrapure water, filtered, and dried. Then, it was regenerated with HCl for subsequent adsorption-desorption cycles. The adsorption-desorption cycle was performed four times.

3. Results and discussion

3.1. Characterization of HMO adsorbent

Fig. 1A shows the XRD patterns of the synthesized LMO precursor, its delithiated form HMO, and r-LMO (Li⁺-reinserted sample in the P-SGFPW). The XRD patterns of the three samples are almost identical, but there is a slight shift in the positions of the spinel peaks. The XRD pattern of LMO is nearly the same as the standard XRD pattern of Li_{1,33}Mn_{1,67}O₄ (PDF No. 46-0810, cubic phase, S.G: Fd3 m (227)), indicating that the LMO precursor was successfully synthesized. The LMO particles show three main diffraction peaks at $2\theta = 18.8^{\circ}$, 36.5° , 44.4° , which correspond to the (111), (311), (400) crystal planes of the spinel structure, with crystal face distances (d) of 4.716 Å, 2.460 Å, 2.039 Å, respectively. The main characteristic peaks of HMO were 19.1°, 37°, 45.05°, with crystal face distances (d) of 4.643 Å, 2.428 Å, 2.011 Å, respectively. r-LMO has three main diffraction peaks at $2\theta=18.95^{\circ}$, 36.7° and 44.7°, with crystal face distances (d) of 4.679 Å, 2.447 Å, 2.026 Å, respectively. Compared to LMO, the diffraction angle of the HMO was slightly higher, resulting in a decrease in the d-spacing values, indicating that the cell size was reduced, due to the replacement of Li⁺ by H⁺ with a smaller ion radius (Han et al.,

2012). After adsorption, the diffraction angle of r-LMO decreased again and was practically the same as LMO, indicating that the adsorbent had a stable structure and good cycle performance, and implying the effective ion exchange between H⁺ and Li⁺.

According to the IUPAC classification (Thommes et al., 2015), the nitrogen adsorption-desorption isotherms of the HMO adsorbent in Fig. 1B is a typical type IV isotherm with an H3 hysteresis loop, indicating that the adsorbent has mesopores. The curve had a distinct uptake of nitrogen adsorption at the relative pressure 0.8-1.0, due to the filling of the mesopores or macropores (Han et al., 2012). The capacity of N₂ adsorption of the HMO adsorbent was 104.7 cm³/g (STP). The pore size distribution was obtained by calculating the isotherm desorption branch by the BJH method, and the results are shown in Fig. 1C. A bimodal mesoporous structure composed of various mesopore sizes was found, where the small peak indicates a small number of mesopores and the large peak represents the major mesopores with an average pore size of 23-27 nm, which is consistent with the report by Lai et al. (Lai et al., 2020). The BET surface area was 37.8 m²/g according to the BET surface area plot (Fig. 1D), and the total volume of pores was 0.158 cm³/g. Fig. 2 shows the SEM images and EDS mapping results of Mn in LMO, HMO, r-LMO samples, indicating similar, hence stable, structures and a uniform distribution on Mn.

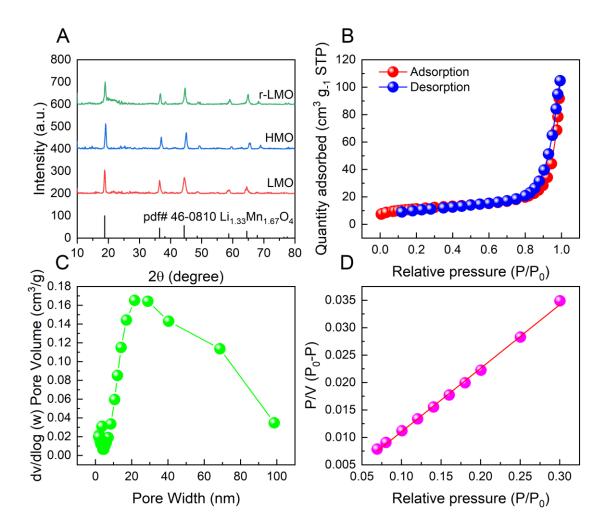


Fig. 1. (A) XRD patterns of the LMO, HMO, and r-LMO. (B) Nitrogen adsorption-desorption isotherms of HMO. (C) Pore size distributions of HMO. (D) BET surface area plot.

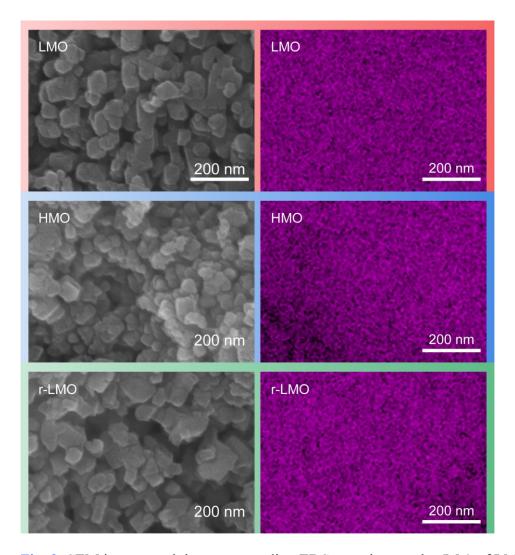


Fig. 2. SEM images and the corresponding EDS mapping results (Mn) of LMO, HMO, r-LMO.

The surface chemistry of LMO, HMO, and r-LMO was analyzed by using X-ray photoelectron spectroscopy. As shown in Fig. 3A, the peaks corresponding to Mn 2p and O 1s were evident, and the peaks corresponding to Mn 3p and Li 1s partially overlapped. Fig. 3B shows the fitting curves of Mn 2p, which were divided into two peaks, Mn 2p1/2 and Mn 2p3/2, with a spin energy separation of 11.2 eV. The O 1s spectra were divided into three peaks including O_{latt} (lattice oxygen), O_{ad} (adsorb

oxygen), and O_{H_2O} (chemisorbed oxygen species) (Luo et al., 2016), as depicted in Fig. 3C. The integral area of the three peaks had a substantial change during the pickling and adsorption process, indicating the change of the chemical environment of the oxygen atom (Qiu et al., 2021). The integral area of HMO was found to be smaller than that of LMO and R-LMO in the Fig. 3D, likely due to the process of ion exchange between H⁺ and Li⁺, indicating that Li⁺ could be successfully adsorbed and desorbed.

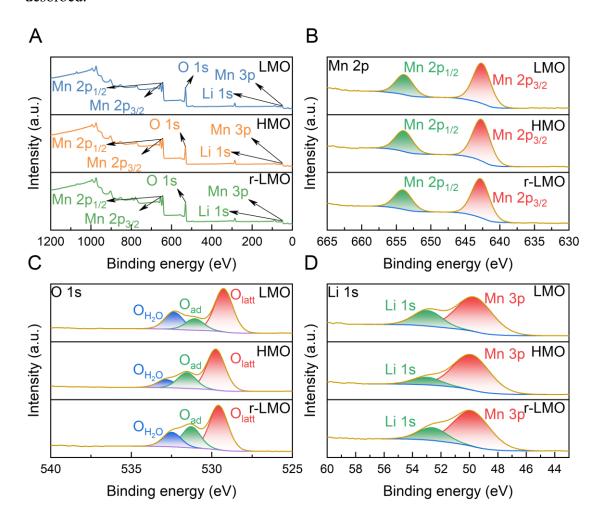


Fig. 3. Analysis of XPS of LMO, HMO, r-LMO. (A) Survey spectra. (B) High resolutions of Mn 2p. (C) High resolutions of O 1s (D) High resolutions of Li 1s.

3.2. Effect of pH on Li⁺ adsorption

Th pH values of the solution has a major influence on lithium adsorption capacity because H⁺ in acidic solutions will occupy the Li⁺ vacancies provided by the HMO adsorbent. The results of experiments assessing the adsorption capacity of lithium at different pH values are shown in Fig 4. The adsorption capacity of lithium increased monotonically with the increase of solution pH. With the increase of pH, the H⁺ activity decreases and the competition between H⁺ and Li⁺ becomes weaker (Qiu et al., 2021; Xu et al., 2019). Also, the surface sites of the HMO adsorbents has a larger density of negative charges as the pH increases (Xiao et al., 2015; Xu et al., 2019), exhibiting stronger affinity for lithium adsorption, which enhance the electrostatic attraction between Li⁺ and the adsorbent.

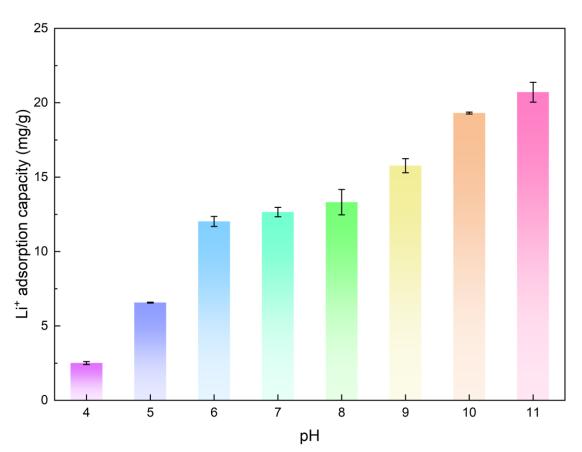


Fig. 4. Lithium adsorption capacity of HMO at different pH values.

3.3. Adsorption isotherms

Adsorption isotherm experiments were performed to investigate the maximum adsorption capacity and the optimal adsorbent dosage. The Langmuir model and Freundlich model are widely used to evaluate the behavior between adsorbates and adsorbents and to study the adsorption mechanism. Fig. 5A and Fig. 5B are the results of fitting the Langmuir model and the Freundlich model to the lithium adsorption isotherm data, respectively. The relevant adsorption isotherm fitting parameters calculated from the two models are summarized in Table 1. The regression coefficients (R²) of the Langmuir model were higher than those from the Freundlich model for both SGFPW and P-SGFPW, indicating that the Langmuir model described the adsorption isotherms slightly better. This result implies that the adsorption of lithium resembles more closely that of a monolayer chemisorption process that occurs at specific homogeneous sites on the surface of HMO adsorbent (Ryu et al., 2016) and that there was no significant competition between the solvent and sorbate to occupy the sorbent sites (Park et al., 2015). Furthermore, the K_L and the equilibrium adsorption capacity of lithium obtained with P-SGFPW was larger than those observed with SGFPW, suggesting that the precipitation process is advantageous for the adsorption of lithium. The increase in the adsorption capacity of lithium may also be partly due to the increase in pH during Na₂CO₃ precipitation process, from 7.97 to

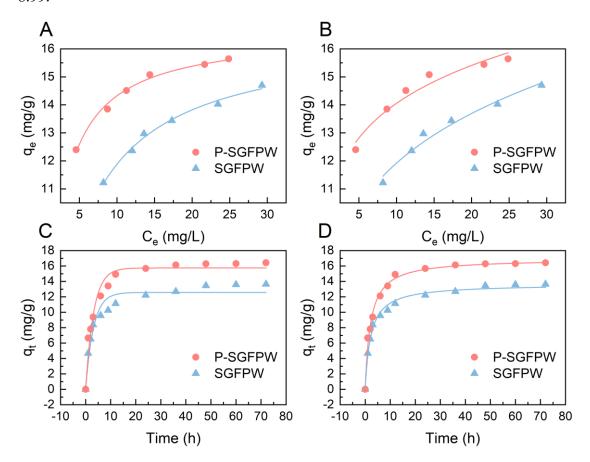


Fig. 5. Adsorption isotherms (A) Langmuir and (B) Freundlich of Li⁺ by the HMO adsorbent. Adsorption kinetics (C) pseudo-first-order kinetic model and (D) pseudo-second-order kinetic model of Li⁺ by the HMO adsorbent.

Table 1. Langmuir and Freundlich model fitting parameters of Li^+ adsorption.

	Langmuir model			Freundlich model			
	Q _m (mg/g)	$K_L (L/mg)$	R^2	$K_F\left(mg^{1-1/n}\cdot L^{1/n}/g\right)$ n		\mathbb{R}^2	
SGFPW	16.52	0.25562	0.9912	7.51	4.99277	0.9762	
P-SGFPW	16.61	0.62743	0.9911	10.35	7.52403	0.9492	

3.4. Adsorption kinetics

The kinetics of lithium adsorption on HMO adsorbents was studied to comprehend the adsorption behavior and explore the adsorption mechanism, which are important for the construction of reactors, such as fixed-bed reactors, and other fluidized bed operation (Hossain et al., 2022). Kinetic data were fitted using the pseudo-first-order kinetic model and the pseudo-second-order kinetic model, with results displayed in Fig. 5C and Fig. 5D, respectively. It can be seen that lithium is rapidly adsorbed in both SGFPW and P-SGFPW within 6 h and then the adsorption slows down significantly until near-equilibrium was reached after 48 h.. The kinetic parameters reported in Table 2 indicate that the pseudo-second-order kinetics model was slightly more suitable than the pseudo-first-order model. This result implies the adsorption rate is mostly controlled by chemical sorption or chemisorption, likely involving valency forces through exchange of electrons between H⁺ and Li⁺ (Ho and McKay, 1999). The final experimental adsorption capacities (16.4 mg/g with P-SGFPW) match well the estimated total adsorption capacity (16.9 mg/g with P-SGFPW) based on the pseudo-second-order model, as well as the maximum capacities calculated with Langmuir model for the isotherms (16.6 mg/g with P-SGFPW).

Table 2

Results of fitting the Li⁺ adsorption kinetics data with pseudo-first-order model and pseudo-second-order models.

	Pseudo-first-model			Pseudo-second-model		
	$q_{e,cal}$ (mg/g)	$\begin{array}{c} k_1 \\ (\text{min}^{-1}) \end{array}$	\mathbb{R}^2	q _{e,cal} (mg/g)	$\begin{array}{c} k_2 \left(g \\ mg^{-1}h^{-1} \right) \end{array}$	R^2
SGFPW	12.57	0.32133	0.9399	13.65	0.03294	0.9905
P-SGFPW	15.75	0.31384	0.9623	16.94	0.02777	0.9913

3.5. Selectivity

SGFPW contains several impurity ions (Na+, K+, Mg2+, Ca2+, Ba2+, Sr2+), thus understanding the selectivity of the HMO adsorbents for Li⁺ is important. Fig. 6A shows the partition coefficients (K_d) of different ions, while other parameters related to selectivity in the SGFPW and P-SGFPW streams are summarized in Table 3 and Table 4, respectively. The values of K_d for Li⁺ in both SGFPW and P-SGFPW were found to be much larger than those of other ions, indicating that Li⁺ is more easily adsorbed. Moreover, the coefficient (1074 mL/g) in P-SGFPW was greater than that (731.6 mL/g) in SGFPW, suggesting that precipitation somewhat improved the selectivity to lithium adsorption. The high selectivity is attributed to the ionic sieve effect of the spinel structure of the HMO adsorbent: the adsorption sites are so narrow that only ion with radius smaller than or equal to that of Li⁺ can diffuse and be adsorbed (Lai et al., 2020). Although the ionic radius of Mg²⁺ (0.065 nm) is smaller than that of Li⁺ (0.068 nm), the hydration free energy of Mg²⁺ (1828 KJ/mol) is nearly four times that of Li⁺ (474 KJ/mol), so Mg²⁺ ions need more energy to dehydrate, a necessary step to effectively reach and interact with the adsorption site (Xu et al., 2021). It should be highlighted that the absolute value of the adsorption capacity of Na⁺ was also large, which may be due to two main reasons. First, the concentration of Na⁺ in solution, which was roughly 400 times that of Li⁺ in weight. Likely, Na⁺

mostly adsorbed on the outer surface of the adsorbent. While the selectivity of lithium with respect to sodium was high, a significant fraction of adsorption sites are consumed by Na⁺, but this effect is inevitable due to the concentration ratio in solution, and in any case less severe than in seawater, whereby the weight-based Na/Li concentration ratio is approximately 200,000, hence 500 times higher than in the SGFPW stream employed in this study. Overall, the HMO adsorbent proved to be a highly selective adsorbent for lithium, especially in pre-treated streams.

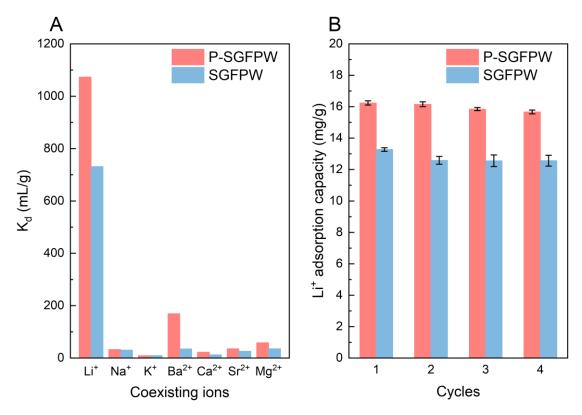


Fig. 6. (A) Partition coefficients (K_d) for coexisting ions. (B) Adsorption capacity of Li⁺ at different cycles.

3.6. Reusability of HMO adsorbent

A good adsorbent should have a stable structure and excellent reusability, which

helps reducing the cost of the adsorption process. The reusability of the HMO adsorbent was evaluated through four adsorption-desorption cycles, and the adsorption capacity of Li⁺ in each cycle is shown in Fig. 6B. The adsorption capacity was relatively stable with only a slight decrease after each regeneration, and it decreased from 16.2 to 15.7 mg/g and from 13.3 to 12.6 mg/g in P-SGFPW and SGFPW, respectively, after four cycles. This phenomenon may be due to the fact that HMO adsorbent has excellent crystallization properties, which is consistent with the XRD results. These results suggest that the HMO adsorbent can be reused in SGFPW and is a promising adsorbent for lithium recovery from SGFPW.

Table 3

Adsorption selectivity of HMO in the SGFPW.

	C_0	C _e	q_e	K _d	CF	\pmb{lpha}_{Me}^{Li}
	(mg/L)	(mg/L)	(mg/g)	(ml/g)	(ml/g)	$a_{\it Me}$
Li ⁺	31.46	18.15	13.27	731.58	421.94	1
Na^+	12122.22	11765.40	355.51	30.22	29.33	24.21
K^{+}	282.40	279.71	2.69	9.60	9.51	79.20
Ba^{2^+}	116.45	112.51	3.92	34.85	33.67	20.99
Ca^{2+}	523.12	516.74	6.35	12.30	12.15	59.49
Sr^{2+}	104.25	101.64	2.60	25.54	24.90	28.64
Mg^{2^+}	74.17	71.64	2.52	35.23	34.03	20.77

Table 4

Adsorption selectivity of HMO in the P-SGFPW.

	C_0	C_{e}	q_e	K_d	CF	$lpha_{Me}^{Li}$
	(mg/L)	(mg/L)	(mg/g)	(ml/g)	(ml/g)	α_{Me}
Li ⁺	31.40	15.13	16.24	1073.58	517.23	1
Na^+	13115.89	12704.55	410.50	32.31	31.30	33.23
K^{+}	281.45	278.99	2.46	8.83	8.76	121.52
Ba^{2+}	12.07	10.32	1.75	169.24	144.70	6.34
Ca^{2+}	43.35	42.39	0.96	22.63	22.13	47.44

Sr ²⁺	37.27	36.00	1.27	35.16	33.96	30.53
Mg^{2^+}	62.90	59.43	3.47	58.38	55.15	18.39

4. Conclusion

H_{1,33}Mn_{1,67}O₄ (HMO) adsorbent was synthesized by solid-phase reaction method and applied in the raw and pre-treated shale gas flowback and produced water (SGFPW and P-SGFPW, respectively). The adsorption capacity of lithium increased monotonically with the increase of pH in the range of 4 to 11. The lithium adsorption properties of the HMO adsorbent in both SGFPW and P-SGFPW matched well with the pseudo-second-order kinetic model and with the Langmuir equilibrium model, suggesting that the adsorption of lithium is likely a substitution process between H⁺ and Li⁺ and that the rate-limiting step may be chemical sorption. According to the adsorption isotherm analysis of lithium, the equilibrium adsorption capacity of lithium in the P-SGFPW was larger than that in the SGFPW. Due to the larger ionic radius and higher free energy of hydration, the HMO adsorbent has high selectivity to lithium, especially after the wastewater was pre-treated by precipitation. The partition coefficient of Li⁺ (1074mL/g) was much larger than that of Ba (169.24 mL/g), Mg (58.4 mL/g), Sr (35.2 mL/g), Na (32.3 mL/g), Ca (22.6 mL/g), K (8.83 mL/g). After four adsorption-desorption cycles, the adsorption capacity of lithium in P-SGFPW decreased from 16.2 mg/g to 15.7 mg/g, indicating that the HMO adsorbent had a stable structure and good reusability. However, it should be highlighted that the HMO adsorbent powder had poor dispersibility in aqueous suspension and its recovery may

be challenging during operation. Therefore, the granulation of powder HMO adsorbent would be important to promote its practical application.

Supporting Information

The Supporting Information is available free of charge online.

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Graphical Abstract

