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# Study on the CO Oxidation over Ceria-Based Nanocatalysts



Marco Piumetti<sup>®</sup>, Tahrizi Andana, Samir Bensaid<sup>\*</sup>, Nunzio Russo, Debora Fino and Raffaele Pirone

### Abstract

A series of ceria nanocatalysts have been prepared to study the structure dependency of the CO oxidation reaction. The ceria samples with well-defined nanostructures (nanocubes/Ce-NC and nanorods/Ce-NR) have been prepared using the hydrothermal method. Mesoporous ceria (Ce-MES) and ceria synthesized with solution combustion technique (Ce-SCS) have also been prepared for comparison. The lowest CO oxidation temperature has been reached by using ceria nanocubes (Ce-NC). This high activity draws immense contributions from the highly reactive (100) and (110) surfaces of the truncated nanocubes. The Ce-MES and Ce-SCS samples, despite their high surface areas, are unable to outdo the activity of Ce-NC and Ce-NR due to the abundant presence of (111) crystalline planes. This finding confirms the structure sensitivity of CO oxidation reaction catalyzed with ceria.

**Keywords:** CO oxidation, Properties of cerium-based oxides, Nanostructured ceria, Ceria-based catalysts, Structure sensitivity

## Background

During the last few decades, ceria-based materials have been widely used for many catalytic applications, including diesel soot combustion, CO oxidation, and VOC abatement [1-4]. These oxidation processes take advantage of the unique redox properties and high oxygen storage capacity (OCS) of ceria. In particular, CO oxidation, a primary function of three-way catalytic converters (TWCs), diesel oxidation catalysts (DOC), and diesel particulate filters (DPF), can be a prototypical reaction for probing the oxidation activity of ceria-based materials [5-8]. As a result, many studies have been carried out to evaluate the catalytic oxidation of CO by ceria and related materials. According to the literature [1, 5, 8, 9], CO oxidation over ceria-based catalysts takes place via a Mars-van Krevelen (MvK)-type mechanism, whereby the reaction involves alternating reductionoxidation steps on the solid surface with the formation of oxygen vacancies and their successive replenishment by gas-phase oxygen. However, a general consensus has been reached on the fact that CO interaction with ceria is structure sensitive [5, 8]. In addition, it has been revealed that ceria nanoparticles (i.e., nanowires and nanorods), exhibiting an abundance of (110) and (100) crystalline planes, are

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Exposure of specific crystallographic facets, together with the increased number of edges, corners, and facets, is pivotal in controlling the surface reactivity. In fact, nanocatalysts with multifaceted morphologies are highly desirable in catalysis. For fcc metal nanoparticles, the surface energy density  $(\gamma)$  for the three lowest index planes can be ordered as  $\gamma_{(111)} < \gamma_{(100)} < \gamma_{(110)}$  [16, 27]. Experimental studies have confirmed that this trend may also hold for ceria-based materials [16, 17]. Theoretical work on the interaction among CO and CeO<sub>2</sub> planes have shown the structure sensitivity for CO adsorption, thus suggesting weak CO adsorption for the most stable (111) surface and stronger chemisorption with (100) and (110) facets [18-23]. In this scenario, the metastable ceria (110) surface seems to be the most promising candidate for CO oxidation, since the formation of oxygen vacancies on the (110) planes needs the least amount of energy [24]. Several theoretical studies have shown that the oxygen vacancy formation energy depends on the nature of CeO<sub>2</sub> surface [25, 26]. It has also been wellestablished that the specific surface area of solid catalysts is a key factor in determining their overall catalytic activity in a wide number of catalytic reactions [3, 4, 27].



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Therefore, it is necessary to maximize the dispersion of the nanoparticles using high-surface-area supports, such as micro- and mesoporous materials [1, 28, 29].

In the present work, a set of  $CeO_2$  samples with different topological and textural properties ( $CeO_2$  nanocubes,  $CeO_2$  nanorods, mesoporous  $CeO_2$ , and  $CeO_2$  prepared by solution combustion synthesis) has been prepared to investigate the shape-dependency activity of ceria toward CO oxidation, a probe reaction for more complex oxidation processes. Then, the physico-chemical features of the prepared materials have been investigated using complementary techniques.

#### Methods

#### Synthesis of Samples

CeO<sub>2</sub> nanoparticles, denoted further as "Ce-NC" and "Ce-NR" for nanocubes and nanorods, respectively, were prepared via hydrothermal procedure, using sodium hydroxide as the precipitating agent, as we described elsewhere [30]. Mesoporous CeO<sub>2</sub>, denoted as "Ce-MES," was synthesized via nanocasting procedure using the SBA-15 silica template [30]. Lastly, a comparative CeO<sub>2</sub> sample was prepared via solution combustion synthesis (SCS) and further referred to as "Ce-SCS." A typical synthesis involved dissolving 3.8 g of Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O and 1.8 g of urea in 60 ml of deionized water at room temperature. The homogeneous solution was put into a ceramic crucible and then placed in a furnace at 650 °C for 20 min.

#### **Catalyst Characterization**

Powder X-ray diffraction (XRD) patterns were acquired on an X'Pert Philips PW3040 diffractometer using Cu K $\alpha$  radiation. The XRD spectra were recorded in the  $2\theta$ range of 20°–70° with 0.02° step size and a time per step of 0.2 s. The diffraction peaks were identified using powder diffraction files by the International Centre of Diffraction Data (ICDD). The average crystallite size was determined using Scherrer's equation,  $D = 0.9\lambda/b\cos\theta$ , where  $\lambda$  is the wavelength of the Cu K $\alpha$  radiation, *b* is the full width at half maximum (FWHM) in radians, 0.9 is the shape factor for spherical particles, and  $\theta$  is the angle of diffraction peaks.

BET-specific surface areas ( $S_{\text{BET}}$ ) and total pore volumes ( $V_{\text{p}}$ ) were measured through N<sub>2</sub> physisorption at -196 °C on a Micrometrics ASAP 2020 instrument. The samples were previously outgassed at 200 °C for 2 h to remove water and other atmospheric contaminants. The specific surface area of the samples was calculated using the BET method.

Sample morphology was investigated through a field emission scanning electron microscope (FESEM; Zeiss MERLIN, Gemini-II column) and transmission electron microscopy (TEM; Jeol JEM 3010 operating at 200 kV). Redox properties of the catalysts were investigated by means of X-ray photoelectron spectroscopy (XPS). The measurements were carried out on XPS PHI 5000 Versa probe apparatus using a band-pass energy of 187.85 eV, a 45° takeoff angle, and a 100.0- $\mu$ m-diameter X-ray spot size. Curve fits were performed by means of Multipack 9.0 software.

Reducibility of the catalysts was determined by H<sub>2</sub>-TPR measurements. Before the analyses, 50 mg of catalyst was pre-treated under air flow (40 cm<sup>3</sup> min<sup>-1</sup>) at 150 °C for 1 h, to remove water and other atmospheric contaminants, and then cooled to 25 °C with Ar. The analysis was conducted by heating the sample up to 950 °C at a constant rate of 5 °C min<sup>-1</sup> under an Ar flow (4.95 % molar H<sub>2</sub> in Ar). H<sub>2</sub> consumption was recorded using a thermal conductivity detector (TCD).

#### **Catalytic Activity Tests**

Catalytic tests for CO oxidation were carried out via a classical temperature-programmed combustion (TPC). The tests were conducted in a fixed-bed quartz reactor (4-mm inner diameter, U-tube) heated by a vertical tubular furnace. The reactor bed comprised of 0.1 g of powder catalyst. A K-type thermocouple was placed at the inlet of the reactor, in such a way that the tip was as close as possible to the catalytic bed. During the reaction, a 50 cm<sup>3</sup> min<sup>-1</sup> gas containing 1000 ppm of CO and 50 %-vol of air in N2 was continuously fed into the reactor, while the temperature of the furnace increased at a rate of 5 °C min<sup>-1</sup> until complete CO conversion was reached. The CO and CO2 concentrations at the outlet of the reactor were measured by NDIR analyzers (ABB Uras 14). Temperatures at which 10, 50, and 90 % of CO was converted were taken as indices of the catalytic activity. The tests were conducted for 2 cycles to evaluate the stability of the catalysts.

#### **Results and Discussion**

#### Textural and Redox Properties of the Catalysts

Some main textural properties of the catalysts, obtained from N<sub>2</sub> physisorption at -196 °C and X-ray diffraction analyses, are summarized in Table 1. Figure 1 shows the XRD diffractograms of the prepared catalysts. All samples exhibit similar patterns referring to cubic fluorite structure, marked by the existence of (111), (200), (220), (311), (222), and (400) planes [31]. The relative amount of the (110), (100), and (111) planes have been estimated by means of the XRD peak intensity ratios. As shown in Table 1, the highest (200)/(111) and (220)/(111) values appear for the Ce-NC sample thus showing the richest population of highly reactive (100) and (110) exposed planes. On the other hand, both the Ce-MES and Ce-SCS samples exhibit the highest amount of (111) surfaces.

Catalyst	S <sub>BET</sub>	Vp	Particle size	Intensity ratio <sup>a</sup>	Intensity ratio <sup>a</sup>
	$(m^2 g^{-1})$	$(cm^3 g^{-1})$	(nm)	(200)/(111)	(220)/(111)
Ce-NC	4	0.01	54	0.33	0.75
Ce-NR	4	0.01	43	0.32	0.71
Ce-MES	75	0.15	5	0.28	0.46
Ce-SCS	69	0.04	35	0.27	0.49

**Table 1** Textural properties of the catalysts, as obtained from  $N_2$  physisorption at -196 °C and X-ray diffraction analysis [30]

<sup>a</sup>XRD peak intensity ratios

However, it is worth noting that peak intensities of Ce-MES are rather broad and low. This is most likely due to nanocrystalline ceria frameworks in the sample that constitute mesoporous walls [32].

Although it is rather difficult to evaluate the average particle size through Scherrer's formula for morphologies that differ from spherical geometry, an evaluation was conducted to corroborate the TEM results. Scherrer's equation was employed to estimate the crystallite sizes of the catalysts: Ce-NC, Ce-NR, and Ce-SCS samples possess similar dimensions; the sizes fall in the range of 30–50 nm, whereas the mesostructured sample (Ce-MES) has the smallest particle dimension (5 nm).



The BET surface areas, obtained from  $N_2$  physisorption measurement at -196 °C, are very low for both the Ce-NC and Ce-NR samples (4 m<sup>2</sup> g<sup>-1</sup>), thus suggesting a small amount of interparticle voids. In contrast, Ce-MES and Ce-SCS have much higher surface areas than their nanostructured counterparts (75 and 69 m<sup>2</sup> g<sup>-1</sup>, respectively). Indeed, particles in the latter samples are arranged in a network that allows a creation of pores. Ce-MES has higher pore volume than Ce-SCS (0.15 and 0.04 cm<sup>3</sup> g<sup>-1</sup> for Ce-MES and Ce-SCS, respectively). This supports the fact that pores in Ce-MES are present in mesoscale and that those in Ce-SCS most likely exist in nanoscale.

Electron microscopy techniques were applied to find out the morphological properties of the catalysts. FESEM images of Ce-NC and Ce-NR are shown in Fig. 2a, b. The images show the morphology of nanostructured CeO<sub>2</sub> (namely cubes and rods, respectively). Nanocubes are present in the size range of 50 to 200 nm, whereas nanorods can be observed, with lengths of about 300-350 nm. In contrast, Ce-MES sample (Fig. 2c) comprises of self-assembled agglomerates of small particles (average size about 50 nm) with interparticle voids. The morphology of Ce-SCS, as reported elsewhere [30, 33, 34], exhibits typically foamy structures formed by agglomerates of small particles. The representative TEM image of Ce-NC, as seen in Fig. 3a, shows the morphology of truncated nanocubes with abundant (100) surfaces and additional (110) surfaces at their truncated corners. The truncated shape is a result of the reduced surface energy; sufficient energy would give rise to a well-defined cubic structure [35]. Figure 3b shows the TEM image of Ce-NR, displaying nanorods with a width of approximately 70 nm and a length of about 350 nm. A more profound analysis with TEM, not reported here for the sake of brevity, shows many (111) planes on Ce-NR. Such a surface has higher stability and possesses lower oxidation activity than the (100) and (110) surfaces [36, 37]. The Ce-MES sample exhibits uniform particles whose sizes range from 4 to 10 nm. These particles are interconnected with each other creating a three-dimensional mesoporous network. Deeper observation with TEM discovered the exposure of many (111) lattice planes [30]. TEM analysis of Ce-SCS, as



reported in our previous work [30], showed nonuniform polycrystalline ceria nanoparticles whose dimension ranges from 20 to 40 nm.

Redox properties of the catalysts were investigated by means of X-ray photoelectron spectroscopy. Figure 4a reports the XPS spectra of all samples in the O 1*s* core level. Two peaks are generally observed in the spectra: the first peak in the higher binding energy range (529.0– 529.7 eV) refers to weakly adsorbed oxygen at the surface of the catalyst ( $O_{\alpha}$  species) and the second peak in the lower binding energy range (530.8–531.5 eV) signifies the presence of oxygen in the crystal lattice of the catalyst [29, 32, 38, 39]. In general, the Ce-NC and Ce-NR samples possess a significantly low population of chemisorbed oxygen species due to their low  $O_{\alpha}$ -to- $O_{\beta}$ ratios (Table 2). Richer lattice oxygen species translate as easier oxidation reaction via a Mars-van Krevelen mechanism, which is known to govern the reaction of CO





oxidation [1, 5, 8, 9]. Figure 4b reports the XPS spectra of all samples in the Ce 3*d* core level. The doublets  $(v_0,u_0)$ ,  $(v_2,u_2)$ , and  $(v_3,u_3)$  were attributed to the Ce<sup>4+</sup> state while the doublet  $(v_1,u_1)$  was attributed to the Ce<sup>3+</sup> state [40]. Relative abundances of Ce<sup>3+</sup> and Ce<sup>4+</sup> (in atomic percentage, %) estimated from deconvoluted peaks of the Ce 3*d* spectra are summarized in Table 2. Among all the samples, Ce-SCS possesses the highest concentration of Ce<sup>3+</sup>, most likely owing to the highly exotermic reaction during the synthesis ( $\geq 600$  °C). Ce-SCS sample is also the most reducible catalyst among the series, mainly due to rich chemisorbed oxygen content on the catalyst surface.

**Table 2** Results of XPS spectra curve fittings on the O 1s andCe 3d core level [30]

Sample	$O_{\alpha}/O_{\beta}$	$Ce^{3+}(u_1+v_1)$ (%-atom)
Ce-NC	0.04	27.6
Ce-NR	0.04	25.5
Ce-MES	1.03	25.5
Ce-SCS	0.62	36.1

### **Reducibility of Catalysts**

Analyses with H2-TPR were carried out to understand the reduction behavior of the catalysts. Figure 5 shows the H<sub>2</sub>-TPR results of all samples. A typical reduction curve for cerium dioxide comprises of two peaks: (1) lowtemperature reduction peak in the range of 400-500 °C that characterizes the presence of chemisorbed oxygen and (2) high-temperature reduction peak in the range of 700-1000 °C that marks the slow release of lattice oxygen ("bulk reduction"). The low-temperature reduction peak for Ce-NC (curve a) is the highest among all catalysts (556 °C), confirming the low reducibility of the catalyst owing to the scarcity of surface-capping oxygen. A similar trend is also observed for Ce-NR (curve b), although the first peak appears at a lower temperature (465 °C). The analysis with Ce-MES sample (curve c) shows two reduction peaks centered at 450 and 535 °C, due to the surface reduction of small CeO<sub>2</sub> crystals, whereas the bulk reduction only occurs at 800 °C. Ce-SCS sample conforms the similar reduction trend; however, the second peak appears at the highest temperature (950 °C). This is most probably because of the low  $O_{\alpha}$ -to- $O_{\beta}$  ratio.



#### **Catalytic Activity Tests**

A series of tests with a classical TPC approach were carried out to investigate the role of structural and electronic properties on the activity of catalysts during CO oxidation reaction. A typical test was conducted according to this sequence: (1) reactor heat-up from ambient to 150 °C with air to remove any possible impurities; (2) introduction of the mixed gas (1000 ppm of CO and 50 %-vol of air in N<sub>2</sub>) to the reactor; and (3) second reactor heat-up from 150 °C to the temperature at which 100 % CO conversion is reached, (TOS = 2 h at 100 % conversion).

Figure 6a shows CO conversions as a function of reactor bed temperature over the fresh catalysts. In general, a complete CO oxidation is attainable between 150 and 400 °C. Table 3 summarizes the temperatures at which 10, 50, and 90 % of CO are converted. It appears that the performance of the catalysts follow this order: Ce-NC > Ce-NR > Ce-MES > Ce-SCS; the lowest oxidation temperature was obtained for Ce-NC ( $T_{50\%}$  = 214 °C) due to the larger number of reactive (100) and (110) surfaces. This suggests the beneficial effect of well-defined nanostructures by which some active surfaces are possibly generated. The nanocubic structures in the Ce-NC sample allow highly reactive (100) and (110) low-index surfaces to



be exploited, due to their low coordination numbers on the surface and better redox properties [30, 41], as revealed by XPS and H<sub>2</sub>-TPR analysis. However, the Ce-SCS and Ce-MES samples, despite their higher surface areas, exhibit lower oxidation reactivity due to the abundance of stable (111) surfaces that constitute the catalysts' structures. This may be inferred that CO oxidation reaction can be structure sensitive over ceria materials. These findings are in agreement with previous work [29, 30, 42–44], according to which ceria-based catalysts, exhibiting highly

 Table 3 CO oxidation activity results over ceria-based catalysts

Sample	1st cycle			2nd cycle		
	T <sub>10%</sub>	T <sub>50%</sub>	T <sub>90%</sub>	T <sub>10%</sub>	T <sub>50%</sub>	T <sub>90%</sub>
Ce-NC	203	214	219	177	185	190
Ce-NR	220	239	268	203	216	237
Ce-MES	232	260	270	214	230	255
Ce-SCS	257	303	338	230	244	266

reactive planes, show promising catalytic activities for diesel soot combustion despite their low textural properties.

In order to evaluate the stability of the catalysts, as well as the reliability of the data, re-tests have been carried out with previously used catalysts. Figure 6b shows CO conversions as a function of temperature over the "used" catalysts. The activity order of the catalysts appears to be preserved, with Ce-NC still being the most active catalyst. Interestingly, it was found that the retests with used catalysts resulted in better CO oxidation activity compared to those with "fresh" catalysts. The temperature indicators showed values 10-20 °C lower than those in the tests with fresh catalysts. The exposure of CO during the first cycle might give some carbonate species (e.g.,  $CO_3^{2-}$ ) on the solid surface that would help improve CO oxidation. It is known that oxygen spillover at catalyst surface contributes to the CO oxidation [45], and so, an abundant presence of surface polycarbonates, stable at high temperature, would have a beneficial effect on the reactivity of ceria nanocatalysts, as already observed by FT-IR analysis over ceria-zirconia catalysts [46].

#### Conclusions

As a whole, it has been observed that the catalytic performance for CO oxidation over CeO<sub>2</sub> nanocatalysts mainly depends on the presence of highly reactive (100) and (110) surfaces, thus confirming the structure sensitivity for this prototypical reaction. The best results, in terms of CO oxidation, were achieved for the Ce-NC catalyst, due to the abundance of coordinative unsaturated atomic sites in the (100) and (110) exposed surfaces of the CeO<sub>2</sub>truncated nanocubes. Moreover, this nanostructured catalyst has shown the lowest reducibility (*vide* H<sub>2</sub>-TPR data), thus validating the structure dependency for the CO oxidation reaction. On the other hand, worse CO conversion values were obtained for the high-surface-area catalysts (CeO<sub>2</sub>-MES and CeO<sub>2</sub>-SCS), thus showing the key role played by the structural properties of ceria. The least active catalyst was the CeO<sub>2</sub>-SCS, exhibiting the highest amount of stable (111) planes.

#### **Competing interests**

The authors declare that they have no competing interests.

#### Authors' contributions

TA synthesized and characterized the reported materials. MP and SB contributed to the interpretation of the results and coordinated the scientific work. NR, DF and RP supervised the work and approved the final version of the manuscript.

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