

Heterostructured ceramic materials based on PZTN-CFO compounds

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Abstract

Multiferroic composites are currently one of the hot research topics [1]. A great research effort is in progress to improve the fabrication of PZT–CoFe₂O₄ (PZT–CFO) composites due to the excellent piezoelectric properties showed by the PZT material class and the large magnetostrictive coefficient of the CFO. Unfortunately unwanted reactions occur during densification of PZT–CFO materials at 1100-1200 °C. They are promoted by initial PbO loss that is calculated through XRD analysis, considering the amount of ZrO₂ and variation of perovskite's tetragonality. The resulting titania reacts with CFO to form cobalt titanate [2]. The microstructure of the composites at 26-81 mol% CFO content was thoroughly investigated; the CFO grain size distribution can be mono- or bi-modal and overgrowth [3] occurs. By setting a quite-fast sintering full densification and prevention of unwanted reactions was achieved for the PZT:CFO 74:26 composites. The high coercivity (789 Oe) shown by these composites is correlated to the good dispersion of euhedral CF grains (250 nm) in the PZT matrix that is affected as well by limited grain growth (240 nm).

bO

X-ray diffraction analysis has been used as "chemical analysis" to calculate the PbO loss [2]

Once the baddeleyite molar fraction (b) and the change of the amount of Zr in the produced perovskite (y) has been quantified by quantitative XRD analysis, the PbO

Experimental

CONVENTIONAL SINTERING

- Heating rate < 300 °C/h
- Sintering temperature \approx 70% T_m
- Soaking time > 0.5 h
- Natural cooling

Results

• QUITE-FAST SINTERING [2]

- Heating rate > 300 °C/h
- Sintering temperature < 70% T_m
- Soaking time < 0.5 h

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Cooling rate > 30 °C/min

T_m: melting point

loss $\delta = \mathbf{f} \cdot m_{Ph}$, expressed as molar fraction, can be calculated through the following equation [2]: Galizia $\delta = m_{Pb} (b/m_{Zr} - 1 + y)/y$

equation! where m_{Ph} and m_{7r} are the stoichiometric coefficients of lead and zirconium in the





- \blacktriangleright Relative density \approx 82%
- PbO loss: 11%
- \blacktriangleright Bi-modal CoFe₂O₄ grain size distribution
- \succ CoFe₂O₄ overgrowth by multiple parallel twinning [3]
- Coercivity: 239 Oe

Discussion

1°

Reduced remnant magnetisation $M_{r}/M_{s} = 0.17$



- \blacktriangleright Relative density \approx 99%
- ➢ PbO loss < 0.2%</p>
- \blacktriangleright mono-modal CoFe₂O₄ grain size distr.
- \blacktriangleright Euhedral CoFe₂O₄ grains \approx 250 nm
- Coercivity: 789 Oe
- Reduced remnant magnetisation $M_{r}/M_{s} = 0.38$

It has been demonstrated that twin boundaries on CFO {111} planes act as the effective pinning centers for the hindrance of domain wall movement [2, 3]



Globus model extension [3]

In the Globus model the linear correlation between the initial susceptibility (χ_i) and the mean grain diameter (D_m) is expressed as:

 $\chi_i = 2\pi M_S^2 D_m / K$



Reactions due to the PbO loss at the PZT/CFO interfaces [2]

)
$$Pb(Zr_{0.52}Ti_{0.48})O_3 \rightarrow fPbO_{(g)}^{S} + (1 - f)Pb(Zr_{0.52 \cdot y}Ti_{0.48 \cdot z})O_3 + 0.52(1 - y + fy)ZrO_2 + 0.48(1 - z + fz)TiO_2$$

 $ZrO_2: baddeleyite$

 2°) Displacement reaction [2, 4]:

TiO₂ + CoFe₂O₄ \rightarrow CoTiO₃ + Fe₂O₃

Acknowledgements

RITMARE flagship project is gratefully acknowledged



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where M_s is the saturation magnetization, and K is the global anisotropy. But, where the domain walls are pinned at twinning boundaries, D_m should be interpreted as the distance between the twinning boundaries, i.e. the span of the domain wall [3].

References

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26-28 September 2016/Limoges - IWAC07 -