

Finite Fracture Mechanics: a deeper investigation on negative T-stress effects

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# Finite Fracture Mechanics: A deeper investigation on negative $T$ -stress effects

A. Sapora , V. Mantič

*Department of Structural, Geotechnical and Building Engineering, Politecnico di Torino, Corso Duca degli Abruzzi 24, 10129, Torino, Italy*

*Group of Elasticity and Strength of Materials, School of Engineering, Universidad de Sevilla, Camino de los Descubrimientos s/n, 41092, Sevilla, Spain*

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## Abstract

In the present work, the coupled stress and energy criterion of Finite Fracture Mechanics (FFM) is applied to investigate negative  $T$ -stress effects related to mixed-mode brittle fracture of cracked elements. Only two material parameters are involved in the analysis, the tensile strength and the fracture toughness, which are independent of the mode-mixity. Below a critical  $T$ -value, the shear contribution to the strain energy release rate (ERR) starts to prevail in the mode  $II$ -dominated zone. This affects FFM **results** in terms of: (i) the fracture loci, with the critical mode  $II$ -stress intensity factor (SIF) never exceeding the fracture toughness; (ii) the critical kinking angle and the actual crack advance (which results to be a structural parameter), both decreasing to infinitesimal quantities as mode  $II$ - loading conditions are approached. **These predictions can be revised by considering a large amount of energy dissipated under mode  $II$  loading conditions and by assuming a mode-mixity dependent ERR. A discussion on experimental data for brittle and quasi-brittle materials available in the literature is included.**

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*Email address:* alberto.sapora@polito.it, mantic@us.es (A. Sapora , V. Mantič )

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9 *Keywords:* Mixed-mode brittle FFM, crack kinking, negative  $T$ -stress, ERR

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## 10 11 12 **1. Introduction**

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15 Effects of  $T$ -stress on mixed-mode brittle fracture of cracked elements have  
16 been extensively investigated by means of models combining a linear-elastic anal-  
17 ysis with an internal material length. The simple point stress criterion was con-  
18 sidered in Kosai et al. (1993); Seweryn (1998) and later developed and applied to  
19 different experimental data in Smith et al. (2001, 2006), where it was renamed as  
20 Maximum Tangential Stress (MTS) criterion. Nonlocal (average) stress and en-  
21 ergy approaches were formalized by Seweryn (1998), although the internal criti-  
22 cal distance related to the latter was apparently underestimated (Pugno and Ruoff,  
23 2004). Stress and energy criteria, despite their simplicity and general accuracy,  
24 remain distinct and the fulfilment of one of them usually implies the violation of  
25 the other one: this could lead to some drawbacks, as pointed out in Carpinteri et al.  
26 (2008). **For a detailed overview on  $T$ -stress effects in fracture mechanics, see**  
27 **Gupta et al. (2015) and therein references.**

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40 More recently, also FFM approaches coupling the energy balance with a stress  
41 requirement, either punctual (Leguillon, 2002) or averaged (Cornetti et al., 2006),  
42 were proposed in this framework. One of the most important feature related to  
43 FFM is that the crack advance becomes a structural parameter, depending also  
44 on the geometrical characteristics (Carpinteri et al., 2011; Sapora et al., 2013).  
45 In order to take  $T$ -stress effects into account, the coupled criterion proposed by  
46 Leguillon was developed numerically (Leguillon and Murer, 2008), by a two-scale  
47 asymptotic matching procedure (Leguillon, 1993). On the other hand, a semi-  
48 analytical approach was adopted by Cornetti et al. (2014), by implementing the  
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9 angular functions tabulated in Tada et al. (1985); Melin (1994), the approximating  
10 expressions of which can be found in Amestoy and Leblond (1992). Eventually,  
11 FFM was applied to investigate  $T$ -stress effects on crack bifurcation phenomena  
12 in ceramic laminates by Ševeček et al. (2014).  
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16 From a qualitative point of view, theoretical models all predict the same gen-  
17 eral behavior for a positive  $T$ -stress: it decreases both the failure load and the  
18 critical kinking angle. In case of pure mode  $I$  loading conditions, if  $T \geq T^* > 0$   
19 the crack ceases to propagate collinearly and the critical value of mode  $I$ -SIF  $K_{If}$   
20 deviates smoothly from the material fracture toughness  $K_{Ic}$  (Smith et al., 2001;  
21 Leguillon and Murer, 2008; Cornetti et al., 2014), see also the stability analysis  
22 carried out in Cotterell and Rice (1980). Estimations for the threshold  $T^*$  differ  
23 slightly between each other according to the implemented failure criterion.  
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27 An opposite trend is observed for negative  $T$ -stresses. Indeed, according to  
28 stress-based criteria, the critical value of mode  $II$ -SIF  $K_{II f}$  can exceed  $K_{Ic}$  for  
29 sufficiently large negative  $T$ -values (Smith et al., 2001). This result was later ex-  
30 ploited to attempt to justify the behavior of some rock and glass material samples  
31 tested under mixed-mode loading conditions (Awaji and Sato, 1978; Khan and  
32 Al-Shayea, 2000; Chang et al., 2002; Aliha et al., 2006; Ayatollahi and Aliha,  
33 2009a).  
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37 By means of a FFM approach based on the coupling of a tensile stress condi-  
38 tion with an incremental energy requirement, in the present work it is shown that  
39 below a critical negative threshold for  $T = T^{**} < 0$ , the shear contribution to the  
40 ERR starts to prevail as mode  $II$ -loading conditions are approached. Theoretical  
41 predictions show a jump to infinitesimal quantities as concerns the critical kinking  
42 angle and the crack advance, and a unit limit value for the ratio between  $K_{II f}$  and  
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9  $K_{Ic}$ . As  $T$  further decreases, the shear energy dominated zone enlarges, giving rise  
10 to a smooth limit-behavior over the whole range of mode-mixities.  
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13 FFM predictions could be modified by assuming a fracture energy dependent  
14 on the mode-mixity, as commonly done in interface fracture mechanics (Hutchin-  
15 son and Suo, 1992; Liechti and Chai, 1992; Banks-Sills and Ashkenazi, 2000;  
16 Mantič et al., 2006). In this case,  $K_{II\bar{f}}$  can theoretically exceed  $K_{Ic}$  **as observed**  
17 **in different experimental tests concerning polymeric materials (Smith et al.,**  
18 **2006), rocks (Awaji and Sato, 1978; Khan and Al-Shayea, 2000; Chang et al.,**  
19 **2002; Aliha et al., 2006), alumina and glass ceramics (Panasyuk et al., 1965;**  
20 **Shetty et al., 1986, 1987), and wood (Anaraki and Fakoor, 2010). Indeed,**  
21 **FFM results (and, more in general, results by any criterion involving a crit-**  
22 **ical distance) reveal to be satisfactory only in case of a very brittle behavior,**  
23 **as long as the asymptotic expressions for the stress field and the SIFS related**  
24 **to a kinked crack are implemented.**  
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36 The analysis presented below is limited to a piecewise straight crack propaga-  
37 tion, i.e. no curved crack-kinking (Amestoy and Leblond, 1992), in the framework  
38 of two-dimensional linear elasticity, i.e. no three-dimensional effects (Berto et al.,  
39 2011; Kotousov et al., 2013), where only the  $T$ -stress component parallel to the  
40 main crack plays a significant role, i.e. negligible constant (compressive) stresses  
41 perpendicular to the main crack (Isaksson and Ståhle, 2002; Li et al., 2009).  
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## 48 **2. FFM criterion**

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51 The present FFM criterion (Carpinteri et al., 2009, 2010) is based on the as-  
52 sumption of a finite crack extension  $\Delta$  and on the contemporaneous fulfilment of  
53 two conditions. The former is a stress requirement: the average circumferential  
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9 stress  $\sigma_{\theta\theta}(r, \theta)$  on  $\Delta$ , prior to the crack extension, must be greater than the mate-  
10 rial tensile strength  $\sigma_u$ . The latter is the energy balance: the integral of the strain  
11 ERR (or crack-driving force)  $G$  on  $\Delta$  (i.e. the energy available for a crack incre-  
12 ment  $\Delta$ ) must be higher than the fracture energy  $G_c$  times the crack increment. By  
13 referring to a cracked element with a polar reference system placed at the main  
14 crack tip (Fig. 1), FFM conditions write:  
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$$20 \left\{ \begin{array}{l} \int_0^\Delta \sigma_{\theta\theta}(r, \theta) dr \geq \sigma_u \Delta, \\ \int_0^\Delta \{ [K_I^k(c, \theta)]^2 + [K_{II}^k(c, \theta)]^2 \} dc \geq K_{Ic}^2 \Delta. \end{array} \right. \quad (1)$$

21 Note that the energy balance in (1) is expressed in terms of the SIFs related to the  
22 kinked crack,  $K_I^k$  and  $K_{II}^k$  for mode  $I$  and mode  $II$ , respectively, by means of the  
23 well-known Irwin's relationships (assuming  $K_I^k \geq 0$  and plain strain conditions):  
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$$32 G = G_I + G_{II} = \frac{(K_I^k)^2 + (K_{II}^k)^2}{E'}, \quad G_c = \frac{K_{Ic}^2}{E'}, \quad (2)$$

33 where  $E' = E/(1 - \nu^2)$ ,  $E$  being Young's modulus and  $\nu$  Poisson's ratio of the  
34 material.  
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39 Once  $\theta$  is fixed, for monotonically decreasing  $\sigma_{\theta\theta}(r, \theta)$  and monotonically  
40 increasing  $G(c, \theta)$  functions (as supposed in the present analysis), the lowest load  
41 satisfying the inequalities in (1) is achieved when the equal sign holds.  
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### 46 2.1. Stress field and SIFs for kinked cracks

47 In order to implement FFM, the the expressions for the stress and the SIFs  
48 related to a kinked crack to be inserted into system (1) are invoked. By taking  
49  $T$ -stress effects into account, the circumferential stress field  $\sigma_{\theta\theta}(r, \theta)$  at the main  
50 crack tip can be approximated as (see Fig.1 with  $c = 0$ ):  
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$$\sigma_{\theta\theta}(r, \theta) = \frac{K_I}{\sqrt{2\pi r}} f_{\theta\theta}^I(\theta) + \frac{K_{II}}{\sqrt{2\pi r}} f_{\theta\theta}^{II}(\theta) + T \sin^2 \theta, \quad (3)$$

where  $K_I$  and  $K_{II}$  are the SIFs related to the main crack, whereas  $f_{\theta\theta}^I$  and  $f_{\theta\theta}^{II}$  are the corresponding angular functions, cf. Seweryn (1998).

On the other hand, by dimensional analysis concepts and the principle of superposition, the SIFs related to a kinked crack of length  $c$  can be expressed as (He et al., 1991; Amestoy and Leblond, 1992):

$$K_I^k(c, \theta) = \beta_{11}(\theta)K_I + \beta_{12}(\theta)K_{II} + \beta_1(\theta)T\sqrt{c}, \quad (4)$$

$$K_{II}^k(c, \theta) = \beta_{21}(\theta)K_I + \beta_{22}(\theta)K_{II} + \beta_2(\theta)T\sqrt{c}. \quad (5)$$

Approximating analytical expressions for the angular functions  $\beta$  are reported in Amestoy and Leblond (1992), and tabulated values can be found in Tada et al. (1985); Melin (1994); Fett et al. (2004). Note that  $\beta_2, \beta_{12}$  and  $\beta_{21}$  are odd functions, whereas  $\beta_1, \beta_{11}$  and  $\beta_{22}$  result to be even. For a null  $T$ -stress the last term in (4) and (5) vanishes, and the two expressions result thus independent of the crack length  $c$ .

Let us now introduce the fracture mode-mixity of the main crack  $\psi = \arctan(K_{II}/K_I)$ , the dimensionless kinked crack advance  $\delta = \Delta/l_{ch}$  (with  $l_{ch} = (K_{Ic}/\sigma_u)^2$ ) and the dimensionless  $T$ -stress,  $\tau = T\sqrt{l_{ch}}/\sqrt{K_I^2 + K_{II}^2}$ . In Fig. 2 the contributions of  $G_I$  and  $G_{II}$  evaluated by inserting (4) and (5) into (2) are presented for  $\psi = 90^\circ$  (mode II loading conditions) and  $\delta = 1/4\pi$ . It can be observed that  $G_I$  decreases

as  $\tau$  decreases, while the maximum of  $G_{II}$  keeps constant and it corresponds to  $\theta = 0^\circ$  (collinear crack propagation).

## 2.2. FFM implementation

As already stated, at incipient failure ( $K_I = K_{If}$ ), the coupled condition (1) becomes a system of two equations in two unknowns: the critical crack advancement  $\delta_c$  and the failure load, implicitly embedded in the  $K_{If}$  function.

The substitution of (3), (4) and (5) into (1) provides after some simple algebraic manipulations, cf. Cornetti et al. (2014):

$$\begin{cases} \frac{K_{If}}{K_{Ic}} = \frac{\sqrt{\delta}}{\bar{f}_{\theta\theta}^I + \tan\psi\bar{f}_{\theta\theta}^{II} + \bar{\tau}\sqrt{\delta}\sin^2\theta}, \\ \delta = \frac{(\bar{f}_{\theta\theta}^I + \tan\psi\bar{f}_{\theta\theta}^{II} + \bar{\tau}\sqrt{\delta}\sin^2\theta)^2}{(\bar{\beta}_{11} + \bar{\beta}_{12}\tan\psi + \bar{\beta}_{22}\tan^2\psi) + \frac{4\bar{\tau}\sqrt{\delta}}{3}(\bar{\beta}_1 + \bar{\beta}_2\tan\psi) + \frac{\bar{\tau}^2\delta}{2}(\beta_1^2 + \beta_2^2)}, \end{cases} \quad (6)$$

where  $\bar{f}_{\theta\theta}^i = \sqrt{2/\pi}f_{\theta\theta}^i$  ( $i = I, II$ ),  $\bar{\tau} = \tau\sqrt{(1 + \tan^2\psi)}$  and the following combinations of the SIF angular functions are defined:

$$\bar{\beta}_1 = \beta_1\beta_{11} + \beta_2\beta_{21}, \quad \bar{\beta}_2 = \beta_1\beta_{12} + \beta_2\beta_{22}, \quad (7)$$

$$\bar{\beta}_{11} = \beta_{11}^2 + \beta_{21}^2, \quad \bar{\beta}_{22} = \beta_{12}^2 + \beta_{22}^2, \quad \bar{\beta}_{12} = 2(\beta_{11}\beta_{12} + \beta_{21}\beta_{22}). \quad (8)$$

Observe that, for given loading and structural properties,  $\psi$  and  $\tau$  are fixed. In order to implement FFM, the latter equation in (6) should be firstly solved: a different crack advance  $\delta$  corresponds to a different kinking angle  $\theta$ . Each couple  $(\delta, \theta)$  must be substituted into the former equation: the actual crack advance  $\delta_c$



and critical kinking angle  $\theta_c$  are those which minimize the  $K_{If}$  function. The relationship  $K_{II} = \tan \psi K_{If}$  then provides the corresponding value for  $K_{II}$ .

### 3. Discussion on FFM results

The results obtained by the above FFM procedure are depicted in Figs. 3, 4 and 5. As can be extracted from Figs. 3 and 4, FFM predicts higher failure loads and critical kinking angles for decreasing negative  $T$ -stresses over the range  $-0.3 \leq \tau \leq 0$ . The crack advance results to be a smooth monotonically decreasing function of  $\psi$  (Fig. 5).

On the other hand, when  $\tau = \tau^{**} \simeq -0.325$ , it is observed that  $K_{II}$  does not increase any more for  $\psi = 90^\circ$  (mode  $II$ -loading conditions), and the corresponding values for  $\theta_c$  and  $\delta_c$  jump from  $-68.75^\circ$  and  $0.5233$ , respectively, to infinitesimal quantities. The limit  $\theta_c = 0^\circ$  can never be reached from a theoretical point of view (taking the imposed tensile stress condition into account), since the angular function  $f_{\theta\theta}^{II}$  would vanish, leading to a null circumferential stress. If one keeps on decreasing  $\tau$ , results start to converge to the above mentioned values in proximity of mode  $II$ , till a smooth transition is observed over the full range of mode mixities  $\psi$ . Observe that the ratio  $K_{II}/K_{Ic}$  never exceeds the unit value (Fig. 3).

The following explanation to this behavior can be provided: the contribution of  $G_I$  in the energy balance of system (1) decreases as  $T$  decreases, but still remains dominant (as it happens for positive  $T$ -stress) over the range  $-0.325 < \tau \leq 0$ . Indeed, starting from  $\tau^{**} \simeq -0.325$  and  $\psi = 90^\circ$ , the maximum energy available for a crack advance is provided by  $G_{II}$  for an infinitesimal angle (Fig. 2). In order for the stress requirement in (1) to match this condition, the crack advance must become infinitesimal too, so that tensile stresses result to be high enough.

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9 The present behavior can not be detected by stress-based approaches, such  
10 as MTS criterion (Smith et al., 2001), **which is based on** the simple condition  
11  $\sigma_{\theta\theta}(r = r_c) \geq \sigma_c$ , with  $r_c = l_{ch}/2\pi$ . In Smith et al. (2001) it was found that,  
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13 even for large negative  $\tau$ , the failure load keeps on increasing continuously and  
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15 the kinking angle decreases uniformly in modulus. The explanation of this trend  
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17 is quite straightforward: for a fixed critical distance, circumferential stresses de-  
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19 crease as  $T$  decreases, as it immediately follows from (3).  
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22 A comparison with *ad hoc* experimental results would reveal as fundamen-  
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24 tal to understand which is the real behavior observed from a testing procedure.  
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26 **Indeed, almost all experiments on brittle polymeric materials involve either**  
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28 **a positive or a moderate negative  $T$ -stress so that  $\tau > -0.3$  (Erdogan and**  
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30 **Sih, 1963; Williams and Ewing, 1972; Carpinteri et al., 1979; Ueda et al.,**  
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32 **1983; Maccagno and Knott, 1989; Ayatollahi and Aliha, 2009b; Saghafi et al.,**  
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34 **2013). Under these assumptions, FFM has already been proven to furnish ac-**  
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36 **curate result (Cornetti et al., 2014). To the best of the authors' knowledge,**  
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38 **the only significant data set refers to PMMA samples tested in Smith et al.**  
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40 **(2006) under specific mode II loading conditions: in this case it was found**  
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42 **that  $K_{II f}/K_{Ic} \simeq 1.21$  and  $\theta_c \simeq -55.2^\circ$ , for  $\tau \simeq -0.41$ . It should be noted how-**  
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44 **ever that some scattering on experimental results was found during tests and**  
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46 **that the curvature of the crack path was considerable, as observed by the**  
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48 **authors themselves.**

49 **Furthermore, disk specimens subjected to diametral compression often**  
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51 **show a ratio  $K_{II f}/K_{Ic} > 1$  in the mode II-dominated zone (even for  $\tau$  lower in**  
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53 **modulus than -0.3) for some materials such as rocks (Awaji and Sato, 1978;**  
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55 **Khan and Al-Shayea, 2000; Chang et al., 2002; Aliha et al., 2006), and alu-**  
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mina and glass ceramics (Panasyuk et al., 1965; Shetty et al., 1986, 1987; Awaji and Kato, 1999). As already outlined in Cornetti et al. (2014), since rock materials are less brittle than polymeric materials (i.e. they present a higher  $l_{ch}$ ), the related critical distance results larger and  $T$ -stress effects become more significant. On the other hand, if the critical length (described quantitatively by  $l_{ch}$ ) is not sufficiently small with respect to the notch depth  $a$ , ( $l_{ch}/a \simeq 1.7$  and  $1.3$ , for instance, for the limestone and marble samples tested in Khan and Al-Shayea (2000) and Aliha et al. (2006), respectively), which in turn has to be sufficiently small with respect to the characteristic dimension of the specimen, the theory of critical distances can not be implemented accurately by simple asymptotic expansions (3), (4) and (5). Moreover, in these cases a complete analysis would require the inclusion of friction, of the influence of the root radius (McClintock (1963); Cotterell (1972)), and the implementation of a different mode of fracture (sliding and not opening), as suggested in Rao et al. (2003). Further studies are in progress. Similar arguments hold for polycrystalline ceramics: past studies showed that crack-surface resistances arising from grain interlocking and abrasion were the main sources of the increased fracture resistance in mode II, Singh and Shetty (1989); Rosenfield and Madjumar (1992).

The fact that  $K_{II_f}/K_{Ic}$  can exceed the unit value seems however to emerge clearly from some experiments. In order to justify this trend by FFM, let us observe that estimates of the toughening of elements under shear should consider possible local plastic and viscoelastic dissipation, crack face asperity shielding and frictional effects. In other words, the assumption of  $G_c$  to be constant is reasonable only if the  $G_I$ -contribution prevails in (2), whereas a larger amount of dis-

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sipated energy should be associated to crack kinking dominated by  $G_{II}$ , typically occurring for  $\theta \simeq 0^\circ$ . This point has been largely investigated in the framework of interface fracture mechanics, where the interface fracture toughness was found to be even more than ten times greater as mode  $II$  is approached (Hutchinson and Suo, 1992; Liechti and Chai, 1992; Banks-Sills and Ashkenazi, 2000; Mantič et al., 2006). In order to overcome this drawback, one of the most implemented fracture criterion writes (Hutchinson and Suo, 1992):

$$\frac{G_I}{G_{Ic}} + \frac{G_{II}}{G_{IIc}} = 1, \quad (9)$$

where  $G_{IIc} = G_{Ic}/\gamma$  has the interpretation of pure mode  $II$  toughness and  $\gamma$  is a parameter weighting the mode  $II$ -contribution. It vanishes for  $\gamma \rightarrow 0$ , whereas  $\gamma = 1$  corresponds to an ideally brittle material. Note that the condition  $\gamma \rightarrow 0$  provides the basis for the well-known  $K_{II}^k = 0$  criterion proposed on the basis of simple symmetry arguments (Goldstein and Salganik, 1974; Amestoy and Leblond, 1992; Becker et al., 2001), and that an equivalent relationship to (9) was adopted in Seweryn (1998) **and suggested in Leguillon and Murer (2008)**.

In order to improve FFM predictions, from an equivalent point of view, one could consider the following modified ERR instead of  $G$  in (2):

$$\bar{G} = G_I + \gamma G_{II}. \quad (10)$$

By summarizing, Fig. 6 shows FFM results for pure mode  $I$  and mode  $II$  loading conditions, over the range  $-1 \leq \tau \leq 1$ . Predictions (represented by continuous lines) are obtained by setting  $\gamma = 1$  in (10), i.e. by considering the classical ERR (2). On the other hand, if the limit case  $\gamma = 0$  is implemented in (10), both the failure load and the critical kinking angle keep on increasing continuously for

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9 decreasing  $T$  (dashed lines).

10 **Results have been applied to PMMA samples tested in (Smith et al., 2006),**  
11 **showing an acceptable agreement as concerns the load (the mean percentage**  
12 **error being nearly  $-12\%$ ) and a significant deviation as regards the angle**  
13 **( $+13^\circ$ ). On the other hand, FFM predictions on data related to a positive**  
14  **$T$ -stress, also reported in Fig. 6 for the sake of completeness, reveal to be**  
15 **satisfactory.**  
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#### 24 **4. Conclusions**

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27 Effects of negative  $T$ -stress on brittle fracture of cracked elements were in-  
28 vestigated by means of FFM. A competition between mode  $I$  and mode  $II$  contri-  
29 butions to the strain ERR was observed: for sufficiently large negative  $T$ -values,  
30 the latter prevails affecting the failure load and the critical kinking angle in the  
31 mode  $II$ -dominated zone. Predictions are in contrast with those provided by cri-  
32 teria based on simple stress considerations and showing monotonically increasing  
33 functions of the failure load and the kinking angle as  $T$  decreases. The idea of  
34 modifying FFM results by considering a mode-mixity dependent fracture tough-  
35 ness was furnished: this assumption takes the larger dissipated energy due to shear  
36 fracture mechanisms into account. **Due to the difficulties in performing tests**  
37 **under prevalent mode  $II$  loading conditions and a sufficiently negative  $T$ -**  
38 **stress, only few experimental results are present in the literature. Therefore,**  
39 **to carry out ad-hoc mode  $II$  tests on very brittle materials, starting from the**  
40 **procedure proposed in Smith et al. (2006), emerges as an important task to**  
41 **further corroborate the present analysis. Eventually, it appears that the rela-**  
42 **tively brittle response of rocks and the behavior of alumina and glass ceram-**  
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9 **ics cannot be described accurately by means of simple asymptotic theories**  
10 **involving a critical distance. In other words, the large ratios  $K_{II_f}/K_{Ic} > 1$**   
11 **detected experimentally cannot be imputable only to the negative  $T$ -stress**  
12 **contribution. In order to improve FFM predictions, the following steps are**  
13 **suggested: i) higher order terms should be included in the series expansions**  
14 **for the stress field and the SIFs related to a kinked crack; ii) the effects of**  
15 **rubbing of these materials during the failure mechanism has to be properly**  
16 **taken into account.**  
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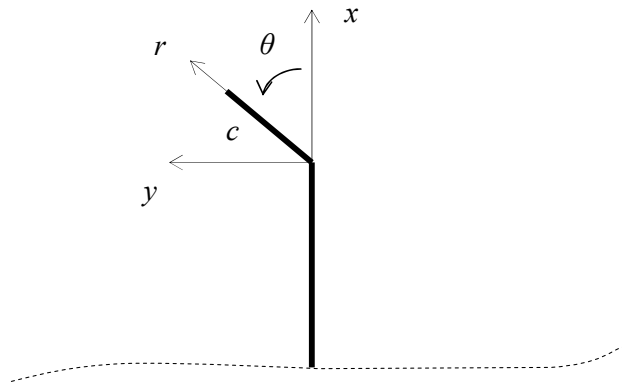


Figure 1: Cracked element with a kinked crack of length  $c$ .

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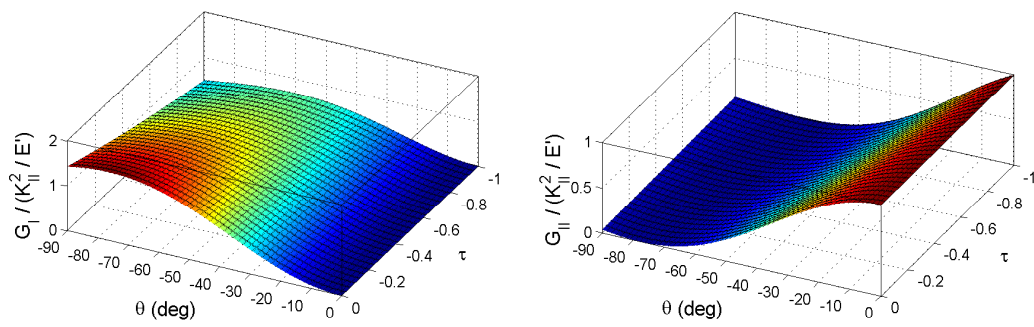


Figure 2: ERR contributions related to mode *I* and mode *II* as functions of  $\theta$  and  $\tau$ . Results refer to  $\psi = 90^\circ$  (mode *II* loading conditions) and  $\delta = 1/4\pi$ .



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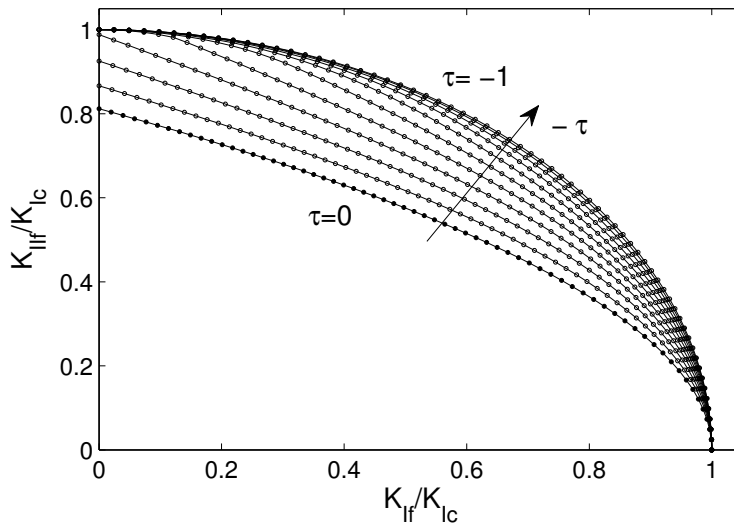


Figure 3: FFM fracture loci: effects of negative dimensionless  $T$ -stresses  $\tau$ , ranging from 0 to -1 with a step equal to -0.1.

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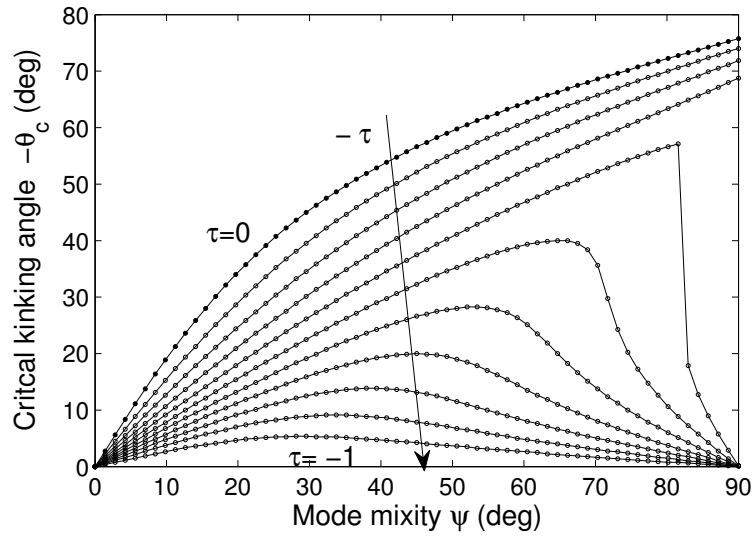


Figure 4: FFM critical kinking angle: effects of negative dimensionless  $T$ -stresses  $\tau$ , ranging from 0 to -1 with a step equal to -0.1.

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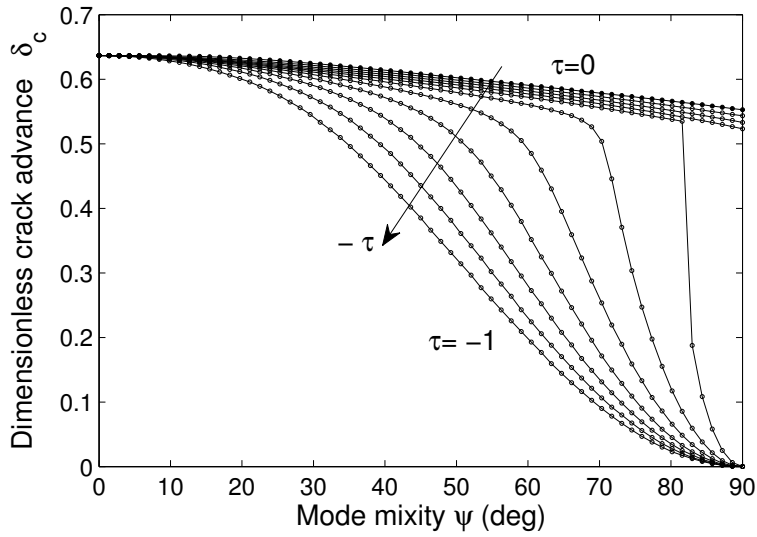


Figure 5: FFM critical crack advance: effects of negative dimensionless  $T$ -stresses  $\tau$ , ranging from 0 to -1 with a step equal to -0.1.

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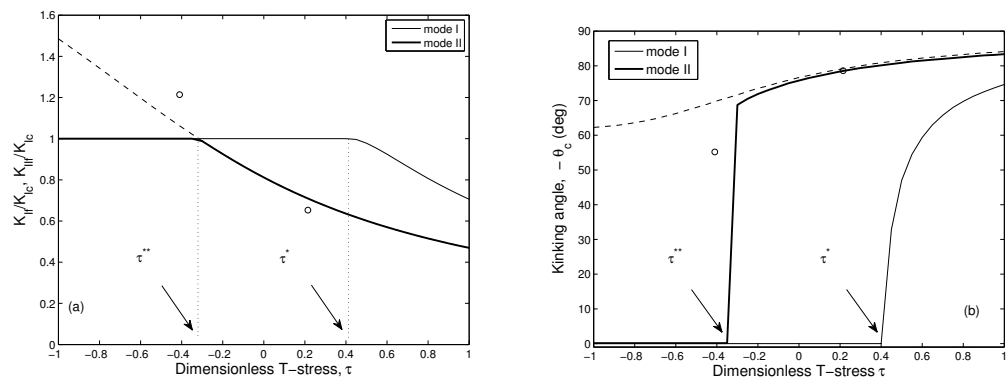


Figure 6: FFM predictions for pure mode *I* and mode *II* loading conditions: *T*-stress effects on (a) the critical SIFs and (b) the critical kinking angle. The dashed line refers to the predictions obtained by assuming  $\gamma = 0$  in the modified ERR (10). The critical thresholds for mode *I* and mode *II* loading conditions are denoted by  $\tau^*$  and  $\tau^{**}$ . **Circles refer to the experimental data obtained by Smith et al. (2006)**